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Density Functional Theory

Modulo: Funzionale Densità

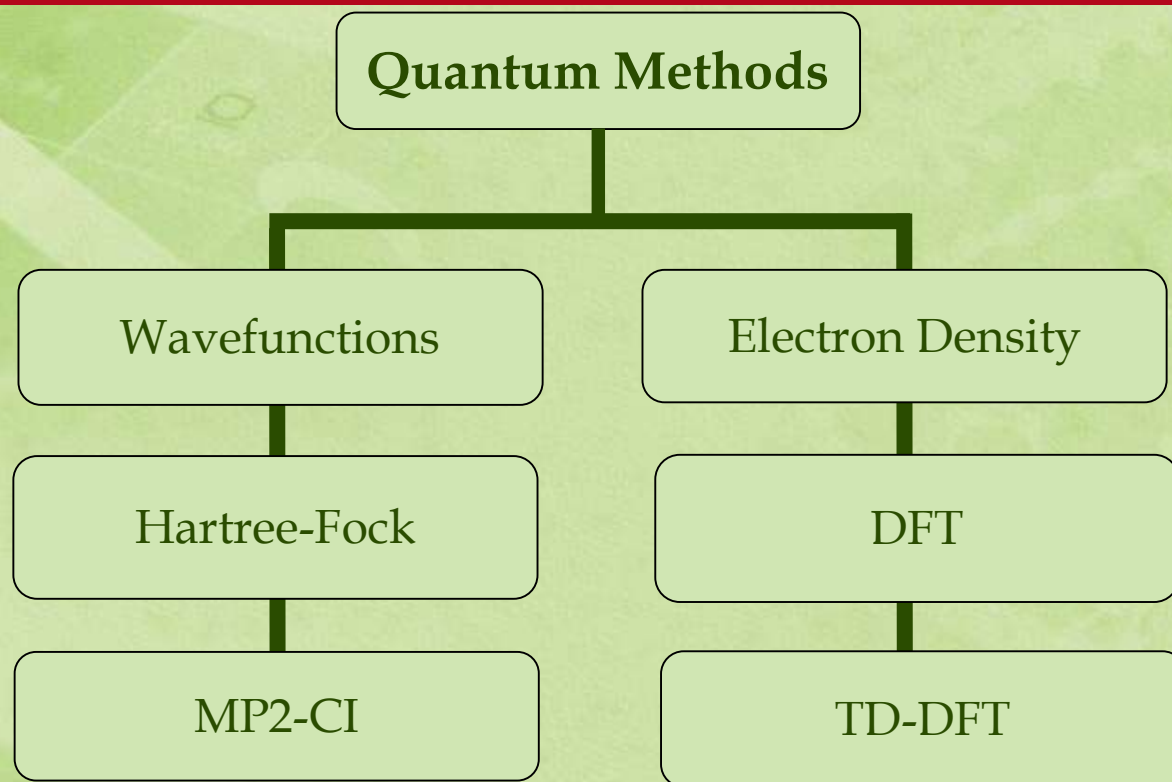
Chimica Computazionale
A.A. 2009-2010

Docente: Maurizio Casarin



Density functional theory (DFT) has revolutionized the quantum chemistry development of the last 20 years

DFT allows to get information about the energy, the structure and the molecular properties of molecules at lower costs than traditional approaches based on the wavefunction use.



The HF equations have to be solved iteratively because V_{HF} depends upon solutions (the orbitals). In practice, one adopts the LCAO scheme, where the orbitals are expressed in terms of N basis functions, thus obtaining matricial equations depending upon N^4 bielectron integrals.



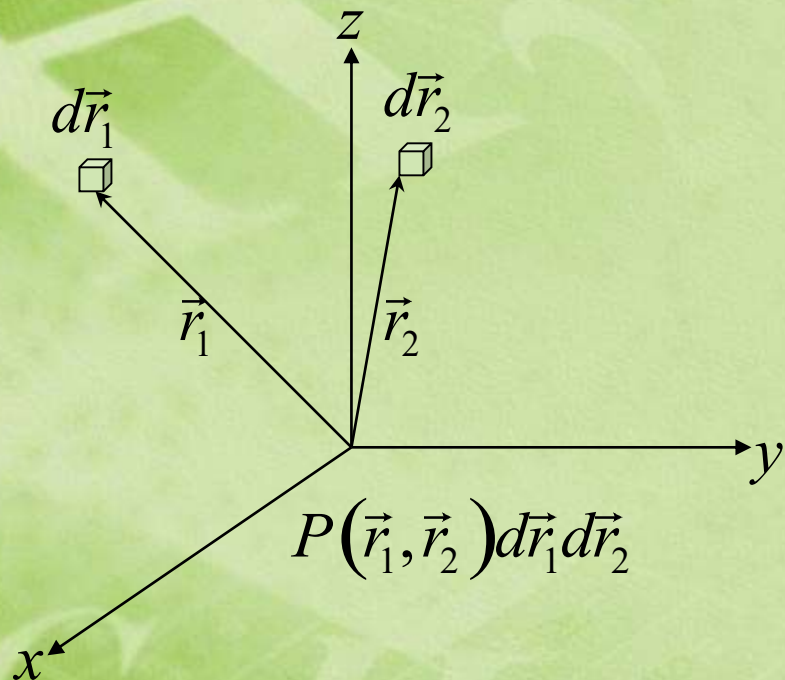
Correlation energy

Exchange correlation:

Electrons with the same spin (m_s) do not move independently as a consequence of the Pauli exclusion principle. $\Psi = 0$ if two electrons with the same spin occupy the same point in space, independently of their charge. HF theory treats exactly the exchange correlation generating a non local exchange correlation potential.

Coulomb correlation:

Electrons cannot move independently as a consequence of their Coulomb repulsion even though they are characterized by different spin (m_s). HF theory completely neglects the Coulomb correlation thus generating, in principle, significant mistakes. Post HF treatments are often necessary.



$$\Psi(\mathbf{x}_1, \mathbf{x}_2) = |\chi_1(\mathbf{x}_1)\chi_2(\mathbf{x}_2)\rangle$$

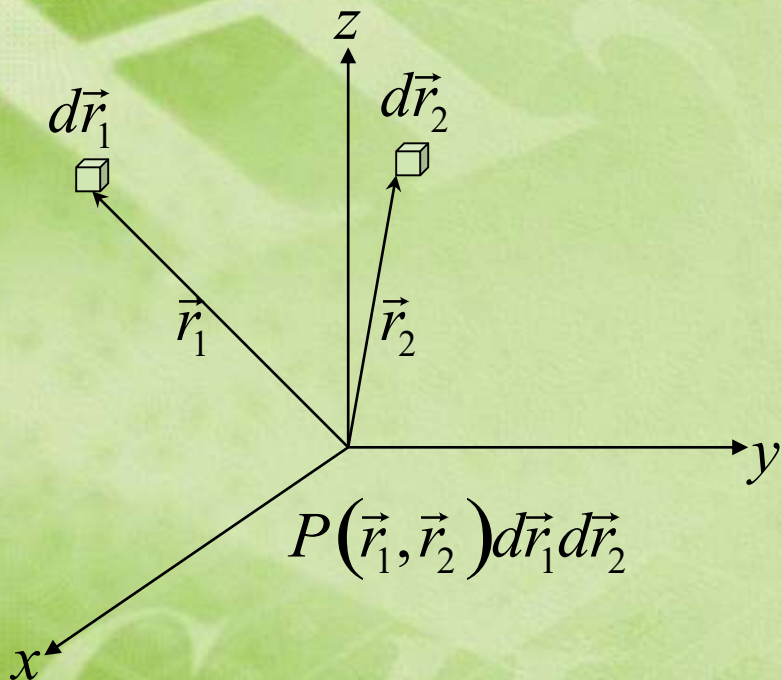
$$\chi_1(\mathbf{x}_1) = \psi_1(r_1)\alpha(\omega_1)$$

$$\chi_2(\mathbf{x}_2) = \psi_2(r_2)\beta(\omega_2)$$



$$\Psi(\mathbf{x}_1, \mathbf{x}_2) = |\chi_1(\mathbf{x}_1)\chi_2(\mathbf{x}_2)\rangle = \frac{1}{\sqrt{2!}} \begin{vmatrix} \chi_1(\mathbf{x}_1) & \chi_1(\mathbf{x}_2) \\ \chi_2(\mathbf{x}_1) & \chi_2(\mathbf{x}_2) \end{vmatrix}$$

$$|\Psi|^2 d\mathbf{x}_1 d\mathbf{x}_2 = \frac{1}{2} \left\{ \begin{array}{l} \psi_1(r_1)\alpha(\omega_1)\psi_2(r_2)\beta(\omega_2) - \\ \psi_1(r_2)\alpha(\omega_2)\psi_2(r_1)\beta(\omega_1) \end{array} \right\}^2 d\mathbf{x}_1 d\mathbf{x}_2$$



If the electrons have
not the same spin

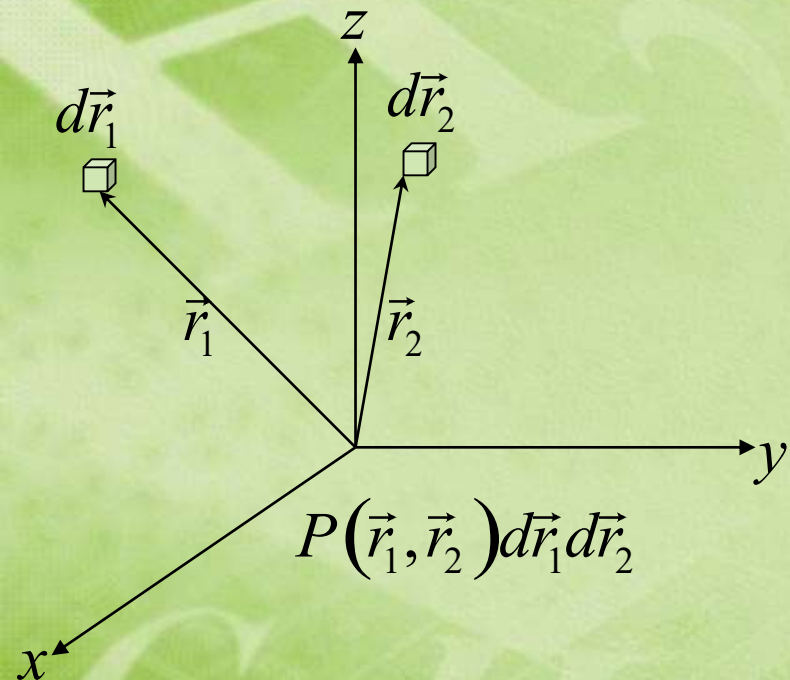


$$P(\mathbf{r}_1, \mathbf{r}_2) d\mathbf{r}_1 d\mathbf{r}_2 = \int d\omega_1 d\omega_2 |\Psi|^2 =$$
$$\frac{1}{2} \left[|\psi_1(\mathbf{r}_1)|^2 |\psi_2(\mathbf{r}_2)|^2 + |\psi_1(\mathbf{r}_2)|^2 |\psi_2(\mathbf{r}_1)|^2 \right] d\mathbf{r}_1 d\mathbf{r}_2$$

if $\psi_1 = \psi_2$

$$P(\mathbf{r}_1, \mathbf{r}_2) = |\psi_1(\mathbf{r}_1)|^2 |\psi_1(\mathbf{r}_2)|^2$$

$$P(\mathbf{r}_1, \mathbf{r}_1) = |\psi_1(\mathbf{r}_1)|^2 |\psi_1(\mathbf{r}_1)|^2 \neq 0$$



If the electrons have
the same spin



$$\chi_1(\mathbf{x}_1) = \psi_1(\mathbf{r}_1) \beta(\omega_1)$$

$$\chi_2(\mathbf{x}_2) = \psi_2(\mathbf{r}_2) \beta(\omega_2)$$

$$P(\mathbf{r}_1, \mathbf{r}_2) = \frac{1}{2} \left\{ \begin{array}{l} \left[|\psi_1(\mathbf{r}_1)|^2 |\psi_2(\mathbf{r}_2)|^2 + |\psi_1(\mathbf{r}_2)|^2 |\psi_2(\mathbf{r}_1)|^2 - \right. \\ \left. \left[\psi_1^*(\mathbf{r}_1) \psi_2(\mathbf{r}_1) \psi_2^*(\mathbf{r}_2) \psi_1(\mathbf{r}_2) + \right. \right. \\ \left. \left. \psi_1(\mathbf{r}_1) \psi_2^*(\mathbf{r}_1) \psi_2(\mathbf{r}_2) \psi_1^*(\mathbf{r}_2) \right] \right] \end{array} \right\}$$

$$P(\mathbf{r}_1, \mathbf{r}_1) = 0$$



Information provided by Ψ is redundant

benzene

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$N = 42e^-$

Number of terms in the determinantal form Ψ : $N! = 1.4 \times 10^{51}$

Number of Cartesian dimensions: $3N = 126$

Ψ is a very complex object including more information than we need!



The use of electron density allows to limit the redundant information

The electron density is a function of three coordinates no matter of the electron number.



Timetable

- 1920s: *Introduction of the Thomas-Fermi model.*
- 1964: *Hohenberg-Kohn paper proving existence of exact DF.*
- 1965: *Kohn-Sham scheme introduced.*
- 1970s and early 80s: *LDA. DFT becomes useful.*
- 1985: *Incorporation of DFT into molecular dynamics (Car-Parrinello)
(Now one of PRL's top 10 cited papers).*
- 1988: *Becke and LYP functionals. DFT useful for some chemistry.*
- 1998: *Nobel prize awarded to Walter Kohn in chemistry for development of DFT.*



Quotation: “If I have seen further [than certain other men] it is by standing upon the shoulders of giants.”*

Isaac Newton (1642–1727), British physicist, mathematician.
Letter to Robert Hooke, February 5, 1675.

*With reference to his dependency on Galileo’s and Kepler’s work in physics and astronomy.



- (a) Thomas, L. H. *Proc. Cambridge Philos. Soc.* **1927**, 23, 542;
- (b) Fermi, E. *Z. Phys.* **1928**, 48, 73;
- (c) Dirac, P. A. M. *Cambridge Philos. Soc.* **1930**, 26, 376;
- (d) Wigner, E. P. *Phys. Rev.* **1934**, 46, 1002.

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Thomas, L. H.
1903-1992

Fermi, E.
1901-1954

Dirac, P.M.A.
1902-1984

Wigner, E.
1902-1995



- (a) Hartree, D. R. *Proc. Cambridge Phil. Soc.* **1928**, 24, 89;
- (b) ibidem **1928**, 24, 111;
- (c) ibidem **1928**, 24, 426;
- (d) Fock, V. Z. *Physic* **1930**, 61, 126;
- (e) Slater, J. C. *Phys. Rev.* **1930**, 35, 210.

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Hartree, D.R.
1897-1958

Fock, D.R.
1898-1974

Slater J. C.
1900 -1976



Definitions

Function: a prescription which maps one or more numbers to another number:

$$y = f(x) = x^2$$



Definitions

Operator: a prescription which maps a function onto another function:

$$\hat{F} = \frac{\partial^2}{\partial x^2}$$

$$\hat{F}f(x) = \frac{\partial^2 f(x)}{\partial x^2}$$



Definitions

Functional: A functional takes a function as input and gives a number as output. An example is:

$$F[f(x)] = y$$

Here $f(x)$ is a function and y is a number. An example is the functional to integrate x from $-\infty$ to ∞ .

$$F[f] = \int_{-\infty}^{\infty} f(x) dx$$

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In base a un regio decreto emanato il 28 agosto 1931 i docenti delle università italiane avrebbero dovuto giurare di essere fedeli non solo allo statuto albertino e alla monarchia, ma anche al regime fascista.

*Francesco ed Edoardo Ruffini e Fabio Luzzatto (giuristi);
Giorgio Levi Della Vida (orientalista);
Gaetano De Sanctis (storico dell'antichità);
Ernesto Buonaiuti (teologo);
Vito Volterra (matematico);
Bartolo Nigrisoli (chirurgo);
Marco Carrara (antropologo);
Lionello Venturi (storico dell'arte);
Giorgio Errera (chimico);
Piero Martinetti (studioso di filosofia).*



Nel 1938, con la promulgazione delle Leggi razziali, perdettero il posto i professori considerati di origine ebraica in base alla normativa razziale

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ab-initio methods can be interpreted as a functional of the wavefunction, with the functional form completely known!

$$E[\Psi] = \frac{\int \Psi^*(x_1, L, x_N) \hat{H} \Psi(x_1, L, x_N) dx_1 L dx_N}{\int \Psi^*(x_1, L, x_N) \Psi(x_1, L, x_N) dx_1 L dx_N}$$



Thomas–Fermi-Dirac Model

Electrons are uniformly distributed over the phase space in cells of $(2\pi\hbar)^3$

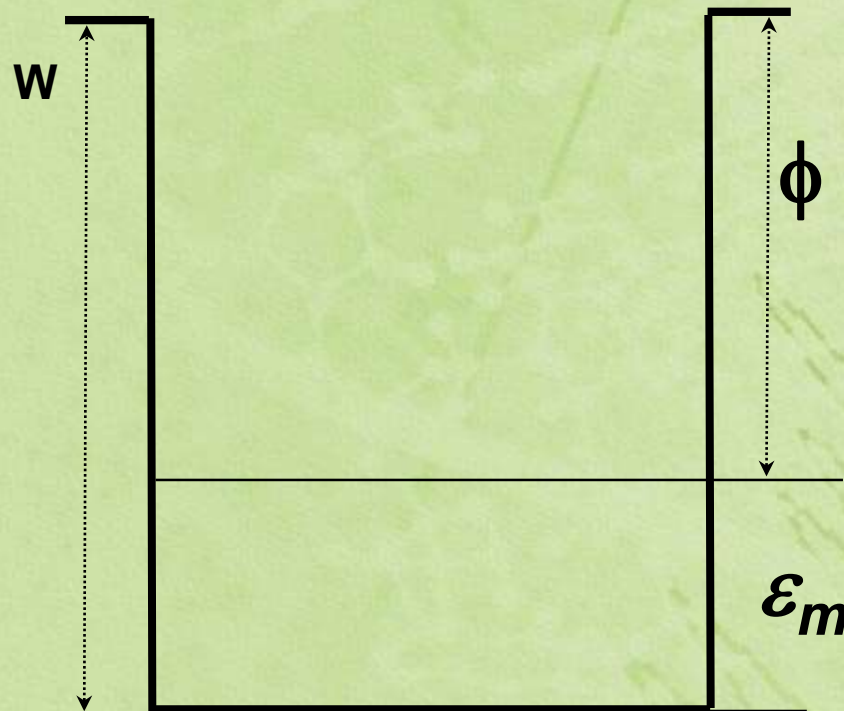
Each cell may contain up to two electrons with opposite spins

Electrons experience a potential field generated by the nuclear charge and by the electron distribution itself.



Let us consider a free electron gas

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Let us consider a free electron gas

$$\frac{\mathcal{N}_A}{V} = 6.023 \times 10^{23} \frac{\text{atoms}}{\text{mol}} \times \frac{8.92 \frac{\text{g}}{\text{cm}^3}}{63.5 \frac{\text{g}}{\text{mol}}} = 8.47 \times 10^{22} \frac{\text{electrons}}{\text{cm}^3}$$

$$\frac{\mathcal{N}_A}{V} = 6.023 \times 10^{23} \frac{\text{atoms}}{\text{mol}} \times \frac{Z\rho}{\text{at.weight}}$$

$$P = \frac{82.06 \left(\frac{\text{cm}^3 \text{atm}}{\text{molK}} \right) 293\text{K}}{7.11 \left(\frac{\text{cm}^3}{\text{mol}} \right)} = 3381 \text{atm}$$

$$7.11 \left(\frac{\text{cm}^3}{\text{mol}} \right) = \frac{6.023 \times 10^{23} \frac{\text{atoms}}{\text{mol}}}{8.47 \times 10^{22} \frac{\text{electrons}}{\text{cm}^3}}$$



$$\frac{V}{\mathcal{N}} = \frac{4\pi}{3} r_s^3 \qquad r_s = \left(\frac{3V}{4\pi\mathcal{N}} \right)^{\frac{1}{3}}$$

$$r_s = 1.72 \text{ \AA} \text{ (Li)}, 2.08 \text{ \AA} \text{ (Na)}, 1.12 \text{ \AA} \text{ (Fe)}$$

$$\text{Na}^+ \text{ radius} = 0.95 \text{ \AA} \qquad \left(\frac{r_s}{0.95} \right)^3 > 10$$

$$\varepsilon_m = \frac{mv_m^2}{2} = \frac{3}{2} k_B T$$

$$v_m = 10^7 \text{ cm / s}$$

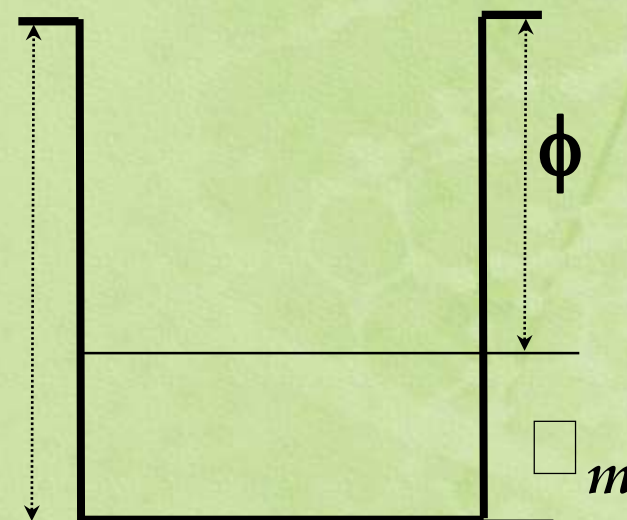


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Let see what happens when QM is applied

*Main Assumption:
independent electron approximation*

$$V(\mathbf{r}) = \text{constant}$$

$$\hat{H}\psi = \left[-\frac{\hbar^2}{2m} \nabla^2 + V(\vec{r}) \right] \psi = E\psi \qquad -\frac{\hbar^2}{2m} \nabla^2 \psi = E\psi$$

$$\frac{\hbar^2}{2m} k^2 \psi = E\psi$$

$$E = \frac{\hbar^2}{2m} k^2$$

$$\psi = A e^{i\vec{k} \cdot \vec{r}}$$



$$\psi = Ae^{i\vec{k}\cdot\vec{r}}$$

$$\hat{p}\psi \equiv -i\hbar \frac{\partial}{\partial r} \psi = -i\hbar \frac{\partial}{\partial r} \left(Ae^{i\vec{k}\cdot\vec{r}} \right) = \hbar\vec{k}\psi$$

De Broglie relation states that

$$p = \frac{h}{\lambda}$$

$$k = |\vec{k}| = \frac{2\pi}{\lambda}$$



Two kinds of boundary conditions

$$\psi(0) = \psi(L) = 0$$

$$\psi_n = \left(\frac{2}{L}\right)^{\frac{1}{2}} \sin\left(\frac{n\pi x}{L}\right)$$

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$$E_n = \frac{\hbar^2}{2m} \left(\frac{\pi}{L}\right)^2 n^2$$



$$\psi(x) = \psi(x + L)$$

$$\psi(x) = Ae^{ik_x x}$$

$$\psi(x + L) = Ae^{ik_x x} e^{ik_x L}$$

$$e^{ik_x L} = 1$$

$$k_x L = 2\pi\bar{m}$$

$$\psi_{\bar{m}}(x) = L^{-\frac{1}{2}} e^{i\frac{2\pi\bar{m}x}{L}}$$

$$E_{\bar{m}} = \frac{\hbar^2}{2m} k^2 = \frac{\hbar^2}{2m} \left(\frac{2\pi\bar{m}}{L} \right)^2$$

$$\bar{m} = 0, \pm 1, \pm 2, \dots$$



Moving to three dimensions

$$\psi_{\vec{m}}(x, y, z) = L^{-\frac{3}{2}} e^{i\left(\frac{2\pi}{L}\right)(\bar{m}_x x + \bar{m}_y y + \bar{m}_z z)}$$

$$E_{\vec{m}} = \frac{\hbar^2}{2m} k^2 = \frac{\hbar^2}{2m} \left(\frac{2\pi}{L}\right)^2 (\bar{m}_x^2 + \bar{m}_y^2 + \bar{m}_z^2)$$

$$\bar{m}_x, \bar{m}_y, \bar{m}_z = 0, \pm 1, \pm 2, \dots$$

$$\vec{k} = \frac{2\pi}{L} \vec{m}$$



$$\vec{k} = \frac{2\pi}{L} \vec{m}$$

$$\psi_{\vec{m}}(x, y, z) = L^{-\frac{3}{2}} e^{i\left(\frac{2\pi}{L}\right)(\bar{m}_x x + \bar{m}_y y + \bar{m}_z z)}$$

$$\psi_k = V^{-\frac{1}{2}} e^{i\vec{k} \cdot \vec{r}}$$

$$E_{\vec{m}} = \frac{\hbar^2}{2m} \left(\frac{2\pi}{L} \right)^2 \left(\bar{m}_x^2 + \bar{m}_y^2 + \bar{m}_z^2 \right)$$

$$E_{\vec{m}} = \frac{\hbar^2}{2m} \left(k_x^2 + k_y^2 + k_z^2 \right)$$



Degeneracies of free electron levels

Typical possibilities				Orbital degeneracy	Total degeneracy
\bar{m}_x	\bar{m}_y	\bar{m}_z	\bar{m}^2		
0	0	0	0	1	2
± 1	0	0	1	6	12
± 1	± 1	0	2	12	24
± 1	± 1	± 1	3	8	16
± 2	0	0	4	6	12

For large m values the degeneracies go up as

$$\left(\bar{m}^2\right)^{\frac{1}{2}}$$



$$\bar{m}_F = \left| \vec{m}_{\max} \right|$$

Energy, temperature and velocity of electrons with m_F

$$\hat{p}\psi \equiv -i\hbar \frac{\partial}{\partial r} \psi = -i\hbar \frac{\partial}{\partial r} \left(A e^{i\vec{k} \cdot \vec{r}} \right) = \hbar \vec{k} \psi$$

$$N = 2 \left(\frac{4\pi}{3} \right) \bar{m}_F^3 = 2 \left(\frac{4\pi}{3} \right) \frac{V}{(2\pi)^3} k_F^3 \quad \vec{k} = \frac{2\pi}{L} \vec{m}$$

$$E_F = \frac{\hbar^2}{2m} k_F^2 = \frac{\hbar^2}{2m} \left(\frac{2\pi}{L} \right)^2 \bar{m}_F^2 = \frac{\hbar^2}{2m} \left(\frac{3\pi^2 N}{V} \right)^{\frac{2}{3}}$$



If we assume that the number of electrons per unit volume is ρ_0 , then the Fermi momentum p_F of a uniform free electron gas is:

$$E_F = \frac{p_F^2}{2m} = \frac{\hbar^2}{2m} \left(\frac{3\pi^2 N}{V} \right)^{\frac{2}{3}}$$

$$p_F = \hbar \left(\frac{3\pi^2 N}{V} \right)^{\frac{1}{3}}$$

$$\rho_0 = \frac{N}{V} = \frac{p_F^3}{3\pi^2 \hbar^3}$$



Thomas and Fermi applied such a relation to an *inhomogeneous* situation as that of atoms, molecules and solids. If the inhomogeneous electron density is denoted by $\rho(\vec{r})$, when the equation defining ρ_0 is applied locally at \vec{r} , it yields

$$\rho(\vec{r}) = \frac{p_F^3(\vec{r})}{3\pi^2 \hbar^3}$$



Let us define the density of states $g(\varepsilon)$, i.e. the number of states between ε and $\varepsilon + d\varepsilon$

The number of states with energy up to ε is

$$2 \left(\frac{4\pi}{3} \right) \bar{m}^3 = 2 \left(\frac{4\pi}{3} \right) \frac{V}{(2\pi)^3} k^3 = \left(\frac{V}{3\pi^2} \right) \left(\frac{2m\varepsilon}{\hbar^2} \right)^{\frac{3}{2}} \quad E_F = \frac{\hbar^2}{2m} k_F^2$$

$$\int_0^\varepsilon g(\varepsilon) d\varepsilon = 2 \left(\frac{4\pi}{3} \right) \bar{m}^3 = 2 \left(\frac{4\pi}{3} \right) \frac{V}{(2\pi)^3} k^3 = \left(\frac{V}{3\pi^2} \right) \left(\frac{2m\varepsilon}{\hbar^2} \right)^{\frac{3}{2}}$$



$$\int_0^{\varepsilon} g(\varepsilon) d\varepsilon = 2 \left(\frac{4\pi}{3} \right) \bar{m}^3 = 2 \left(\frac{4\pi}{3} \right) \frac{V}{(2\pi)^3} k^3 = \left(\frac{V}{3\pi^2} \right) \left(\frac{2m\varepsilon}{\hbar^2} \right)^{\frac{3}{2}}$$

$$g(\varepsilon) = \frac{3}{2} \left(\frac{V}{3\pi^2} \right) \left(\frac{2m}{\hbar^2} \right)^{\frac{3}{2}} \varepsilon^{\frac{1}{2}} = \sqrt{2} \left(\frac{V}{\pi^2} \right) \left(\frac{m}{\hbar^2} \right)^{\frac{3}{2}} \varepsilon^{\frac{1}{2}} = C \varepsilon^{\frac{1}{2}}$$



T_0 = total kinetic energy

$$T_0 = \int_0^{\varepsilon_F} \varepsilon g(\varepsilon) d\varepsilon = C \int_0^{\varepsilon_F} \varepsilon^{\frac{3}{2}} d\varepsilon = \frac{2}{5} C \varepsilon_F^{\frac{5}{2}} \quad p_F = \hbar \left(\frac{3\pi^2 N}{V} \right)^{\frac{1}{3}}$$

$$\varepsilon_F = \frac{p_F^2}{2m} = \frac{\hbar^2}{2m} \left(\frac{3\pi^2 N}{V} \right)^{\frac{2}{3}} \quad (\varepsilon_F)^{\frac{3}{2}} = \left(\frac{\hbar^2}{2m} \right)^{\frac{3}{2}} \left(\frac{3\pi^2 N}{V} \right)$$


$$T_0 = \frac{2}{5} C \varepsilon_F^{\frac{5}{2}} = \frac{2}{5} \sqrt{2} \left(\frac{V}{\pi^2} \right) \left(\frac{m}{\hbar^2} \right)^{\frac{3}{2}} \left(\frac{\hbar^2}{2m} \right)^{\frac{3}{2}} \left(\frac{3\pi^2 N}{V} \right) \varepsilon_F = \frac{3}{5} N \varepsilon_F$$



$$T_0 = \frac{3}{5} N \varepsilon_F$$

$$T_0 = \frac{2}{5} C \varepsilon_F^{\frac{5}{2}}$$

$$T_0[\rho_0] = \frac{2}{5} \underbrace{\sqrt{2} \left(\frac{V}{\pi^2} \right) \left(\frac{m}{\hbar^2} \right)^{\frac{3}{2}}}_{C} \left(\frac{\hbar^2}{2m} (3\pi^2)^{\frac{2}{3}} \rho_0^{\frac{2}{3}} \right)^{\frac{5}{2}}$$



$$T_0[\rho_0] = \frac{3h^2}{10m} \left(\frac{3}{8\pi} \right)^{\frac{2}{3}} V \rho_0^{\frac{5}{3}} \quad T[\rho(r)] = \frac{3h^2}{10m} \left(\frac{3}{8\pi} \right)^{\frac{2}{3}} V \rho(r)^{\frac{5}{3}}$$



One can next write the classical energy equation for the fastest electrons as

$$\mu = \frac{p_F^2(\vec{r})}{2m} + V(\vec{r})$$

$$\rho(\vec{r}) = \frac{p_F^3(\vec{r})}{3\pi^2\hbar^3} \quad \hbar(3\pi^2)^{\frac{1}{3}} \{\rho(\vec{r})\}^{\frac{1}{3}} = p_F(\vec{r})$$

$$\mu = \left(\frac{\hbar^2}{2m} \right) (3\pi^2)^{\frac{2}{3}} \{\rho(\vec{r})\}^{\frac{2}{3}} + V(\vec{r})$$

The basic equation of the TF theory. It is a classical expression, and consequently it can be applied only in those cases for which $\mu - V > 0$



$$T[\rho(\mathbf{r})] = \frac{3h^2}{10m} \left(\frac{3}{8\pi} \right)^{\frac{2}{3}} |\rho(\mathbf{r})|^{\frac{5}{3}}$$

$$T_{TF}[\rho(\mathbf{r})] = C_{TF} \int \rho^{\frac{5}{3}}(\mathbf{r}) d\mathbf{r}$$

$$E_{TF}[\rho(\vec{r})] = C_{TF} \int \rho^{\frac{5}{3}}(\vec{r}) d\vec{r} - Z \int \frac{\rho(\vec{r})}{r} d\vec{r} + \frac{1}{2} \iint \frac{\rho(\vec{r}_1)\rho(\vec{r}_2)}{|\vec{r}_1 - \vec{r}_2|} d\vec{r}_1 d\vec{r}_2$$



A results of the free electron gas theory is that the mean kinetic energy per particle is $3/5$ of the Fermi energy. The total kinetic energy T_0 of a free electron gas constituted by N particles is then:

$$T_0 = \frac{3}{5} \left(\frac{p_F^2}{2m} \right) N$$

and hence, t_0 (the kinetic energy per unit volume) is

$$t_0 = \frac{T_0}{V} = \frac{3}{5} \left(\frac{p_f^2}{2m} \right) \rho_0 = C_k \rho_0^{5/3}, \quad C_k = \frac{3\hbar^2}{10m} \left(3\pi^2 \right)^{2/3}$$



$$t = C_k \left\{ \rho(\vec{r}) \right\}^{5/3}$$

$$E = C_k \int \left\{ \rho(\vec{r}) \right\}^{5/3} d\vec{r} + \int \rho(\vec{r}) V_M(\vec{r}) d\vec{r} + \frac{e^2}{2} \iint \frac{\rho(\vec{r}) \rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d\vec{r} d\vec{r}'$$

The physical meaning of the last equation is that the electronic properties of a system are determined as **functionals** of the electronic density by applying, locally, relations appropriate to a homogeneous free electron gas. This approximation, known as local density approximation (LDA), is probably one of the most important concept of the modern DFT!



$$N = N[\rho(\mathbf{r})] = \int \rho(\mathbf{r}) d\mathbf{r}$$

$$\delta \left\{ E_{TF}[\rho] - \mu_{TF} \left(\int \rho(\mathbf{r}) d\mathbf{r} - N \right) \right\} = 0$$

$$\mu_{TF} = \frac{\delta E_{TF}[\rho]}{\delta \rho(\mathbf{r})} = \frac{5}{3} C_F \rho^{\frac{2}{3}}(\mathbf{r}) - \Phi(\mathbf{r})$$

$$\Phi(\mathbf{r}) = \frac{Z}{r} - \int \frac{\rho(\mathbf{r}_2)}{|\mathbf{r} - \mathbf{r}_2|} d\mathbf{r}_2$$



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$$\hat{v}_X^{HF} \approx v_{X\alpha} \left(\left[\rho(\mathbf{r}) \right]; \mathbf{r} \right) = -\frac{3}{2} \alpha \left[\frac{3}{\pi} \rho(\mathbf{r}) \right]^{\frac{1}{3}}$$



A *non local* operator is characterized by the general equation

$$\langle \vec{r}' | \hat{A} | \Psi \rangle = \int d\vec{r} \hat{A}(\vec{r}', \vec{r}) \Psi(\vec{r}) = \Psi'(\vec{r}') \quad \hat{A}(\vec{r}', \vec{r}) = \langle \vec{r}' | \hat{A} | \vec{r} \rangle$$

$$\langle \vec{r}' | \hat{A} | \vec{r} \rangle = \hat{A}(\vec{r}) \delta(\vec{r}' - \vec{r})$$



The Kohn-Sham method

$$T = \sum_{i=1}^N q_i \langle \psi_i | -\frac{1}{2} \nabla^2 | \psi_i \rangle$$

$$\rho(\vec{r}) = \sum_{i=1}^N q_i \sum_s |\psi_i(\vec{r}, s)|^2$$

$$T_s = \sum_{i=1}^N \langle \psi_i | -\frac{1}{2} \nabla^2 | \psi_i \rangle$$

$$\rho(\vec{r}) = \sum_{i=1}^N \sum_s |\psi_i(\vec{r}, s)|^2$$



In ogni punto si associa alla densità $\rho(\mathbf{r})$ l'energia XC che avrebbe un gas elettronico uniforme con la stessa densità. Ciò è ripetuto per ogni punto e i valori usati nelle formule

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decompressor
are needed to see this picture.



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The seven major kings of science

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Theodor
von Kàrmàn

George
de Hevesy

Michael
Polanyi

Leo Szilard

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Eugene
Wigner

John
von Neumann

Edward
Teller



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QuickTime™ and a
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are needed to see this picture.

QuickTime™ and a
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are needed to see this picture.

P. Hohenberg

W. Kohn



$$E_{HF} = \langle \Psi_{HF} | \hat{H} | \Psi_{HF} \rangle = \sum_{i=1}^N H_i + \frac{1}{2} \sum_{i,j=1}^N (J_{ij} - K_{ij})$$

$$H_i = \int \psi_i^*(\mathbf{x}) \left[-\frac{1}{2} \nabla^2 + v(\mathbf{x}) \right] \psi_i(\mathbf{x}) d\mathbf{x}$$

$$J_{ij} = \iint \psi_i(\mathbf{x}_1) \psi_i^*(\mathbf{x}_1) \frac{1}{r_{12}} \psi_j^*(\mathbf{x}_2) \psi_j(\mathbf{x}_2) d\mathbf{x}_1 d\mathbf{x}_2$$

$$K_{ij} = \iint \psi_i^*(\mathbf{x}_1) \psi_j(\mathbf{x}_1) \frac{1}{r_{12}} \psi_i(\mathbf{x}_2) \psi_j^*(\mathbf{x}_2) d\mathbf{x}_1 d\mathbf{x}_2$$



Minimization and orthonormalization conditions

$$\hat{P}\psi_i(\mathbf{x}) = \sum_{j=1}^N \varepsilon_{ij} \psi_j(\mathbf{x})$$

$$\hat{P}\psi = -\frac{1}{2} \nabla^2 \psi + v \psi + \mathcal{G}\psi$$



Background

- 1920s: **Introduction of the Thomas-Fermi model.**
- 1964: **Hohenberg-Kohn paper proving existence of exact DF.**
- 1965: Kohn-Sham scheme introduced.
- 1970s and early 80s: LDA. DFT becomes useful.
- 1985: Incorporation of DFT into molecular dynamics (Car-Parrinello) (Now one of PRL's top 10 cited papers).
- 1988: Becke and LYP functionals. DFT useful for some chemistry.
- 1998: Nobel prize awarded to Walter Kohn in chemistry for development of DFT.



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decompressor
are needed to see this picture.



The first HK theorem legitimates ρ as basic variable.

The external potential is determined, within a trivial additive constant, by the electron density.

$$E_v[\rho] = T[\rho] + V_{em}[\rho] + V_{ee}[\rho] = \int \rho(\vec{r}) v(\vec{r}) d\vec{r} + F_{HK}[\rho]$$

$$F_{HK}[\rho] = T[\rho] + V_{ee}[\rho]$$

$$V_{ee}[\rho] = J[\rho] + \text{non classical terms}$$



The second HK theorem supplies the variational principle for the ground state energy.

$$\int \tilde{\rho}(\vec{r}) d\vec{r} = N$$

$$E_0 \leq E_v[\tilde{\rho}]$$

$$\delta \left\{ E_v[\rho] - \mu \left[\int \rho(\vec{r}) d\vec{r} - N \right] \right\} = 0$$

The ground state energy and density correspond to the minimum of some functional E_v subject to the constraint that the density contains the correct number of electrons. The Lagrange multiplier of this constraint is the electronic chemical potential μ .



In matematica e in fisica teorica, la derivata funzionale è una generalizzazione della derivata direzionale. La differenza è che la seconda differenzia nella direzione di un vettore, mentre la prima differenzia nella direzione di una funzione. Entrambe possono essere viste come estensioni dell'usuale derivata.

$$F[\rho] = \int f(\vec{r}, \rho, \nabla\rho, \nabla^2\rho, \dots) d^3r$$

$$\frac{\delta F[\rho]}{\delta\rho} = \frac{\delta f}{\delta\rho} - \nabla \cdot \frac{\delta f}{\delta(\nabla\rho)} + \nabla^2 \frac{\delta f}{\delta(\nabla^2\rho)} - \dots$$



$$J[\rho] = \frac{1}{2} \int \left(\frac{\rho(\vec{r})\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3 r' \right) d^3 r$$

$$\frac{\delta J[\rho]}{\delta \rho} = \frac{\delta j}{\delta \rho} = \int \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3 r'$$

$$j = \frac{1}{2} \int \frac{\rho(\vec{r})\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3 r'$$

$$\frac{\delta^2 J[\rho]}{\delta \rho^2} = \frac{\delta}{\delta \rho} \int \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d^3 r' = \frac{\delta}{\delta \rho} \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} = \frac{1}{|\vec{r} - \vec{r}'|}$$



$$E_v[\rho] = T[\rho] + V_{em}[\rho] + V_{ee}[\rho] = \int \rho(\vec{r})v(\vec{r})d\vec{r} + F_{HK}[\rho]$$

$$\mu = \frac{\delta E_v[\rho]}{\delta \rho(\vec{r})} = v(\vec{r}) + \frac{\delta F_{HK}[\rho]}{\delta \rho(\vec{r})}$$



Despite the importance of the HK theorems, it is noteworthy that the result they give is somehow incomplete. Actually, the first HK theorem refers only to the ground state energy and ground state density. Furthermore, as far as the second HK theorem is concerned, it is *simply* an existence theorem and no information about how to get the ground state energy functional is provided. Nevertheless, the existence of an exact theory justifies the research of new functionals that, though approximate version of the correct one, are more and more accurate.



Background

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The Kohn-Sham method.

The massive usage of DFT is tightly bound to its use in orbitalic theories. This is not very surprising because of the role played by these theories, in particular the HF one, in quantum chemistry. Thus, the major DFT developments have implied either the improvement of existing orbitalic theories, for instance the $X\alpha$ method [Slater, 1951a-b], or the proposal of new approaches [Kohn & Sham, 1965].

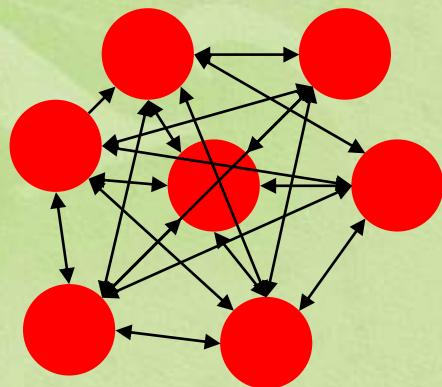


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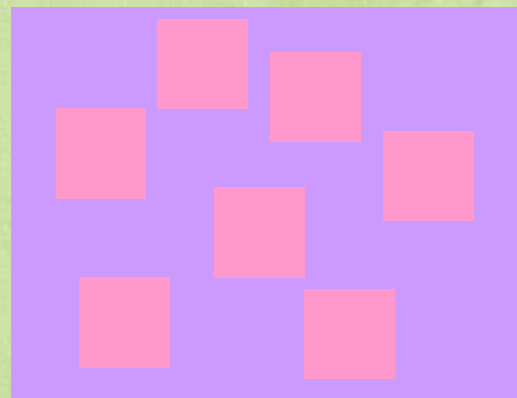
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Interacting electrons
+ real potential



Non-interacting fictitious
particles + effective potential



$$\hat{H}_s = \sum_i^N \left(-\frac{1}{2} \nabla_i^2 \right) + \sum_i^N v_s(\vec{r}_i)$$

$$\Psi_s = \frac{1}{\sqrt{N!}} \det [\psi_1 \psi_2 \cdots \psi_N]$$

$$\hat{h}_s \psi_i = \left[-\frac{1}{2} \nabla^2 + v_s(\vec{r}) \right] \psi_i = \varepsilon_i \psi_i$$

$$T_s[\rho] = \langle \Psi_s | \sum_i^N \left(-\frac{1}{2} \nabla_i^2 \right) | \Psi_s \rangle = \sum_i^N \langle \psi_i | -\frac{1}{2} \nabla^2 | \psi_i \rangle$$



$$F[\rho] = T_s[\rho] + J[\rho] + E_{xc}[\rho] \quad J[\rho] = \frac{1}{2} \iint \frac{1}{r_{12}} \rho(\vec{r}_1) \rho(\vec{r}_2) d\vec{r}_1 d\vec{r}_2$$

$$E_{xc}[\rho] \equiv T[\rho] - T_s[\rho] + V_{ee}[\rho] - J[\rho]$$

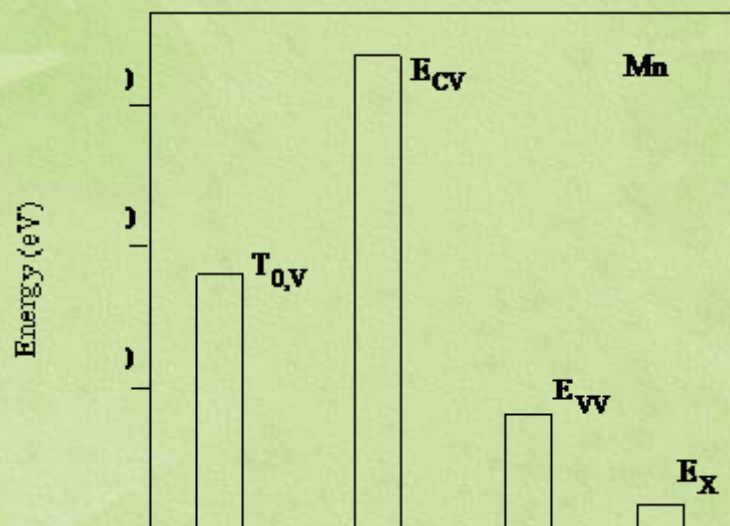
$$\mu = \frac{\delta E_v[\rho]}{\delta \rho(\vec{r})} = v(\vec{r}) + \frac{\delta F_{HK}[\rho]}{\delta \rho(\vec{r})} \quad \mu = v_{eff}(\vec{r}) + \frac{\delta T_s[\rho]}{\delta \rho(\vec{r})}$$

$$v_{eff}(\vec{r}) = v(\vec{r}) + \frac{\delta J[\rho]}{\delta \rho(\vec{r})} + \frac{\delta E_{xc}[\rho]}{\delta \rho(\vec{r})} = v(\vec{r}) + \int \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} d\vec{r}' + v_{xc}(\vec{r})$$

$$v_{xc}(\vec{r}) = \frac{\delta E_{xc}[\rho]}{\delta \rho(\vec{r})}$$



With reference to the single Euler - Lagrange equation, the introduction of N orbitals allows us to treat exactly T_s , the dominant part of the true kinetic energy $T[\rho]$. The cost we have to pay is the needed of N equations rather than one expressed in terms of the total electron density. The KS equations have the same form of the Hartree equations unless the presence of a more general local potential, $v_{eff}(\vec{r})$. The computational effort for their solution is comparable to that required for the Hartree equations and definitely smaller than that pertinent to the HF ones. HF equations are characterized by a one-electron Hamiltonian including a non local potential and for this reason they cannot be considered a special case of the KS equations.



Relative magnitudes of contributions to total valence energy (in eV) of the Mn atom



The exchange-correlation potential

While DFT in principle gives a good description of ground state properties, practical applications of DFT are based on approximations for the so-called exchange-correlation potential. The exchange-correlation potential describes the effects of the Pauli principle and the Coulomb potential beyond a pure electrostatic interaction of the electrons.

Possessing the exact exchange-correlation potential means that we solved the many-body problem exactly.

A common approximation is the so-called local density approximation (LDA) which locally substitutes the exchange-correlation energy density of an inhomogeneous system by that of an electron gas evaluated at the local density.



The Local Density Approximation (LDA)

The LDA approximation assumes that the density is slowly varying and the inhomogeneous density of a solid or molecule can be calculated using the homogeneous electron gas functional.

While many ground state properties (lattice constants, bulk moduli, etc.) are well described in the LDA, the dielectric constant is overestimated by 10-40% in LDA compared to experiment. This overestimation stems from the neglect of a polarization-dependent exchange correlation field in LDA compared to DFT.

The method can be improved by including the gradient of the density into the functional. The generalized gradient approximation GGA is an example of this type of approach.



The Slater exchange functional

The predecessor to modern DFT is Slater's X_α method. This method was formulated in 1951 as an approximate solution to the Hartree-Fock equations. In this method the HF exchange was approximated by:

$$E_{X_\alpha}[\rho] = -\frac{9}{4}\alpha\left(\frac{3}{4\pi}\right)^{1/3}\int_0^\infty \rho^{4/3}(r)dr$$

The exchange energy E_{X_α} is a fairly simple function of the electron density ρ .

The adjustable parameter α was empirically determined for each atom in the periodic table. Typically α is between 0.7 and 0.8. For a free electron gas $\alpha = 2/3$.



The VWN Correlation Functional

In ab initio calculations of the Hartree-Fock type electron correlation is also not included. However, it can be included by inclusion of configuration interaction (CI). In DFT calculations the correlation functional plays this role. The Vosko-Wilk-Nusair correlation function is often added to the Slater exchange function to make a combination exchange-correlation functional.

$$E_{xc} = E_x + E_c$$

The nomenclature here is not standardized and the correlation functionals themselves are very complicated functions. The correlation functionals can be seen on the MOLPRO website

<http://www.molpro.net/molpro2002.3/doc/manual/node146.html>.



Application of the LDA

of a
resistor
picture.

Application of LDA methods to semi-conductor materials and insulators gives good agreement for the lattice constant and bulk modulus.

The lattice constants are typically accurate to within 1-2% up the second row in the periodic table. Since the crystal volume V is accurately calculated the density is, of course, also obtained.

The bulk modulus is:

$$B = -V \left(\frac{\partial P}{\partial V} \right) = -V \left(\frac{\partial^2 E}{\partial V^2} \right)$$

Bulk moduli are calculated by systematically varying the lattice parameters and plotting the energy as a function of V . The curvature at the minimum of the $E(V)$ plot is proportional to the lattice constant.

Extensions of the LDA approach

Generalized gradient approximation (GGA). Take density gradient into account. Useful for molecules.

Spin density functional theory. Two independent variables: density and magnetization.

$$m(r) = -\mu_0(\rho_{\uparrow} - \rho_{\downarrow})$$

Exact exchange density functional theory. Calculate exchange exactly and correlation approximately using DFT.

Generalized density functional theory. Modify K-S energy partitioning to obtain a non-local hamiltonian.

Generalized Gradient Approach (GGA)

The GGA approach takes into account variations in the density by including the gradient of the density in the functional. One commonly used GGA functional is that of Becke.

$$V_{xc}^B = -\beta\rho^{1/3} \frac{x^2}{\left(1 + 6\beta x \sinh^{-1} x\right)}, \quad x = \frac{|\nabla\rho|}{\rho^{4/3}}$$

This functional has only one adjustable parameter, β . The value of $\beta = 0.0042$ was determined based on the best fit to the energies of six noble gas atoms using the sum of the LDA and GGA exchange terms.

The GGA option in DMol3 is that of Perdew and Wang.

Lee-Yang-Parr Correlation Functional

As was discussed above for the Slater exchange functional (no gradient), the VWN correlation functional provides a significant improvement in the calculation of the energies and properties such as bulk modulus, vibrational frequencies etc. In a similar manner the Becke exchange functional (including a gradient correlation) and the Lee-Yang-Parr functional are used together. The Lee-Yang-Parr or LYP correlation functional is quite complicated. It can be viewed on the MOLPRO website.

Thus, two of the most commonly used functionals are:
S-VWN Slater exchange - VWN correlation (no gradients)
B-LYP Becke exchange - LYP correlation (gradients)

Integrated Computational Materials Engineering Education

Lecture on Density Functional Theory *An Introduction*

Mark Asta

Department of Materials Science and Engineering
University of California, Berkeley
Berkeley, CA 94720

Acknowledgements

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- The Division of Materials Research at the National Science Foundation is acknowledged for financial support in the development of the lecture and module

Use of DFT in Materials Research

Table I. Tools Cited by Employers, Categorized and Ranked by the (Normalized) Frequency of Cites, along with Corresponding Results from the Computational Faculty Survey

Category	Example	Employer	Faculty ⁺
Mechanics (mostly FEA)	DEFORM, ABAQUS	80%	14%
Thermodynamics (CALPHAD)	ThermoCalc, Pandat	53%	7%
Density Functional Theory	VASP, ABNIT	47%	21%
Programming Language/Integration	Matlab, Fortran, iSight	40%	43%
Casting	ProCAST, MAGMAsoft	40%	—
Molecular Dynamics/ Monte Carlo	LAMMPS	27%	14%
Fluid Flow/ Heat Transfer	COMSOL, Fluent	20%	7%
Diffusion/ Microstructural Evolution	DICTRA, PrecipiCalc, JMatPro	20%	—
Statistics	Informatics	13%	7%
Materials Modeling Suite	Materials Studio	13%	—
General Visualization	Mathematica, Tecplot	7%	29%
General Data Processing	Spreadsheet	7%	21%
Special Purpose	K-Flow, WARP 3D	7%	—
Materials Selection	CES Materials Selector	—	36%
Crystallography	CaRIne	—	7%

⁺Some of the responses did not provide specific software/categories, and therefore we expect some degree of undercounting in this data.

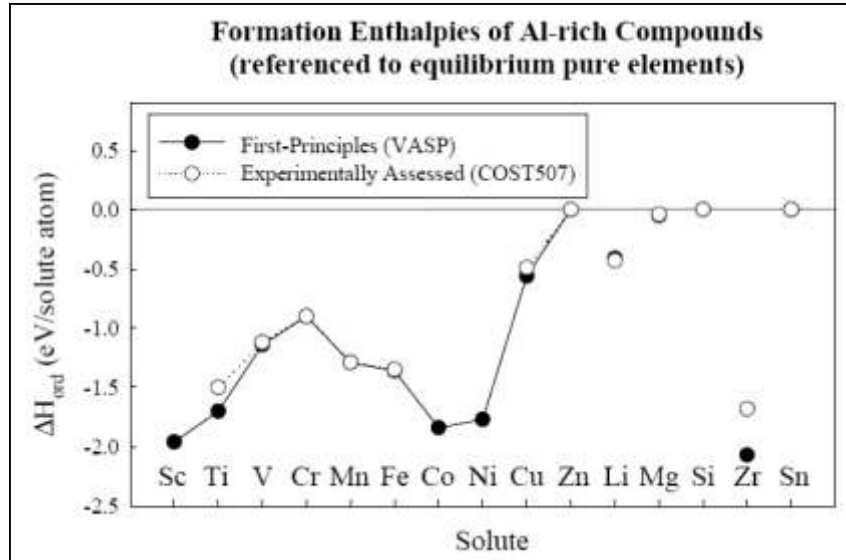
K. Thornton, S. Nola, R. E. Garcia, MA and G. B. Olson, “*Computational Materials Science and Engineering Education: A Survey of Trends and Needs*,” JOM (2009)

The Role of Electronic Structure Methods in ICME

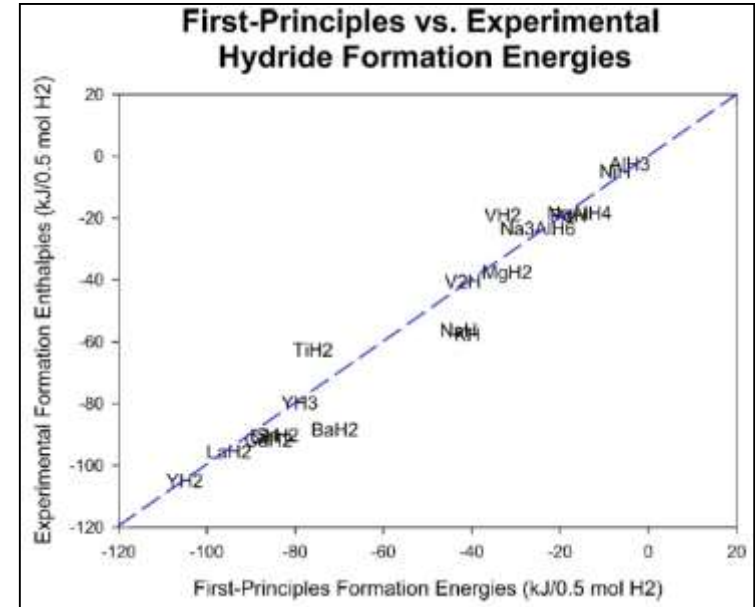
- A wide variety of relevant properties can be calculated from knowledge of atomic numbers alone
 - Elastic constants
 - Finite-temperature thermodynamic and transport properties
 - Energies of point, line and planar defects
- For many classes of systems accuracy is quite high
 - Can be used to obtain “missing” properties in materials design when experimental data is lacking, hard to obtain, or “controversial”
 - Can be used to discover new stable compounds with target properties
- The starting point for “hierarchical multiscale” modeling
 - Enables development of interatomic potentials for larger-scale classical modeling

Accuracy for Some Selected Materials

C. Wolverton and V. Ozolins
(Phys Rev B, 2005)



C. Wolverton, V. Ozolins, MA
(Phys Rev B, 2004)



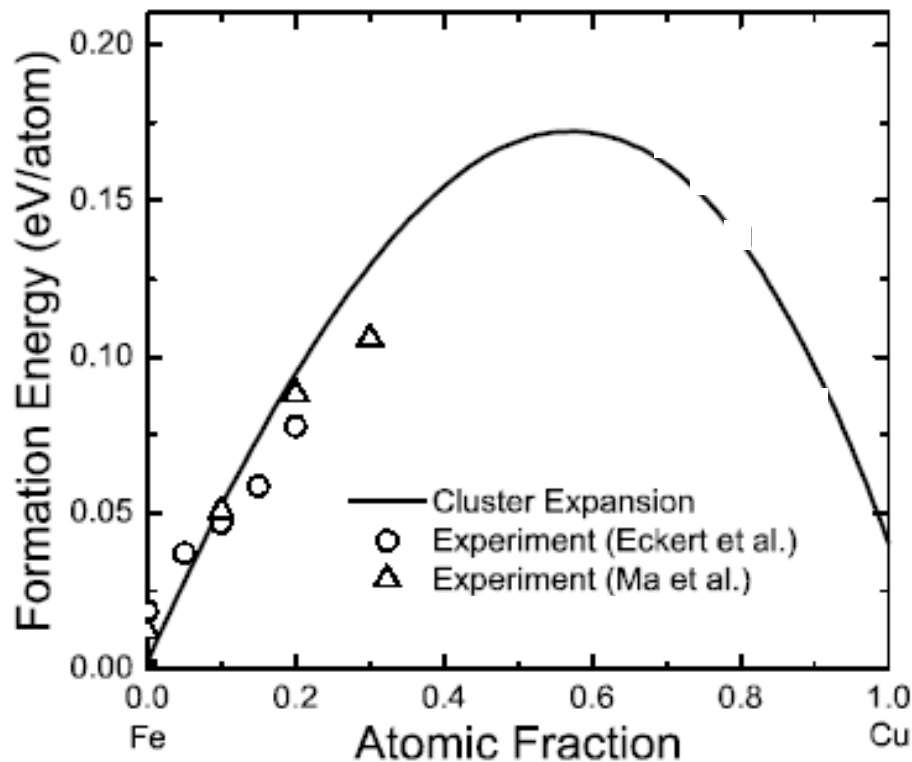
~95 % Success in High Throughput
Study Comparing Predicted and
Observed Stable Compounds for 80
Binary Systems
S. Curtarolo et al., CALPHAD (2004)

*DFT Provides Accurate
Framework for Predicting Alloy
Phase Stability and Defect
Energetics for Wide Range of
Alloy Systems*

1st-Principles Modeling of Alloy Phase Stability

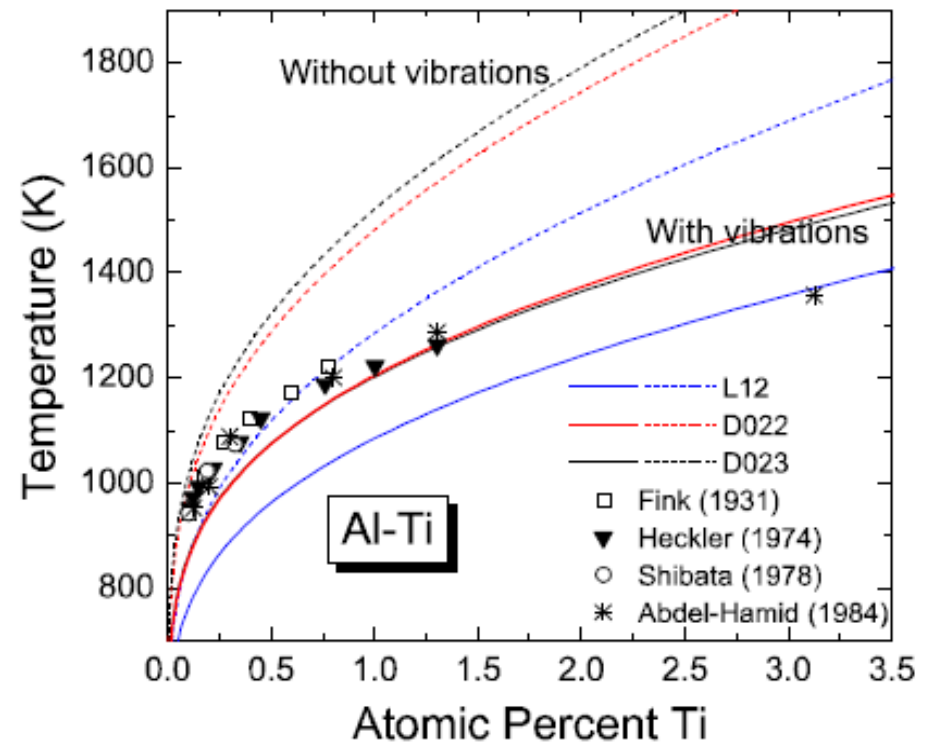
Mixing Energies of BCC Fe-Cu

J. Z. Liu, A. van de Walle, G. Ghosh and MA (2005)



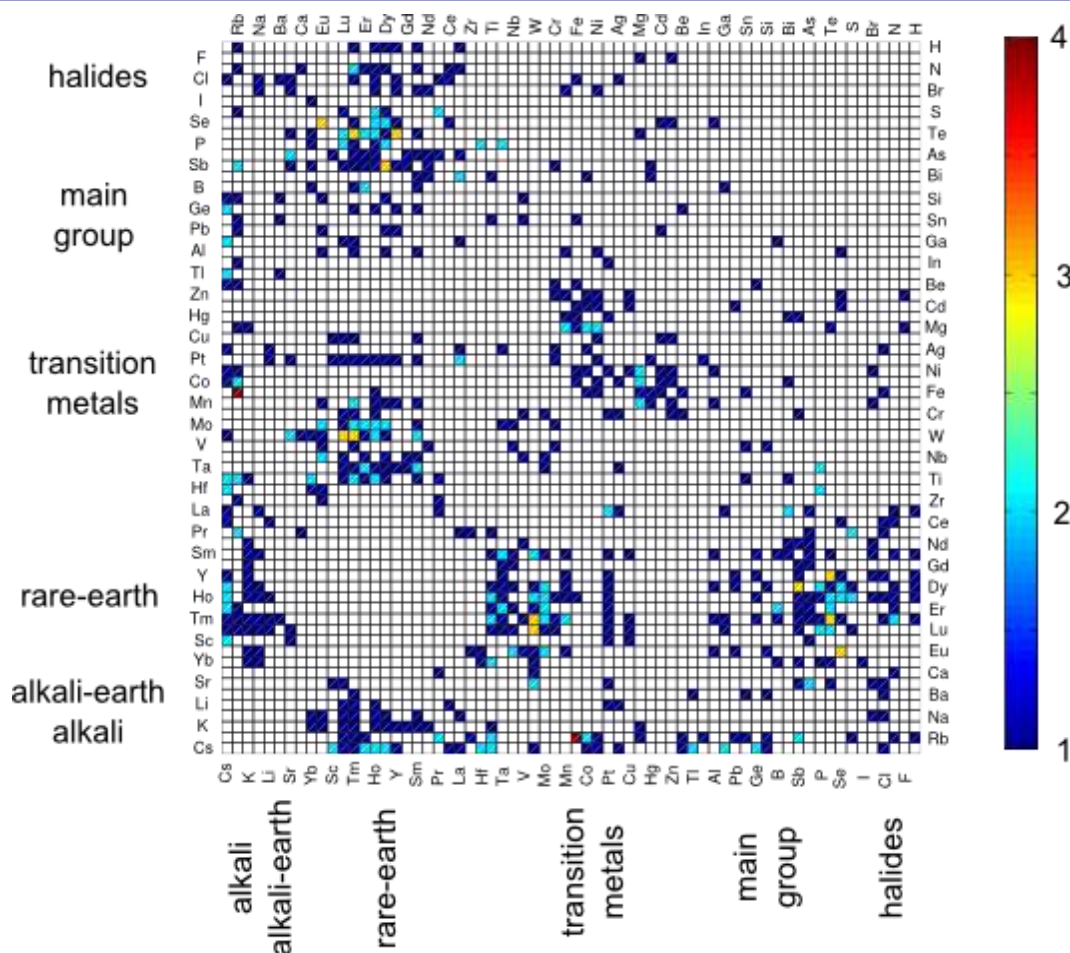
Solvus Boundaries in Al-Ti

J. Z. Liu, G. Ghosh, A. van de Walle and MA (2006)



Predictions for Both Stable and *Metastable* Phases

Discovery of New Materials



G. Hautier, C.C. Fischer, A. Jain, T. Mueller, and G. Ceder, “*Finding Nature’s Missing Ternary Oxide Compounds Using Machine Learning and Density Functional Theory,*” Chem. Mater. 22, 3762-3767 (2010)

Materials Data for Discovery & Design

The screenshot displays the Materials Project website interface. At the top, there are navigation links: Home, About, Apps, Documentation, API, and Dashboard. The main content area is titled "Explore Materials" and features a search bar with the formula "Fe2O3" entered. Below the search bar is a periodic table where elements are color-coded by groups. A sidebar on the left, titled "MATERIAL PROPERTIES", contains several filters: "Material Tags" (with a search box containing "Imaginary" and a "Submit" button), "Band Gap (eV)" (ranging from 0 to 10), "Energy Above Hull" (ranging from 0 to 27), "Formation Energy" (ranging from -27 to 4), "# unit cell sites" (ranging from 1 to 256), "Density" (ranging from 0.2 to 24.8), "Volume" (ranging from 7 to 7887), and "Crystal Systems" (set to "Any"). Below these filters is a checkbox for "Has bandstructure".

The search results for "Fe2O3" are displayed in a central panel. The material is identified as "MATERIAL Al₂O₃" with ID "mp-1143". A 3D ball-and-stick model of the crystal structure is shown, with atoms represented by spheres of different colors (red for oxygen, grey for aluminum). The model is set against a black background with a white grid. The structure type is listed as "Conventional Standard" and "Primitive". The space group is "R $\bar{3}c$ [167]". The material details panel on the right provides the following information:

- Final Magnetic Moment: 0.000 μ_B
- Formation Energy/Atom: -3.442 eV
- Energy Above Hull: 0.000 eV/atom
- Density: 3.87 g/cm³
- Decomposes To: Stable
- Band Gap: 5.854 eV
- Magnetic Ordering: Non-magnetic
- Space Group: Hermann Mauguin R $\bar{3}c$ [167]

The lattice parameters are listed as: a = 5.178 Å, b = 5.178 Å, c = 13.290 Å, and the angles are $\alpha = 90.290^\circ$, $\beta = 90.290^\circ$, $\gamma = 90.290^\circ$. The final structure is shown with fractional coordinates: x = 0.0561, y = 0.75, z = 0.4439.

<https://www.materialsproject.org/>

A. Jain, S.P. Ong, G. Hautier, W. Chen, W.D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner, G. Ceder, K.A. Persson, *Applied Physics Letters Materials*, 2013, 1(1), 011002.

Outline

- Formalism
 - Hydrogen Atom
 - Density Functional Theory
 - Exchange-Correlation Potentials
 - Pseudopotentials and Related Approaches
 - Some Commercial and Open Source Codes
- Practical Issues
 - Implementation
 - Periodic boundary conditions
 - k-Points
 - Plane-wave basis sets
 - Parameters controlling numerical precision
- Example Exercise

Introduction

The Hydrogen Atom

Proton with mass M_1 , coordinate \mathbf{R}_1
Electron with mass m_1 , coordinate \mathbf{r}_1

$$\left(-\frac{\hbar^2}{2M_1} \nabla_1^2 - \frac{\hbar^2}{2m_2} \nabla_2^2 - \frac{e^2}{r} \right) \Psi(R_1, r_2) = E \Psi(R_1, r_2)$$

$$r = r_1 - r_2, \quad R = \frac{M_1 R_1 + m_2 r_2}{M_1 + m_2}, \quad m = \frac{M_1 m_2}{M_1 + m_2}, \quad M = M_1 + m_2$$

$$\left(-\frac{\hbar^2}{2M} \nabla_R^2 - \frac{\hbar^2}{2m} \nabla_r^2 - \frac{e^2}{r} \right) \Psi(R, r) = E \Psi(R, r)$$

$$\Upsilon(R, r) = \Upsilon_{cm}(R) \Upsilon_r(r)$$

$$\left(-\frac{\hbar^2}{2M} \nabla_R^2 \right) \psi_{cm}(R) = E_{cm} \psi_{cm}(R)$$

$$\left(-\frac{\hbar^2}{2m} \nabla_r^2 - \frac{e^2}{r} \right) \psi_r(r) = E_r \psi_r(r)$$

Hydrogen Atom

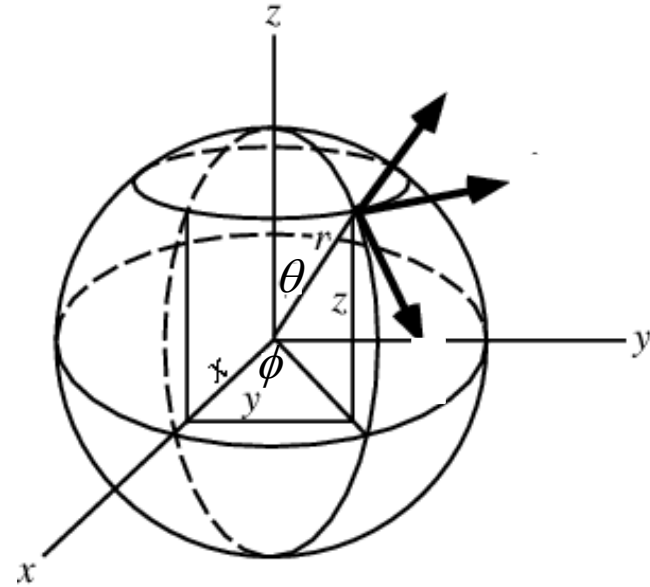
Switch to Spherical Coordinates

$$-\frac{\hbar^2}{2m_e} \frac{1}{r^2} \frac{d}{dr} \left(r^2 \frac{d\psi}{dr} \right) + \frac{1}{r^2 \sin \theta} \frac{d}{d\theta} \left(\sin \theta \frac{d\psi}{d\theta} \right) + \frac{1}{r^2 \sin^2 \theta} \frac{d^2 \psi}{d\phi^2} - \frac{e^2}{r} \psi = E \psi$$

$$\psi(r, \theta, \phi) = R_{nl}(r) Y_l^m(\theta, \phi)$$

$$-\frac{\hbar^2}{2m_e} \frac{1}{r^2} \frac{d}{dr} \left(r^2 \frac{dR}{dr} \right) - \frac{l(l+1)}{r^2} R - \frac{e^2}{r} R = E_n R_{nl}(r)$$

$$E_n = -\frac{me^4}{2\hbar^2} \frac{1}{n^2} = -\frac{13.6}{n^2} \text{ eV}$$

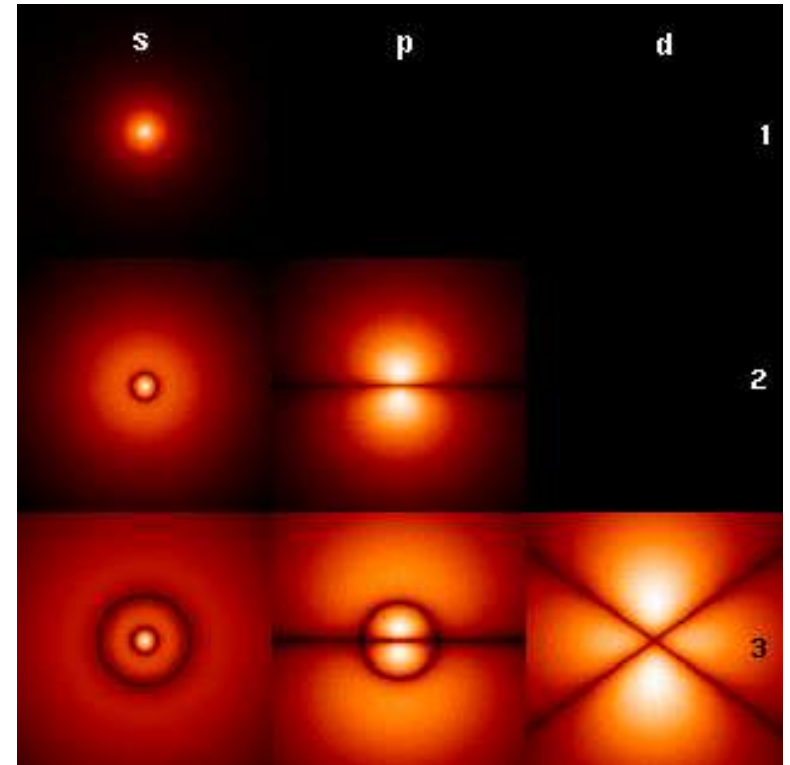
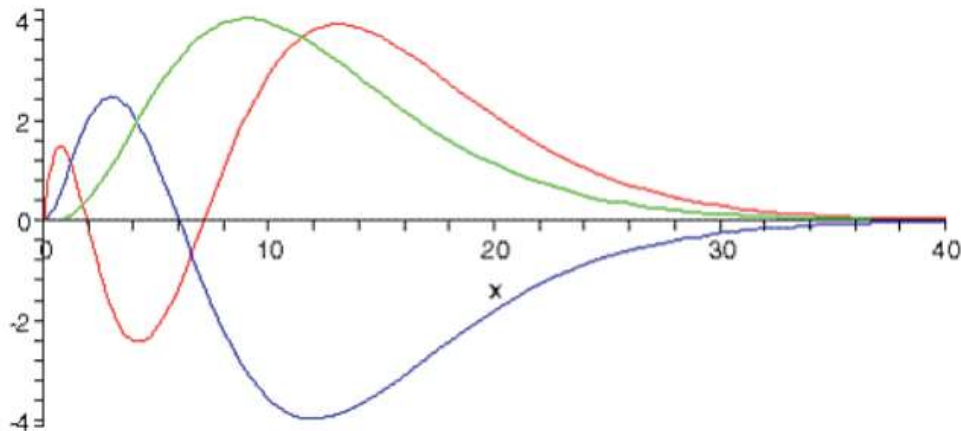


Hydrogen Atom

Wavefunctions

$$n = 1, 2, 3, \dots$$

$$l = 0 \text{ (s)}, 1 \text{ (p)}, 2 \text{ (d)}, \dots, n-1$$



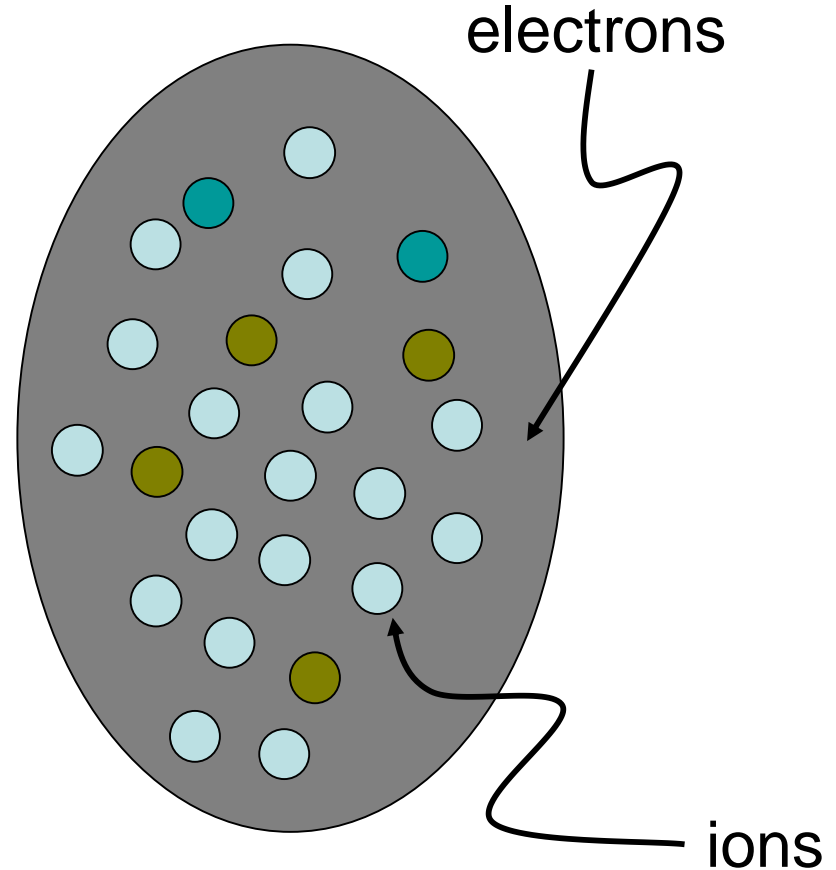
Probability densities through the xz -plane for the electron at different quantum numbers (l , across top; n , down side; $m = 0$)

http://en.wikipedia.org/wiki/Hydrogen_atom

<http://galileo.phys.virginia.edu/classes/751.mf1i.fall02/HydrogenAtom.htm>

The Many-Electron Problem

- collection of
 - N ions
 - n electrons
- total energy computed as a function of ion positions
 - must employ quantum mechanics



Born-Oppenheimer Approximation

- Mass of nuclei exceeds that of the electrons by a factor of 1000 or more
 - we can neglect the kinetic energy of the nuclei
 - treat the ion-ion interaction classically
 - significantly simplifies the Hamiltonian for the electrons
- Consider Hamiltonian for n electrons in potential of N nuclei with atomic numbers Z_i

$$H = -\frac{\hbar^2}{2m} \sum_{i=1}^n \nabla_{\mathbf{r}_i}^2 - \sum_{i=1}^N \sum_{j=1}^n \frac{Z_i e^2}{|\mathbf{R}_i - \mathbf{r}_j|} + \frac{1}{2} \sum_{i=1}^N \sum_{\substack{j=1 \\ i \neq j}}^N \frac{e^2}{|\mathbf{R}_i - \mathbf{R}_j|}$$

external potential $\circ V_{\text{ext}}(\mathbf{r}_j)$

Density Functional Theory

Hohenberg and Kohn (1964), Kohn and Sham (1965)

- For each external potential there is a unique ground-state electron density
- Energy can be obtained by minimizing of a *density functional with respect to density of electrons $n(r)$*

$$E_{\text{groundstate}} = \min\{E_{\text{tot}}[n(\mathbf{r})]\}$$

$$E_{\text{tot}}[n(\mathbf{r})] = T[n(\mathbf{r})] + E_{\text{int}}[n(\mathbf{r})] + \int d\mathbf{r} V_{\text{ext}}(\mathbf{r}) n(\mathbf{r}) + E_{\text{ion-ion}}$$

Kinetic Energy Electron-Electron Interactions Electron-Ion Interactions

Kohn-Sham Approach

$$n(r) = \sum_{i=1}^n e \dot{a} |f_i(r)|^2$$

$$E[\{\phi_i\}] = -\frac{\hbar^2}{2m_e} \sum_{i=1}^n \int \phi_i^* \nabla_i^2 \phi_i d^3 r + \int V_{ext}(r) n(r) d^3 r \\ + \frac{1}{2} \int \frac{n(r)n(r')}{|r-r'|} d^3 r d^3 r' + E_{xc}[n(r)]$$

Many-Body Electron-Electron Interactions Lumped into $E_{xc}[n(r)]$

“Exchange-Correlation Energy”

Kohn-Sham Equations

$$\left[-\frac{\hbar^2}{2m_e} \nabla_i^2 + V_{nuclei}(r) + \int \frac{n(r')}{|r-r'|} d^3 r' + V_{xc}(r) \right] \phi_i(r) = \varepsilon_i \phi_i(r)$$

$$V_{xc}(r) \circ \frac{dE_{xc}[n(r)]}{dn(r)}$$

Local Density Approximation

(e.g., *J. P. Perdew and A. Zunger, Phys. Rev. B* **23**, 5048 (1981))

$$E_{xc}[n(r)] = \int e_{xc}^{\text{hom}}(n(r))n(r)d^3r$$

$\epsilon_{xc}^{\text{hom}}(n(r)) \rightarrow$ Exchange – Correlation Energy of Homogeneous
Electron Gas of Density $n(r)$

Generalized Gradient Approximation

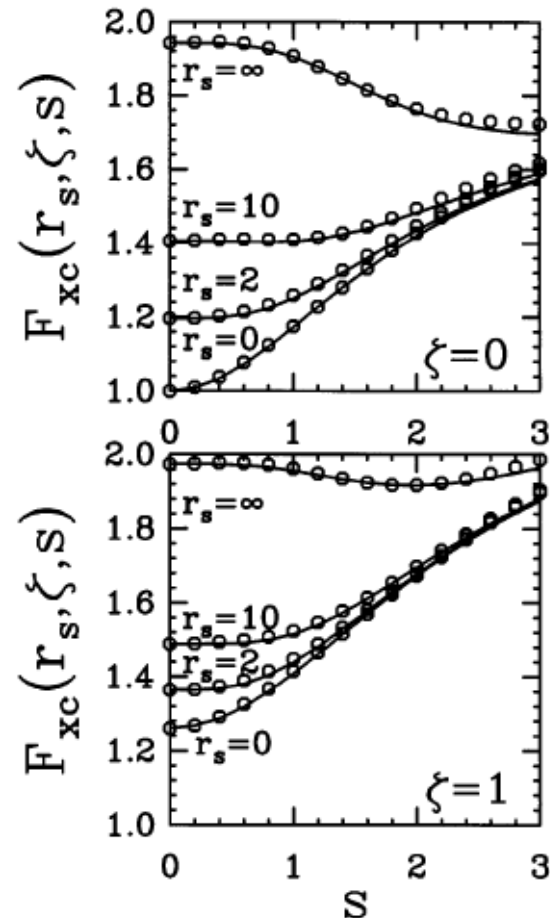
J. P. Perdew, K. Burke and M. Ernzerhof, Phys. Rev. Lett. 77 (1996)

$$E_{xc}^{GGA}[n(r)] = \int e_x^{\text{hom}}(n(r)) n(r) F_{xc}(r_s, Z, s) d^3r$$

$$n = 3/4\rho r_s^3 = k_F^3 / 3\rho^2$$

$$Z = (n_+ - n_-) / n$$

$$s = |\nabla n| / 2k_F n$$

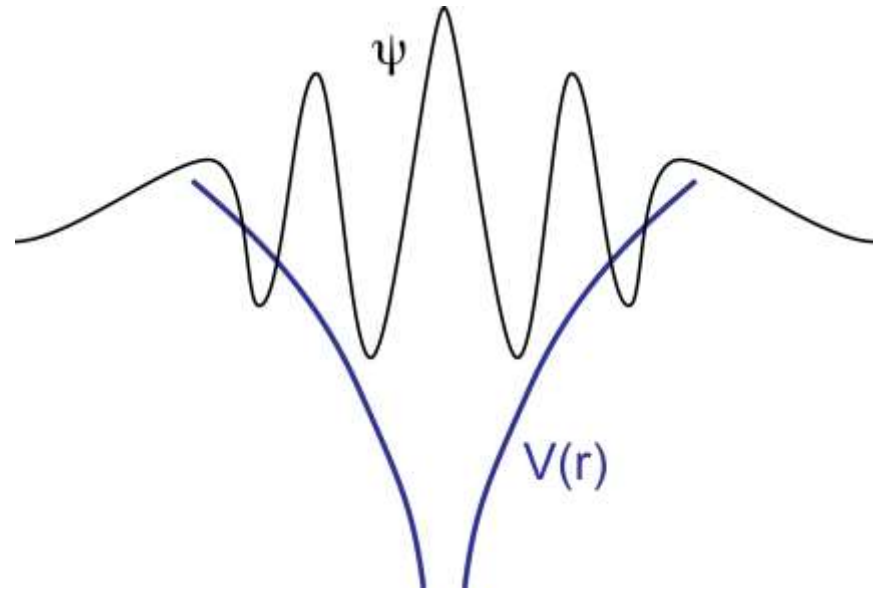


A Note on Accuracy and Ongoing Research

- LDA leads to “overbinding”
 - Lattice constants commonly 1-3 % too small, elastic constants 10-15 % too stiff, cohesive energies 5-20 % too large
- BUT, errors are largely systematic
 - Energy differences tend to be more accurate
- GGA corrects for overbinding
 - Sometimes “overcorrects”
- “Beyond DFT” Approaches
 - For “highly correlated” systems LDA & GGA perform much worse and corrections required (DFT+U, Hybrid Hartree-Fock/DFT, ...)
 - Non-bonded interactions, e.g., van der Waals interactions in graphite, require additional terms or functionals (e.g., vdW-DF)

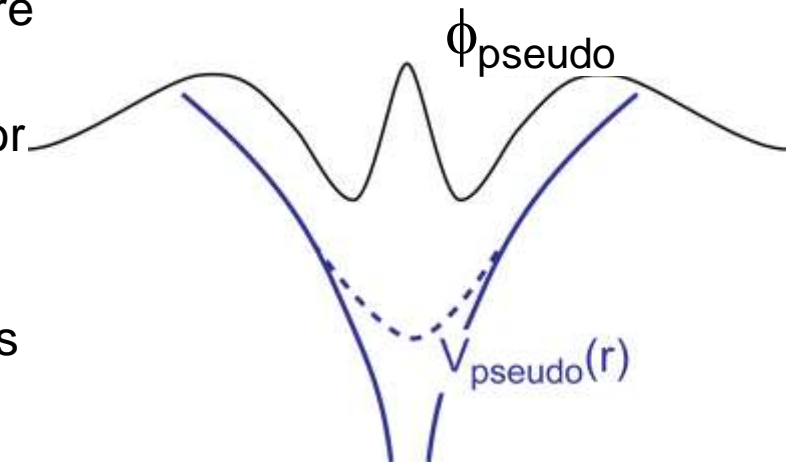
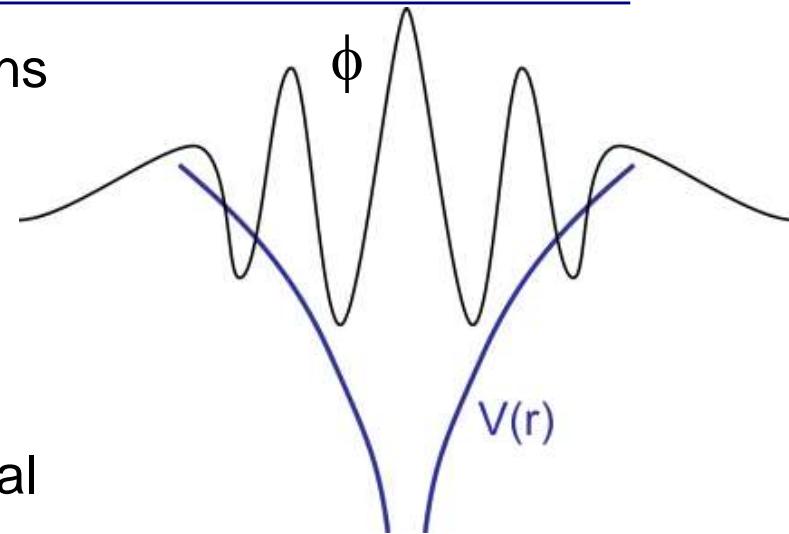
Pseudopotentials

- Potential due to ions is singular at ion core
- Eigenfunctions oscillate rapidly near singularity
- Eigenfunction in bonding region is smooth



Pseudopotentials

- For plane-wave basis sets, rapid oscillations require large number of basis functions
 - expensive
 - unnecessary
 - these oscillations don't alter bonding properties
- Replace potential with nonsingular potential
 - preserve bonding tails of eigenfunction
 - preserve distribution of charge between core and tail regions
 - reduces number of plane waves required for accurate expansion of wavefunction
- Transferable
 - developed from properties of isolated atoms
 - applied in other situations



Summary of Approaches

- Pseudopotentials
 - Core electrons removed from problem and enter only in their effect of the pseudopotential felt by the valence electrons
 - Kohn-Sham equations solved for valence electrons only
- “Augment” Plane Waves with atomic-like orbitals
 - An efficient basis set that allows all electrons to be treated in the calculations
 - Basis for “all-electron” codes
- Projector-Augmented-Wave method
 - Combines features of both methods
 - Generally accepted as the basis for the most accurate approach for calculations requiring consideration of valence electrons only

Some of the Widely Used Codes

- **VASP** (<http://cms.mpi.univie.ac.at/vasp/>)
 - Commercial, Plane-Wave Basis, Pseudopotentials and PAW
- **PWSCF** (<http://www.quantum-espresso.org/>)
 - Free (**and available to run on nanohub**), Plane-Wave Basis, Pseudopotentials and PAW
- **CASTEP** (<http://ccpforge.cse.rl.ac.uk/gf/project/castep/>)
 - Free in UK, licensed by Accelrys elsewhere, Plane-Wave Basis, Pseudopotentials
- **ABINIT** (<http://www.abinit.org/>)
 - Free (**and available to run on nanohub**), plane-wave basis, pseudopotentials and PAW
- **WIEN2K** (<http://www.wien2k.at/>)
 - Commercial (modest license fee), all-electron augmented wave method

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- Example Exercise

Total Energy in Density Functional Theory

$$E[\{\phi_i\}] = -\frac{\hbar^2}{2m_e} \sum_{i=1}^{N_e} \int \phi_i^* \nabla_i^2 \phi_i d^3 r + \int V_{ext}(r) n(r) d^3 r \\ + \frac{1}{2} \int \frac{n(r)n(r')}{|r-r'|} d^3 r d^3 r' + E_{xc}[n(r)]$$

Electron Density $n(r) = -e \sum_{i=1}^n \dot{\mathbf{a}} |f_i(r)|^2$

Electron Wavefunctions $f_i(r)$

Potential Electrons Feel from Nuclei $V_{ext}(r)$

Exchange-Correlation Energy $E_{xc}[n(r)]$

Form depends on whether you use LDA or GGA

Kohn-Sham Equations

Schrödinger Equation for Electron Wavefunctions

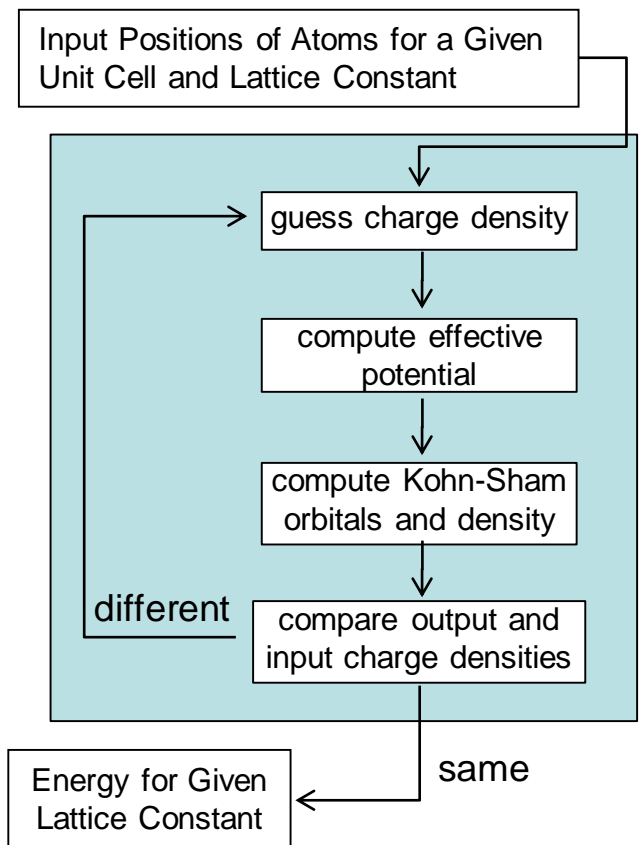
$$\left[-\frac{\hbar^2}{2m_e} \nabla_i^2 + V_{ext}(r) + \int \frac{n(r')}{|r-r'|} d^3r' + V_{xc}(r) \right] f_i(r) = e_i f_i(r)$$

Exchange-Correlation Potential $V_{xc}(r) \circ \frac{dE_{xc}[n(r)]}{dn(r)}$

Electron Density $n(r) = -e \mathring{a} \sum_{i=1}^n |f_i(r)|^2$

Note: ϕ_i depends on $n(r)$ which depends on $\phi_i \rightarrow$
Solution of Kohn-Sham equations must be done iteratively

Self-Consistent Solution to DFT Equations



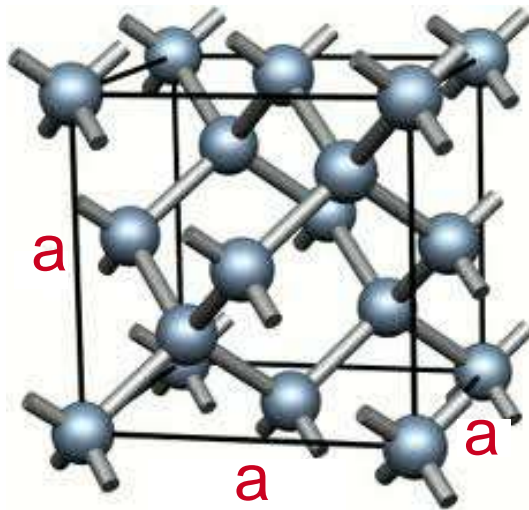
1. Set up atom positions
2. Make initial guess of “input” charge density (often overlapping atomic charge densities)
3. Solve Kohn-Sham equations with this input charge density
4. Compute “output” charge density from resulting wavefunctions
5. If energy from input and output densities differ by amount greater than a chosen threshold, mix output and input density and go to step 2
6. Quit when energy from input and output densities agree to within prescribed tolerance (e.g., 10^{-5} eV)

Note: In your exercise, positions of atoms are dictated by symmetry. If this is not the case another loop must be added to minimize energy with respect to atomic positions.

Implementation of DFT for a Single Crystal

Crystal Structure Defined by Unit Cell Vectors and Positions of Basis Atoms

Example: Diamond Cubic Structure of Si



Unit Cell Vectors

$$\mathbf{a}_1 = a \left(-\frac{1}{2}, \frac{1}{2}, 0\right)$$

$$\mathbf{a}_2 = a \left(-\frac{1}{2}, 0, \frac{1}{2}\right)$$

$$\mathbf{a}_3 = a \left(0, \frac{1}{2}, \frac{1}{2}\right)$$

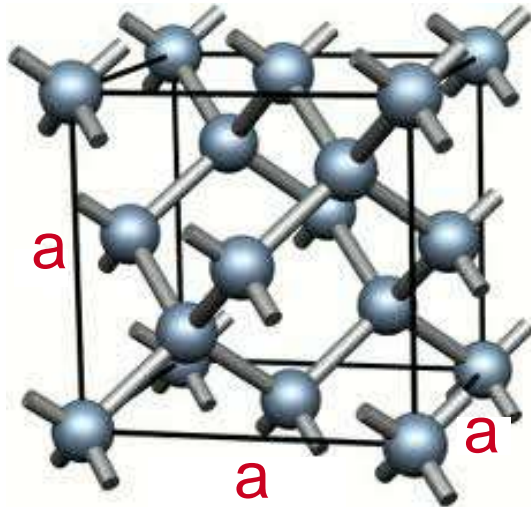
Basis Atom Positions

$$0 \ 0 \ 0$$

$$\frac{1}{4} \ \frac{1}{4} \ \frac{1}{4}$$

All atoms in the crystal can be obtained by adding integer multiples of unit cell vectors to basis atom positions

Electron Density in Crystal Lattice



Unit-Cell Vectors

$$\mathbf{a}_1 = a \left(-\frac{1}{2}, \frac{1}{2}, 0 \right)$$

$$\mathbf{a}_2 = a \left(-\frac{1}{2}, 0, \frac{1}{2} \right)$$

$$\mathbf{a}_3 = a \left(0, \frac{1}{2}, \frac{1}{2} \right)$$

Electron density is periodic with periodicity given by \mathbf{R}_{uvw}

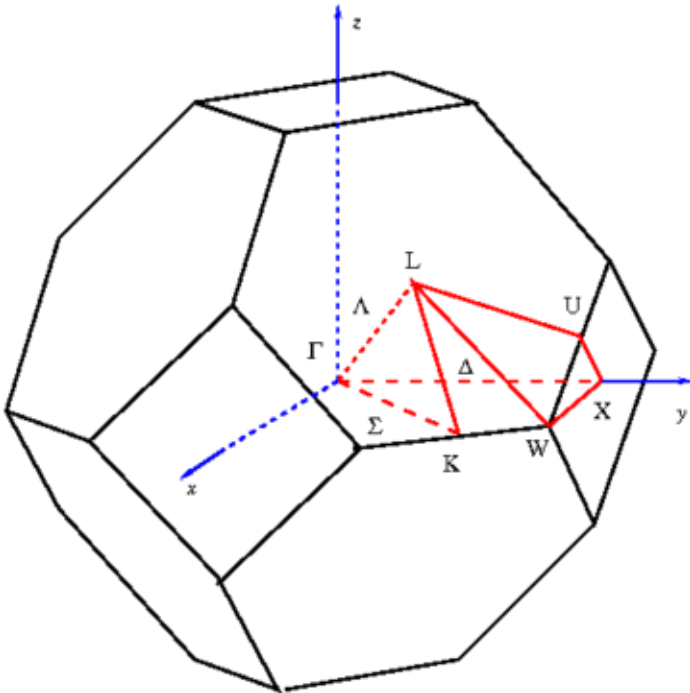
$$n(\mathbf{r}) = n(\mathbf{r} + \mathbf{R}_{uvw})$$

Translation Vectors: $\mathbf{R}_{uvw} = u\mathbf{a}_1 + v\mathbf{a}_2 + w\mathbf{a}_3$

Electronic Bandstructure

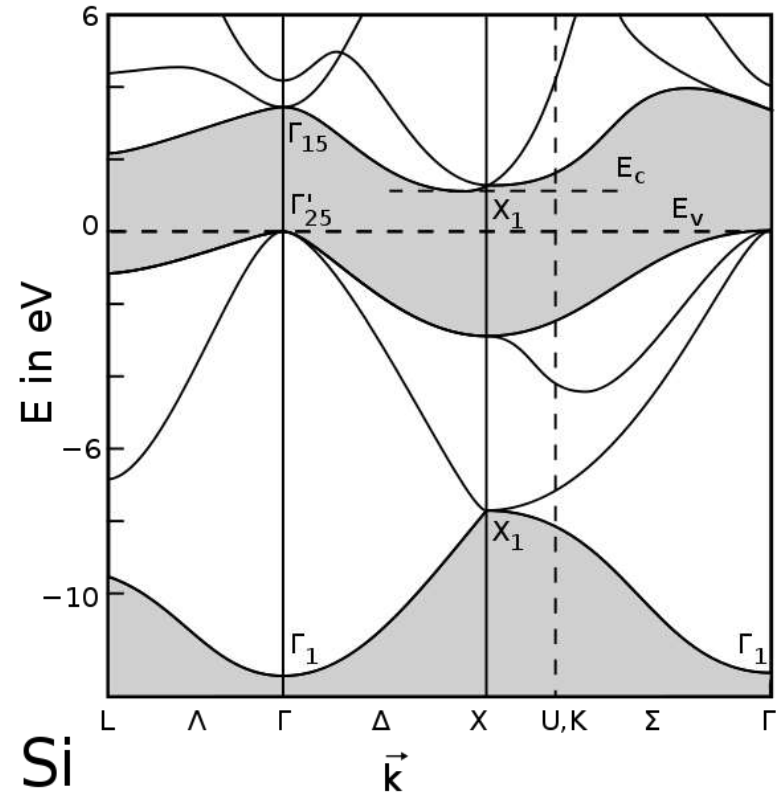
Example for Si

Brillouin Zone



http://en.wikipedia.org/wiki/Brillouin_zone

Bandstructure



http://de.wikipedia.org/wiki/Datei:Band_structure_Si_schematic.svg

Electronic wavefunctions in a crystal can be indexed by point in reciprocal space (\mathbf{k}) and a band index (β)

Why?

Wavefunctions in a Crystal Obey Bloch's Theorem

For a given band β

$$f_{\mathbf{k}}^b(\mathbf{r}) = \exp(i\mathbf{k} \times \mathbf{r}) u_{\mathbf{k}}^b(\mathbf{r})$$

Where $u_{\mathbf{k}}^b(\mathbf{r})$ is periodic in real space: $u_{\mathbf{k}}^b(\mathbf{r}) = u_{\mathbf{k}}^b(\mathbf{r} + \mathbf{R}_{uvw})$

Translation Vectors: $\mathbf{R}_{uvw} = u\mathbf{a}_1 + v\mathbf{a}_2 + w\mathbf{a}_3$

Representation of Electron Density

$$f_{\mathbf{k}}^b(\mathbf{r}) = \exp(i\mathbf{k} \times \mathbf{r}) u_{\mathbf{k}}^b(\mathbf{r})$$

$$n(\mathbf{r}) = -e \overset{N_e}{\underset{i=1}{\dot{\mathbf{a}}}} |f_i(\mathbf{r})|^2 \longrightarrow n(\mathbf{r}) = -e \sum_b \int_{W_{BZ}} |f_{\mathbf{k}}^b(\mathbf{r})|^2 f(e_{\mathbf{k}}^b - e_F) \frac{d^3k}{W_{BZ}}$$

Integral over k-points in first Brillouin zone

$f(\varepsilon - \varepsilon_F)$ is Fermi-Dirac distribution function with Fermi energy ε_F

In practice the integral over the Brillouin zone is replaced with a sum over a finite number of k-points (N_{kpt})

$$n(\mathbf{r}) \gg -e \overset{N_{kpt}}{\underset{b}{\dot{\mathbf{a}}}} \underset{j=1}{\dot{\mathbf{a}}} w_j |f_{\mathbf{k}_j}^b(\mathbf{r})|^2 f(e_{\mathbf{k}_j}^b - e_F)$$

One parameter that needs to be checked for numerical convergence is number of k-points

Representation of Wavefunctions

Fourier-Expansion as Series of Plane Waves

For a given band: $f_{\mathbf{k}}^b(\mathbf{r}) = \exp(i\mathbf{k} \times \mathbf{r}) u_{\mathbf{k}}^b(\mathbf{r})$

Recall that $u_{\mathbf{k}}^b(\mathbf{r})$ is periodic in real space: $u_{\mathbf{k}}^b(\mathbf{r}) = u_{\mathbf{k}}^b(\mathbf{r} + \mathbf{R}_{uvw})$

$u_{\mathbf{k}}^b(\mathbf{r})$ can be written as a Fourier Series:

$$u_{\mathbf{k}}^b(\mathbf{r}) = \sum_{lmn} \hat{a}_{lmn} u_{\mathbf{k}}^b(\mathbf{G}_{lmn}) \exp(i\mathbf{G}_{lmn} \times \mathbf{r})$$

$$\mathbf{G}_{lmn} = l\mathbf{a}_1^* + m\mathbf{a}_2^* + n\mathbf{a}_3^*$$

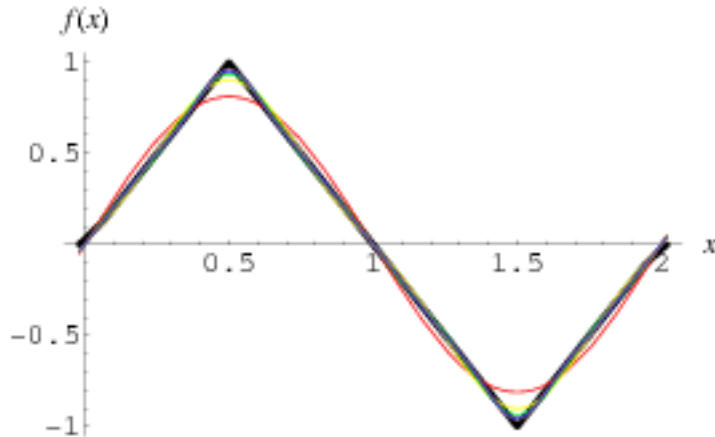
where the \mathbf{a}_i^* are primitive *reciprocal lattice vectors*

$$\mathbf{a}_1^* \times \mathbf{a}_1 = 2\rho \quad \mathbf{a}_1^* \times \mathbf{a}_2 = 0 \quad \mathbf{a}_1^* \times \mathbf{a}_3 = 0$$

$$\mathbf{a}_2^* \times \mathbf{a}_1 = 0 \quad \mathbf{a}_2^* \times \mathbf{a}_2 = 2\rho \quad \mathbf{a}_2^* \times \mathbf{a}_3 = 0$$

$$\mathbf{a}_3^* \times \mathbf{a}_1 = 0 \quad \mathbf{a}_3^* \times \mathbf{a}_2 = 0 \quad \mathbf{a}_3^* \times \mathbf{a}_3 = 2\rho$$

Recall Properties of Fourier Series



Black line = (exact) triangular wave

Colored lines = Fourier series truncated at different orders

<http://mathworld.wolfram.com/FourierSeriesTriangleWave.html>

General Form of Fourier Series:
$$f(x) = \frac{1}{2} a_0 + \sum_{n=1}^{\infty} a_n \cos(nx) + \sum_{n=1}^{\infty} b_n \sin(nx).$$

For Triangular Wave:
$$f(x) = \frac{8}{\pi^2} \sum_{n=1,3,5,\dots}^{\infty} \frac{(-1)^{(n-1)/2}}{n^2} \sin\left(\frac{n\pi x}{L}\right).$$

Typically we expect the accuracy of a truncated Fourier series to improve as we increase the number of terms

Representation of Wavefunctions

Plane-Wave Basis Set

For a given band $f_{\mathbf{k}}^b(\mathbf{r}) = \exp(i\mathbf{k} \times \mathbf{r}) u_{\mathbf{k}}^b(\mathbf{r})$

Use Fourier Expansion

$$f_{\mathbf{k}}^b(\mathbf{r}) = \sum_{\mathbf{G}} \hat{a}_{\mathbf{G}} u_{\mathbf{k}}^b(\mathbf{G}) \exp(i(\mathbf{G} + \mathbf{k}) \times \mathbf{r})$$

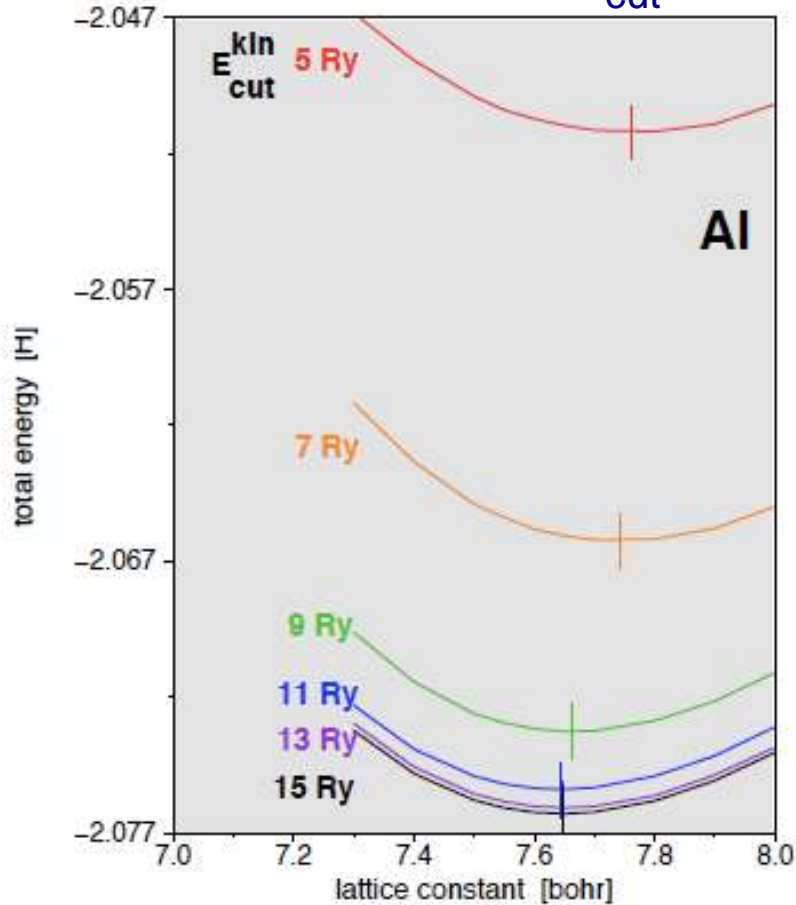
In practice the Fourier series is truncated to include all \mathbf{G} for which:

$$\frac{\hbar^2}{2m} (\mathbf{G} + \mathbf{k})^2 < E_{\text{cut}}$$

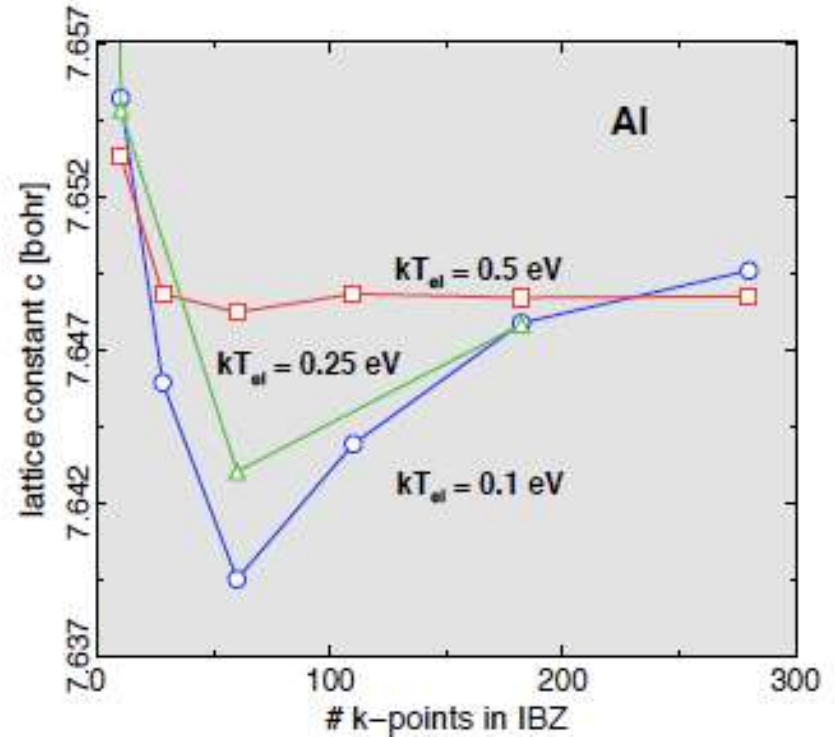
Another parameter that needs to be checked for convergence is the “plane-wave cutoff energy” E_{cut}

Examples of Convergence Checks

Effect of E_{cut}



Effect of Number of k Points

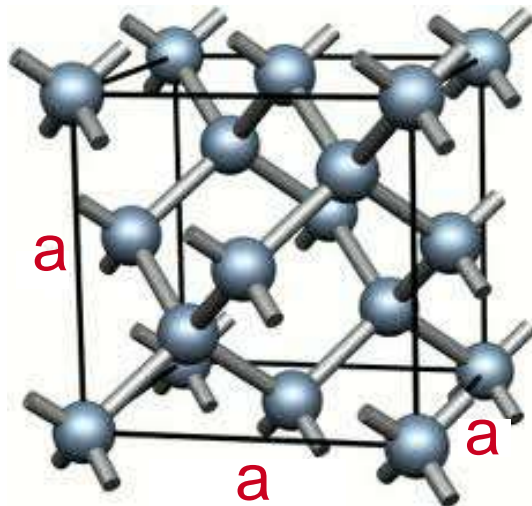


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Your Exercise: Part 1

- Calculate equation of state of diamond cubic Si using Quantum Espresso on Nanohub (<http://nanohub.org/>)
- You will compare accuracy of LDA and GGA
- You will check numerical convergence with respect to number of k-points and plane-wave cutoff
- You will make use of the following unit cell for diamond-cubic structure



Lattice Vectors

$$a_1 = a \left(-\frac{1}{2}, \frac{1}{2}, 0\right)$$

$$a_2 = a \left(-\frac{1}{2}, 0, \frac{1}{2}\right)$$

$$a_3 = a \left(0, \frac{1}{2}, \frac{1}{2}\right)$$

Basis Atom Positions

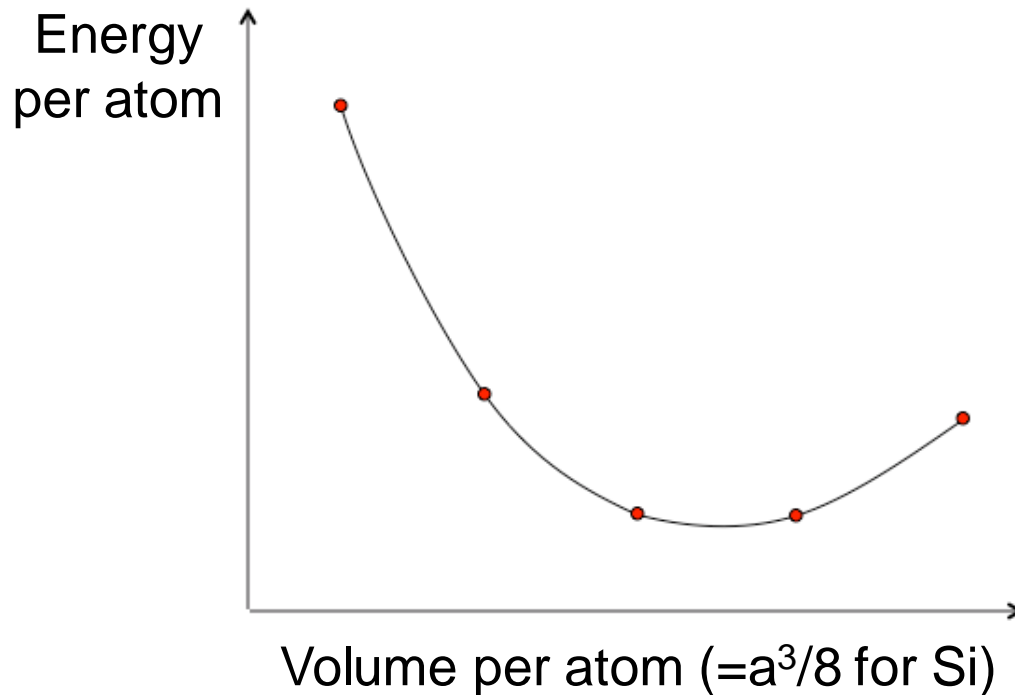
$$0 \ 0 \ 0$$

$$\frac{1}{4} \ \frac{1}{4} \ \frac{1}{4}$$

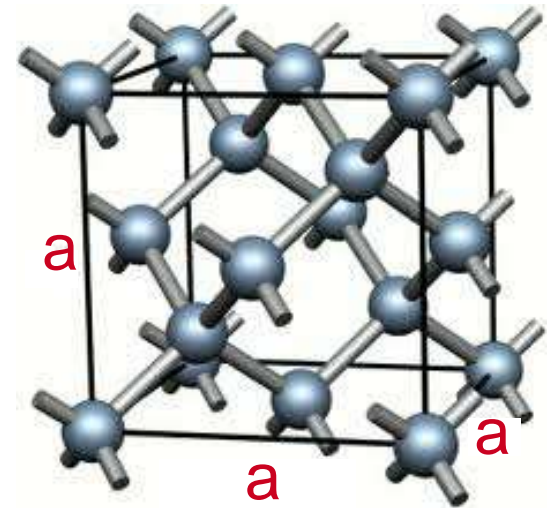
Equation of State

A Probe of Interatomic Interactions

Schematic Energy vs. Volume Relation



Diamond Cubic Structure of Si



<http://www.e6cvd.com/cvd/page.jsp?pageid=361>

Equation of State

What Properties Can we Learn from It?

Pressure versus Volume Relation

$$P = - \frac{\partial E}{\partial V}$$

Given $E(V)$ one can compute $P(V)$ by taking derivative

Recall 1st Law of Thermo: $dE = T dS - P dV$ and consider $T = 0 K$

Equilibrium Volume (or Lattice Constant)

Volume corresponding to zero pressure = Volume where slope of $E(V)$ is zero
 \approx Volume measured experimentally at $P = 1 \text{ atm}$

Bulk Modulus

$$B = -V \frac{\partial P}{\partial V} = V \frac{\partial^2 E}{\partial V^2}$$

B related to curvature of $E(V)$ Function

Your Exercise: Part 2

Non-hydrostatic Stress and Strain

Stress-Strain Relations in Linear Elasticity

$$s_{ij} = \underset{k,l}{\mathring{a}c}_{ijkl} e_{kl}$$

Stress Strain

C_{ijkl} \longrightarrow *Single-Crystal Elastic Constants*

Stress-Strain Relations in Linear Elasticity

Consider Single Strain $\varepsilon_{33}=\varepsilon$

$$\sigma_{\square\square\square} = C_{11}\varepsilon$$

$$\sigma_{22\square} = C_{12}\varepsilon$$

Voigt Notation (for Cubic Crystal)

$$C_{3333}=C_{2222}=C_{1111}=C_{11}$$

$$C_{2233}=C_{1133}=C_{1122}=C_{2211}=C_{3311}=C_{3322}=C_{12}$$

Lecture 8: Introduction to Density Functional Theory

Marie Curie Tutorial Series: Modeling Biomolecules
December 6-11, 2004

Mark Tuckerman
Dept. of Chemistry
and Courant Institute of Mathematical Science
100 Washington Square East
New York University, New York, NY 10003

Background

- **1920s**: Introduction of the Thomas-Fermi model.
- **1964**: Hohenberg-Kohn paper proving existence of exact DF.
- **1965**: Kohn-Sham scheme introduced.
- **1970s and early 80s**: LDA. DFT becomes useful.
- **1985**: Incorporation of DFT into molecular dynamics (Car-Parrinello)
(Now one of PRL's top 10 cited papers).
- **1988**: Becke and LYP functionals. DFT useful for some chemistry.
- **1998**: Nobel prize awarded to Walter Kohn in chemistry for development of DFT.

Motivation

2. Electrons are quantum mechanical spin-1/2 particles and must be described by an antisymmetric wavefunction:

$$\Psi(\mathbf{r}_1, s_1, \dots, \mathbf{r}_N, s_N)$$

3. DFT allows all properties to be determined (in principle) by the electron density, $n(\mathbf{r})$, a function of just three variables, $\mathbf{r} = (x, y, z)$.
4. DFT offers an elegant formulation of N -particle quantum mechanics, conceptual simplicity, and computational efficiency.
5. Limitations of accuracy from approximation to unknown energy terms.

The many electron Hamiltonian

Consider a molecule with N nuclei at $\mathbf{R}_1, \dots, \mathbf{R}_N \equiv \mathbf{R}$ and N_e electrons ($\mathbf{r}_1, \dots, \mathbf{r}_{N_e}$) in the Born-Oppenheimer approximation:

$$H_e = T_e + V_{ee} + V_{eN}$$

Kinetic energy:

$$T_e = -\frac{1}{2} \sum_{i=1}^{N_e} \nabla_i^2$$

Electron-electron repulsion:

$$V_{ee} = \sum_{i>j} \frac{1}{|\mathbf{r}_i - \mathbf{r}_j|}$$

External Potential:

$$V_{eN} = - \sum_{i=1}^{N_e} \sum_{I=1}^N \frac{Z_I}{|\mathbf{r}_i - \mathbf{R}_I|} = \sum_{i=1}^{N_e} V_{\text{ext}}(\mathbf{r}_i, \mathbf{R})$$

Total Molecular Hamiltonian:

$$H = H_e + T_N + V_{NN}$$

$$T_N = -\sum_{I=1}^N \frac{1}{2M_I} \nabla_I^2$$

$$V_{NN} = \sum_{I,J>I}^N \frac{Z_I Z_J}{|\mathbf{R}_I - \mathbf{R}_J|}$$

Born-Oppenheimer Approximation:

$$\checkmark [T_e + V_{ee} + V_{eN}] \Psi(\mathbf{x}_1, \dots, \mathbf{x}_{N_e}; \mathbf{R}) = E_0(\mathbf{R}) \Psi(\mathbf{x}_1, \dots, \mathbf{x}_{N_e}; \mathbf{R})$$

$$[T_N + V_{NN} + E_0] \chi(\mathbf{R}, t) = i \frac{\partial}{\partial t} \chi(\mathbf{R}, t)$$

$$\mathbf{x}_i = \mathbf{r}_i, s_i$$

Hohenberg-Kohn Theorem

- Two systems with the same number N_e of electrons have the same $T_e + V_{ee}$. Hence, they are distinguished only by V_{en} .
- Knowledge of $|\Psi_0\rangle$ determines V_{en} .
- Let ζ be the set of external potentials such solution of

$$H_e \Psi = [T_e + V_{ee} + V_{eN}] \Psi = E_0 \Psi$$

yields a non-degenerate ground state $|\Psi_0\rangle$.

Collect all such ground state wavefunctions into a set Ψ . Each element of this set is associated with a Hamiltonian determined by the external potential.

There exists a 1:1 mapping C such that

$$C : \zeta \longrightarrow \Psi$$

Proof that C exists

Let V_{eN} and V'_{eN} be two elements of \mathcal{V} , such that

$$V_{eN} \neq V'_{eN} + \text{const}$$

Let $|\Psi_0\rangle$ and $|\Psi'_0\rangle$ be the associated ground state wavefunctions:

$$(T_e + V_{ee} + V_{eN})|\Psi_0\rangle = E_0|\Psi_0\rangle \quad (1)$$

$$(T_e + V_{ee} + V'_{eN})|\Psi'_0\rangle = E'_0|\Psi'_0\rangle \quad (2)$$

We will show that $|\Psi_0\rangle = |\Psi'_0\rangle$ leads to a contradiction.

Subtract (2) from (1):

$$(V_{eN} - V'_{eN})|\Psi_0\rangle = (E_0 - E'_0)|\Psi_0\rangle$$

Since V_{eN} and V'_{eN} are multiplicative operators:

$$V_{eN} - V'_{eN} = E_0 - E'_0$$

thus *contradicting* the original assumption.

Inverse Map:

$$C^{-1} : \Psi \longrightarrow \mathcal{V}$$

$$|\Psi_0\rangle \neq |\Psi'_0\rangle \Rightarrow V_{eN} \neq V'_{eN}$$

Assume $V_{eN} = V'_{eN}$, then:

$$(T_e + V_{ee} + V_{eN})|\Psi_0\rangle = E_0|\Psi_0\rangle$$

$$(T_e + V_{ee} + V_{eN})|\Psi'_0\rangle = E'_0|\Psi'_0\rangle$$

$|\Psi'_0\rangle$ cannot be a ground state of H_e . But it must be a ground state of $H'_e = H_e$, which is a contradiction.

Hohenberg-Kohn Theorem (part II)

Given an antisymmetric ground state wavefunction from the set Ψ , the ground-state density is given by

$$n(\mathbf{r}) = N_e \sum_{s_1} \cdots \sum_{s_{N_e}} \int d\mathbf{r}_2 \cdots d\mathbf{r}_{N_e} \left| \Psi(\mathbf{r}, s_1, \mathbf{r}_2, s_2, \dots, \mathbf{r}_{N_e}, s_{N_e}) \right|^2$$

Knowledge of $n(\mathbf{r})$ is sufficient to determine $|\Psi\rangle$

Let \mathbf{N} be the set of ground state densities obtained from N_e -electron ground state wavefunctions in Ψ . Then, there exists a 1:1 mapping

$$\mathbf{D} : \Psi \longrightarrow \mathbf{N} \qquad \mathbf{D}^{-1} : \mathbf{N} \longrightarrow \Psi$$

The formula for $n(\mathbf{r})$ shows that \mathbf{D} exists, however, showing that \mathbf{D}^{-1} exists is less trivial.

Proof that D^{-1} exists

Ground state expectation values:

$$E_0 = \langle \Psi_0 | H_e | \Psi_0 \rangle = \langle \Psi_0 | T_e + V_{ee} + V_{eN} | \Psi_0 \rangle$$

$$E'_0 = \langle \Psi'_0 | H'_e | \Psi'_0 \rangle = \langle \Psi'_0 | T_e + V_{ee} + V'_{eN} | \Psi'_0 \rangle$$

By the Ritz principle:

$$E_0 < \langle \Psi'_0 | H_e | \Psi'_0 \rangle$$

However

$$\begin{aligned} \langle \Psi'_0 | H_e | \Psi'_0 \rangle &= \langle \Psi'_0 | H'_e + V_{eN} - V'_{eN} | \Psi'_0 \rangle \\ &= E'_0 + \langle \Psi'_0 | V_{eN} - V'_{eN} | \Psi'_0 \rangle \end{aligned}$$

Note:

$$\begin{aligned}\langle \Psi_0 | V_{eN} | \Psi_0 \rangle &= \int dx_1 \cdots dx_{N_e} \sum_{i=1}^{N_e} V_{\text{ext}}(\mathbf{r}_i) |\Psi_0(\mathbf{x}_1, \dots, \mathbf{x}_{N_e})|^2 \\ &= \sum_{i=1}^{N_e} \int dx_1 \cdots dx_{N_e} V_{\text{ext}}(\mathbf{r}_i) |\Psi_0(\mathbf{x}_1, \dots, \mathbf{x}_{N_e})|^2 \\ &= N_e \int d\mathbf{r} V_{\text{ext}}(\mathbf{r}) \sum_s \int dx_2 \cdots dx_{N_e} |\Psi_0(\mathbf{r}, s, \dots, \mathbf{x}_{N_e})|^2 \\ &= \int d\mathbf{r} V_{\text{ext}}(\mathbf{r}) n_0(\mathbf{r})\end{aligned}$$

Starting instead with

$$E'_0 < \langle \Psi_0 | H'_e | \Psi_0 \rangle$$

one can show that

$$E'_0 < E_0 - \int d\mathbf{r} n_0(\mathbf{r}) [V_{\text{ext}}(\mathbf{r}) - V'_{\text{ext}}(\mathbf{r})]$$

Composite Map

The maps C^{-1} and D^{-1} exist, hence

$$(CD)^{-1} : \mathcal{N} \longrightarrow \zeta$$

is a 1:1 map that maps ground state densities uniquely to external potentials.

- Knowledge of $n_0(\mathbf{r})$ determines V_{eN} and, thus, H_e .
- $n_0(\mathbf{r})$ determines all of the properties of a system.
- Given an operator \hat{O} , the expectation value:

$$\langle \Psi_0[n_0] | \hat{O} | \Psi_0[n_0] \rangle = O[n_0]$$

is a *unique functional* of $n_0(\mathbf{r})$

The theorems are generalizable to degenerate ground states!

The energy functional

The energy expectation value is of particular importance

$$\langle \Psi_0[n_0] | H_e | \Psi_0[n_0] \rangle = E[n_0]$$

From the variational principle, for $|\Psi\rangle$ in Ψ :

$$\langle \Psi | H_e | \Psi \rangle \geq \langle \Psi_0 | H_e | \Psi_0 \rangle$$

Thus,

$$\langle \Psi[n] | H_e | \Psi[n] \rangle = E[n] \geq E[n_0]$$

Therefore, $E[n_0]$ can be determined by a minimization procedure:

$$E[n_0] = \min_{n(\mathbf{r}) \in \mathcal{N}} E[n]$$

Hohenberg-Kohn Functional

Two systems with N_e electrons are distinguished *only* by the external potential. Therefore,

$$\begin{aligned} E[n] &= \langle \Psi | T_e + V_{ee} | \Psi \rangle + \int d\mathbf{r} n(\mathbf{r}) V_{\text{ext}}(\mathbf{r}, \mathbf{R}) \\ &= T_e[n] + V_{ee}[n] + \int d\mathbf{r} n(\mathbf{r}) V_{\text{ext}}(\mathbf{r}, \mathbf{R}) \\ &= F[n] + \int d\mathbf{r} n(\mathbf{r}) V_{\text{ext}}(\mathbf{r}, \mathbf{R}) \end{aligned}$$

The functional $F[n]$ is the same for *all* N_e -electron systems.

$F[n]$ is, therefore, a *universal* functional known as the **Hohenberg-Kohn** functional.

Hohenberg-Kohn Minimization

Minimize $E[n]$ subject to the condition:

$$\int d\mathbf{r} n(\mathbf{r}) = N_e$$

Minimization Procedure:

$$\frac{\delta}{\delta n(\mathbf{r})} \left[E[n] - \mu \left(\int d\mathbf{r} n(\mathbf{r}) - N_e \right) \right] = 0$$

μ is a *Lagrange Multiplier* related to the chemical potential.

Hohenberg-Kohn equation:

$$\frac{\delta F}{\delta n(\mathbf{r})} + V_{\text{ext}}(\mathbf{r}, \mathbf{R}) = \mu$$

where use has been made of the identity:

$$\frac{\delta}{\delta n(\mathbf{r})} \int d\mathbf{r}' f(n(\mathbf{r}')) = \int d\mathbf{r}' \frac{\partial f}{\partial n(\mathbf{r}')} \delta(\mathbf{r}-\mathbf{r}') = \frac{\partial f}{\partial n(\mathbf{r})}$$

v-representability

A density $n(\mathbf{r})$ is *v-representable* if it is associated with an antisymmetric ground state wavefunction of *some* Hamiltonian H_e distinguished by a *particular* external potential $V_{\text{ext}}(\mathbf{r}, \mathbf{R})$.

- Not all densities are *v-representable*.
- There is no general way to tell if a density is *v-representable*.
- Since Hohenberg-Kohn theorem maps *v-representable* densities to ground state wavefunctions, the minimization must be carried out over *v-representable* densities.

N -representability

A density $n(\mathbf{r})$ is N -representable if it is associated with an antisymmetric wavefunction, not necessarily a ground state wavefunction associated with a Hamiltonian H_e .

Requires:

$$n(\mathbf{r}) \geq 0 \quad \int d\mathbf{r} n(\mathbf{r}) = N_e$$

- Given $|\Psi\rangle$, easily determine $n(\mathbf{r})$.
- Given $n(\mathbf{r})$, cannot easily determine $|\Psi\rangle$.
- Particularly, given $n_0(\mathbf{r})$, how can we tell if it came from $|\Psi_0\rangle$ or some other $|\Psi_{n_0}\rangle$.

Levy constrained search

Hohenberg-Kohn functional satisfies

$$F[n_0] = \min_{|\psi_{n_0}\rangle \rightarrow n_0} \langle \Psi_{n_0} | T_e + V_{ee} | \psi_{n_0} \rangle$$

Define:

$$\tilde{F}[n] = \min_{|\Psi\rangle \rightarrow n(\mathbf{r})} \langle \Psi | T_e + V_{ee} | \Psi \rangle$$

Since $\tilde{F}[n_0] = F[n_0]$, we can reformulate Hohenberg-Kohn minimization as:

$$\begin{aligned} E_0 &= \min_{n(\mathbf{r}) \in \mathcal{N}} \left[\min_{|\Psi\rangle \rightarrow n(\mathbf{r})} \langle \Psi | T_e + V_{ee} | \Psi \rangle + \int d\mathbf{r} n(\mathbf{r}) V_{\text{ext}}(\mathbf{r}) \right] \\ &= \min_{n(\mathbf{r})} \left[F[n] + \int d\mathbf{r} n(\mathbf{r}) V_{\text{ext}}(\mathbf{r}) \right] \end{aligned}$$

which only requires N -representable densities.

Thomas-Fermi Theory

Ideal Fermi Gas of constant density n :

$$\psi_{\mathbf{n}}(\mathbf{r}) = \frac{1}{L^{3/2}} e^{2\pi i \mathbf{n} \cdot \mathbf{r}}$$

$$\epsilon_{\mathbf{n}} = \frac{2\pi^2}{L^2} \mathbf{n}^2$$

$$f_{\mathbf{n}} = \begin{cases} 1 & \epsilon_{\mathbf{n}} < \epsilon_F \\ 0 & \epsilon_{\mathbf{n}} > \epsilon_F \end{cases}$$

Kinetic energy per particle:

$$t = \sum_{\mathbf{n}} f_{\mathbf{n}} \epsilon_{\mathbf{n}} = C_F n^{5/3}$$

Local density approximation: $n \rightarrow n(\mathbf{r})$

$$T_e[n] = C_F \int d\mathbf{r} n^{5/3}(\mathbf{r})$$

Exchange energy:

- **One-particle density matrix:**

$$\rho(\mathbf{r}, \mathbf{r}') = N_e \sum_{\{s\}} \int d\mathbf{r}_2 \cdots d\mathbf{r}_{N_e} \Psi^*(\mathbf{r}, s_1, \mathbf{r}_2, s_2, \dots, \mathbf{r}_{N_e}, s_{N_e}) \Psi(\mathbf{r}', s_1, \mathbf{r}_2, s_2, \dots, \mathbf{r}_{N_e}, s_{N_e})$$

- **Exchange energy density:**

$$k = -\frac{1}{4V} \int d\mathbf{r} d\mathbf{r}' \frac{|\rho(\mathbf{r}, \mathbf{r}')|^2}{|\mathbf{r} - \mathbf{r}'|} = -C_x n^{4/3}$$

- **Local density approximation:**

$$E_x[n] = -C_x \int d\mathbf{r} n^{4/3}(\mathbf{r})$$

Approximate energy functional:

$$\begin{aligned} E[n] &= C_F \int d\mathbf{r} n^{5/3}(\mathbf{r}) + \frac{1}{2} \int d\mathbf{r} d\mathbf{r}' \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} \\ &- C_x \int d\mathbf{r} n^{4/3}(\mathbf{r}) + \int d\mathbf{r} n(\mathbf{r}) V_{\text{ext}}(\mathbf{r}, \mathbf{R}) \end{aligned}$$

The Kohn-Sham Formulation

Central assertion of KS formulation: Consider a system of N_e Non-interacting electrons subject to an “external” potential V_{KS} . It is possible to choose this potential such that the ground state density of the non-interacting system is the same as that of an interacting System subject to a particular external potential V_{ext} .

A non-interacting system is separable and, therefore, described by a set of single-particle orbitals $\psi_i(\mathbf{r},s)$, $i=1,\dots,N_e$, such that the wave function is given by a Slater determinant:

$$\Psi(\mathbf{x}_1, \dots, \mathbf{x}_{N_e}) = \frac{1}{\sqrt{N_e!}} \det[\psi_1(\mathbf{x}_1) \cdots \psi_{N_e}(\mathbf{x}_{N_e})]$$

The density is given by

$$n(\mathbf{r}) = \sum_{i=1}^{N_e} \sum_s |\psi_i(\mathbf{x})|^2 \quad \langle \psi_i | \psi_j \rangle = \delta_{ij}$$

The kinetic energy is given by

$$T_s = -\frac{1}{2} \sum_{i=1}^{N_e} \sum_s \int d\mathbf{r} \psi_i^*(\mathbf{x}) \nabla^2 \psi_i(\mathbf{x})$$

Kohn-Sham Equations

The variational problem can now be formulated as:

$$\frac{\delta}{\delta \psi_i^*(\mathbf{x})} \left[-\frac{1}{2} \sum_j \int d\mathbf{x} \psi_j^*(\mathbf{x}) \nabla^2 \psi_j(\mathbf{x}) + \int d\mathbf{x} V_{\text{KS}}(\mathbf{r}) n(\mathbf{r}) \right. \\ \left. - \sum_{j,k} \lambda_{jk} \left(\int d\mathbf{x} \psi_j^*(\mathbf{x}) \psi_k(\mathbf{x}) - \delta_{ij} \right) \right] = 0$$

The Kohn-Sham Potential

Kohn-Sham functional:

$$\begin{aligned} E_{\text{KS}}[n] &= T_{\text{nonint}}[\{\psi\}] + \int d\mathbf{r} V_{\text{KS}}(\mathbf{r})n(\mathbf{r}) \\ &= T_{\text{nonint}}[\{\psi\}] + \frac{1}{2} \int d\mathbf{r} d\mathbf{r}' \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} \\ &\quad + E_{\text{xc}}[n] + \int d\mathbf{r} V_{\text{ext}}(\mathbf{r})n(\mathbf{r}) \end{aligned}$$

where, in principle,

$$E_{\text{xc}}[n] = F[n] - \frac{1}{2} \int d\mathbf{r} d\mathbf{r}' \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} - T_{\text{nonint}}[n]$$

is called the *exchange-correlation* energy functional. Its exact form is not known, so approximations are necessary.

$$V_{\text{KS}} = V_{\text{ext}}(\mathbf{r}) + \int d\mathbf{r}' \frac{n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} + \frac{\delta E_{\text{xc}}}{\delta n(\mathbf{r})}$$

Closed shell systems

Suppose there is an even number of electrons, so that they all can be paired up and the external potential $V_{\text{ext}}(\mathbf{r})$ is independent of spin. Spin-up and spin-down contribute equally to the total density:

$$n^{(\uparrow)}(\mathbf{r}) = n^{(\downarrow)}(\mathbf{r}) = \frac{1}{2}n(\mathbf{r})$$

Therefore, we only need $N_e/2$ Kohn-Sham orbitals, to each of which we assign an *occupation number* of $f = 2$. These orbitals satisfy the orthogonality condition:

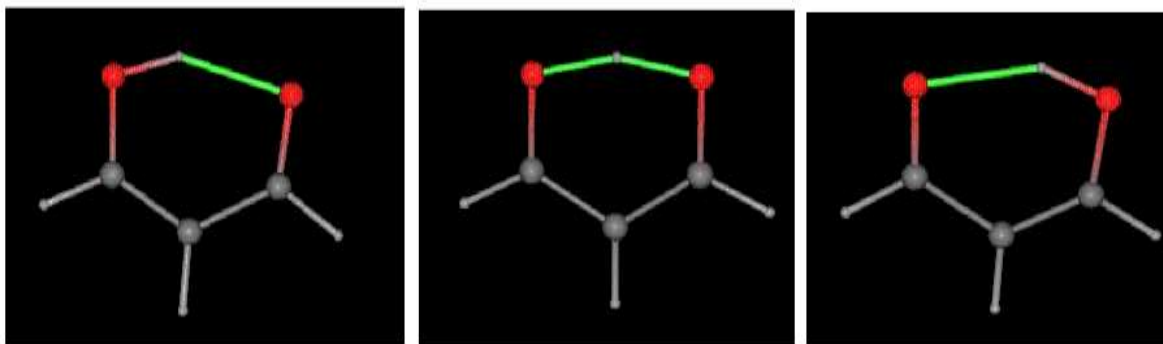
$$\int d\mathbf{r} \psi_i^*(\mathbf{r}) \psi_j(\mathbf{r}) = f \delta_{ij}$$

The density and kinetic energy are then given by

$$n(\mathbf{r}) = \sum_{i=1}^{N_e/2} |\psi_i(\mathbf{r})|^2$$
$$T_s = -\frac{1}{2} \sum_{i=1}^{N_e/2} \langle \psi_i(\mathbf{r}) | \nabla^2 | \psi_i(\mathbf{r}) \rangle$$

Some simple results from DFT

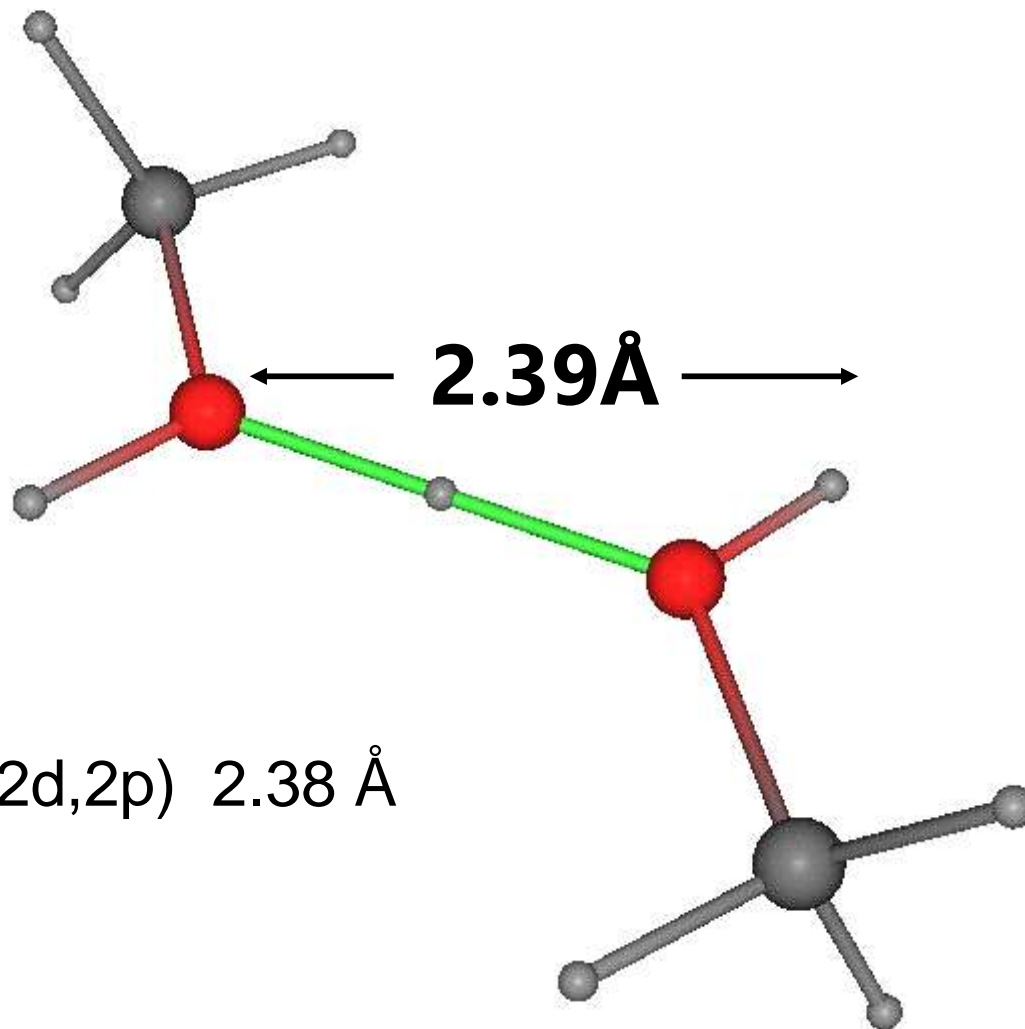
	TF	KS-LDA	HF	Expt.
Ne	-165.61	-128.12	-128.55	-128.94
Ar	-652.72	-525.85	-526.82	-527.6



$$E_{\text{barrier}}(\text{DFT}) = 3.6 \text{ kcal/mol}$$

$$E_{\text{barrier}}(\text{MP4}) = 4.1 \text{ kcal/mol}$$

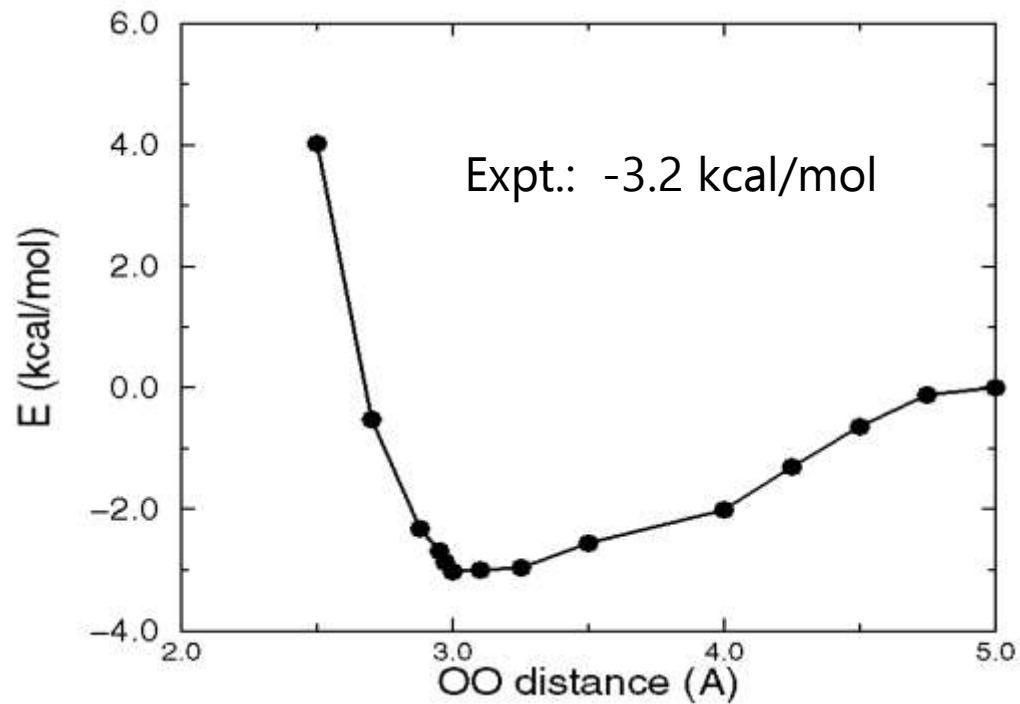
Geometry of the protonated methanol dimer



MP2 6-311G (2d,2p) 2.38 Å

Results methanol

Dimer dissociation curve of a *neutral* dimer



Lecture Summary

- Density functional theory is an exact reformulation of many-body quantum mechanics in terms of the probability density rather than the wave function
- The ground-state energy can be obtained by minimization of the energy functional $E[n]$. All we know about the functional is that it exists, however, its form is unknown.
- Kohn-Sham reformulation in terms of single-particle orbitals helps in the development of approximations and is the form used in current density functional calculations today.

experimental methods

Get a diamond anvil cell



Get beamtime on a synchrotron



Load your cell. Put medium.



Go to synchrotron



Run your experiment



computational methods

abinit.org

Get an ab initio software package



Get time on a supercomputer

```
kptopt 1
ngkpt 6 6 6
ecut 16 pawecutdg 36
nstep 1000 ixc 7 nshiftk 1
natom 20 ntypat 3 znuc1 26
typat 1 2 3 3 1 2 3
ecutsm 0.5 dilatmx 1.2
```

Input your structure. Choose pseudos, XCs.

```
: vi fesio3.MxS1.inp
```

Sign In

User Name

Password

Go to supercomputer

```
[rcaracas@service3] /home/rcaracas $ qu
```

Job ID	Username	Queue	Jobname	SessID	NDS	TSK	Req'd Memory	Req'd Time	Elap S	Time
1011554.service	rcaracas	long_hpt	d30_d00Y.P	24139	2	16	16gb	120:0	R	06:16
10115159.service	rcaracas	long_hpt	d35_d00Y.P	5509	2	16	16gb	120:0	R	00:36
10115626.service	rcaracas	long_hpt	pv8_00GPa	8203	1	8	7813mb	120:0	R	39:48
10115627.service	rcaracas	long_hpt	pv8_20GPa	27010	1	8	7813mb	120:0	R	39:39
10115628.service	rcaracas	long_hpt	pv8_40GPa	32353	1	8	7813mb	120:0	R	39:38
10115629.service	rcaracas	long_hpt	pv8_60GPa	30866	1	8	7813mb	120:0	R	39:38
10115630.service	rcaracas	long_hpt	pv8_80GPa	26176	1	8	7813mb	120:0	R	39:04
10115631.service	rcaracas	long_hpt	pv8_100GPa	29547	1	8	7813mb	120:0	R	38:48
10115632.service	rcaracas	long_hpt	pv8_120GPa	24968	1	8	7813mb	120:0	R	38:45

Run your experiment

What is it hard to calculate ?

Transport properties: thermal conductivity, electrical conductivity of insulators, rheology, diffusion

Excited electronic states: optical spectra ☹️ (=constants?)

Width of IR/Raman peaks, **Melting curves**, **Fluid properties**

What we can calculate ?

Electronic properties: orbital energies, chemical bonding, electrical conductivity

Structural properties: prediction of structures (under extreme conditions),
phase diagrams, surfaces, interfaces, amorphous solids

Mechanical properties: elasticity, compressibility, thermal expansion

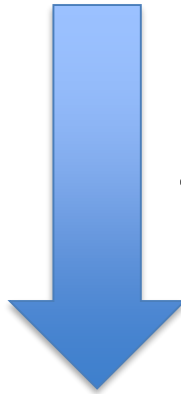
Dielectric properties: hybridizations, atomic dynamic charges, dielectric susceptibilities,
polarization, non-linear optical coefficients, piezoelectric tensor

Spectroscopic properties: Raman spectra with peak position and intensity, IR peaks

Dynamical properties: phonons, lattice instabilities, prediction of structures, thermodynamic properties, phase diagrams, thermal expansion

A set of N particles with masses m_n and initial positions X_n

$$t: \quad \mathbf{x}_t \quad \dot{\mathbf{x}}_t (=V) \quad \ddot{\mathbf{x}}_t (=A) \quad m$$

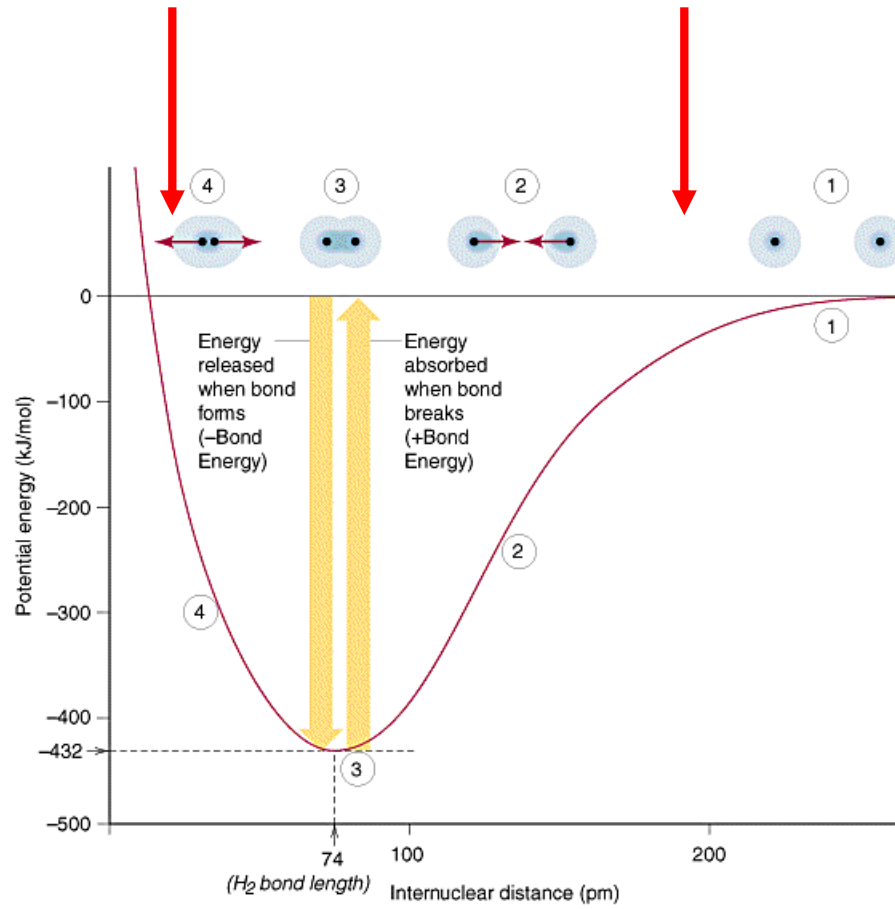


Compute new F
then $F = ma$

$$t+1: \quad \mathbf{x}_{t+1} \quad \dot{\mathbf{x}}_{t+1} \quad \ddot{\mathbf{x}}_{t+1} \quad m$$

Repulsive zone

Attractive zone



Two-body potentials *or* pair potentials

Lennard-Jones

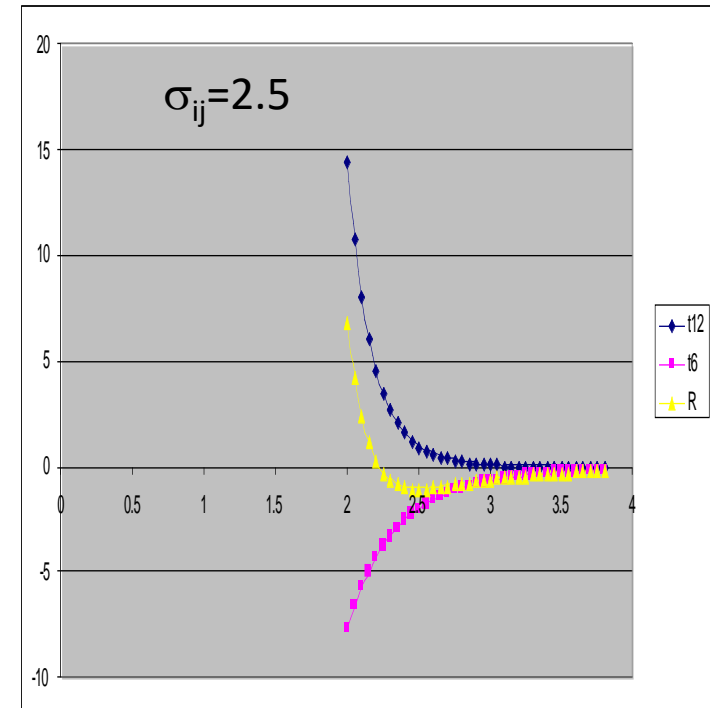
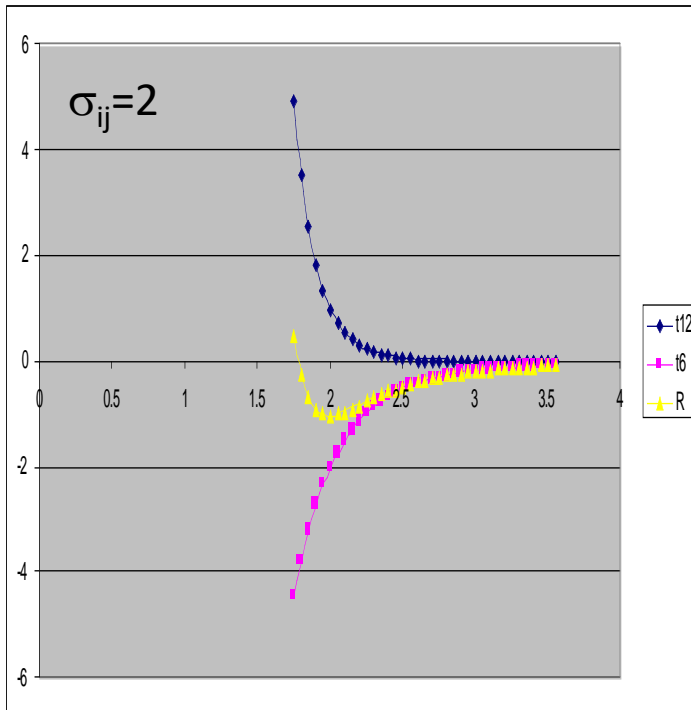
$$V(r_{ij}) = \epsilon_{ij} \left[\left(\frac{\sigma_{ij}}{r_{ij}} \right)^{12} - 2 \left(\frac{\sigma_{ij}}{r_{ij}} \right)^6 \right]$$

Morse

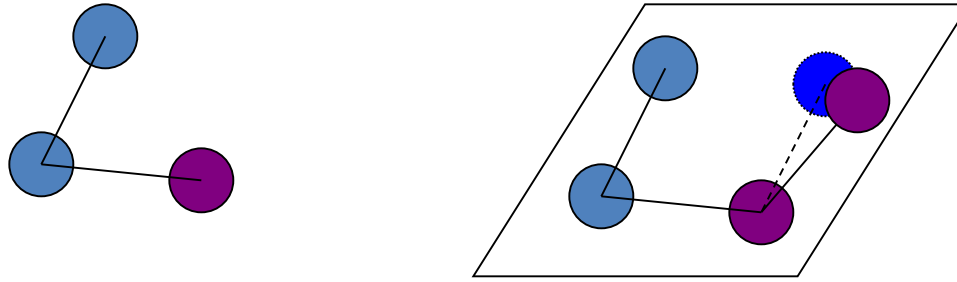
$$V(r_{ij}) = \epsilon_{ij} \left[\left(1 - e^{-\alpha_{ij}(r_{ij} - \sigma_{ij})} \right)^2 - 1 \right]$$

Buckingham

$$V(r_{ij}) = A_{ij} \exp \left(-\frac{r_{ij}}{\sigma_{ij}} \right) - C_{ij} \left(\frac{\sigma_{ij}}{r_{ij}} \right)^6$$



Multibody potentials



$$V_{ij}(r_{ij}) = V_{repulsive}(r_{ij}) + b_{ijk} V_{attractive}(r_{ij})$$

2 body



3+ body



Force fields – very good for molecules

The functional form of the AMBER force field is

$$V(r^N) = \sum_{\text{bonds}} \frac{1}{2} k_b (l - l_0)^2 + \sum_{\text{angles}} \frac{1}{2} k_a (\theta - \theta_0)^2$$
$$+ \sum_{\text{torsions}} \frac{1}{2} V_n [1 + \cos(n\omega - \gamma)] + \sum_{j=1}^{N-1} \sum_{i=j+1}^N \left\{ 4\epsilon_{i,j} \left[\left(\frac{\sigma_{ij}}{r_{ij}} \right)^{12} - \left(\frac{\sigma_{ij}}{r_{ij}} \right)^6 \right] + \frac{q_i q_j}{4\pi\epsilon_0 r_{ij}} \right\}$$

Many other examples:

CHARMM, polarizable, valence-bond models,

Tersoff interatomic potential

$$E = \sum_i V_1(r_i) + \sum_{i<j} V_2(r_i, r_j) + \sum_{i<j<k} V_3(r_i, r_j, r_k) + \dots$$

$$E = \sum_i E_i = \frac{1}{2} \sum_{i \neq j} V_{ij}$$

$$V_{ij} = f_C(r_{ij}) [f_R(r_{ij}) + b_{ij} f_A(r_{ij})]$$

$$f_R(r) = A e^{(-\lambda_1 r)}$$

$$f_C(r) = \begin{cases} 1, & r < R - D \\ \frac{1}{2} - \frac{1}{2} \sin\left[\frac{\pi}{2}(r - R)/D\right], & R - D < r < R + D \\ 0, & r > R + D \end{cases}$$

$$b_{ij} = \frac{1}{(1 + \beta^n \zeta_{ij}^n)^{1/2n}}$$

$$\zeta_{ij} = \sum_{k \neq i, j} f_C(r_{ik}) g(\theta_{ijk}) e^{[\lambda_3^3 (r_{ij} - r_{ik})^3]}$$

$$g(\theta) = 1 + \frac{c^2}{d^2} - \frac{c^2}{[d^2 + (h - \cos\theta)^2]}$$

Table 7.1: Parameters for the Tersoff potential, for atoms of carbon and silicon.

	C	Si
A (eV)	1393.6	1830.8
B (eV)	346.7	471.18
λ (\AA^{-1})	3.4879	2.4799
μ (\AA^{-1})	2.2119	1.7322
β	1.5724×10^{-7}	1.1×10^{-6}
n	0.72751	0.78734
c	3.8049×10^4	1.0039×10^5
d	4.384	16.217
h	-0.57058	-0.59825
R (\AA)	1.8	2.7
S (\AA)	2.1	3.0

Non-empirical = first-principles or *ab initio*

- the energy is *exactly* calculated
- no experimental input

- + transferability, accuracy, many properties
- small systems

Schrödinger equation

time-dependent $H(t) |y_n(t)\rangle = i\hbar \frac{d}{dt} |y_n(t)\rangle$

time-independent

$$\left[-\frac{\hbar^2}{2m} \nabla^2 + U(r) \right] |y_n\rangle = E_n |y_n\rangle$$

E_n

Eigenvalues

$|y_n\rangle$

Eigenstates

$$H\Psi = [T + V + U] \Psi = \left[\sum_i^N -\frac{\hbar^2}{2m} \nabla_i^2 - \sum_i^N V(\vec{r}_i) + \sum_{i<j} U(\vec{r}_i, \vec{r}_j) \right] \Psi = E\Psi$$

Kinetic energy of the electrons

External potential

Schrödinger equation involves many-body interactions

Wavefunction

$|\psi_n\rangle$

-contains all the measurable information
-gives a measure of probability:

$$\langle \psi_n | \psi_n \rangle = \psi^* \psi$$

$|\psi_n\rangle$

~ many-particle wavefunction:
depends on the position of electrons and nuclei
scales factorial

For a system like C atom: 6 electrons : 6! evaluations = 720

For a system like O atom: 8 electrons : 8! evaluations = 40320

For a system like Ne atom: 10 electrons: 10! Evaluations = 3628800

For one SiO₂ molecule: 30electrons+3nuclei= 8.68E³⁶ evaluations

UNPRACTICAL!

DENSITY FUNCTIONAL THEORY

- What is DFT ?
- Codes
- Planewaves and pseudopotentials
- Types of calculation
- Input key parameters
- Standard output
- Examples of properties:
 - Electronic band structure
 - Equation of state
 - Elastic constants
 - Atomic charges
 - Raman and Infrared spectra
 - Lattice dynamics and thermodynamics

THEORETICAL
ASPECTS

PRACTICAL
ASPECTS

EXAMPLES

What is DFT

Idea:

one determines the electron density (Kohn, Sham in the sixties: the one responsible for the chemical bonds) from which by proper integrations and derivations all the other properties are obtained.

INPUT

Structure: atomic types + atomic positions =
initial guess of the geometry

There is no experimental input !

What is DFT

$$E[n(r)] = T_s[n(r)] + E_{ext}[n(r)] + E_{col}[n(r)] + E_{xc}[n(r)]$$

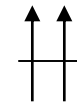
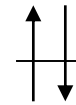
Kinetic energy of non-interacting electrons

Energy term due to exterior

Coulombian energy =
 $E_{ee} + E_{eN} + E_{NN}$

Exchange correlation energy

Electron spin:



Decrease
energy

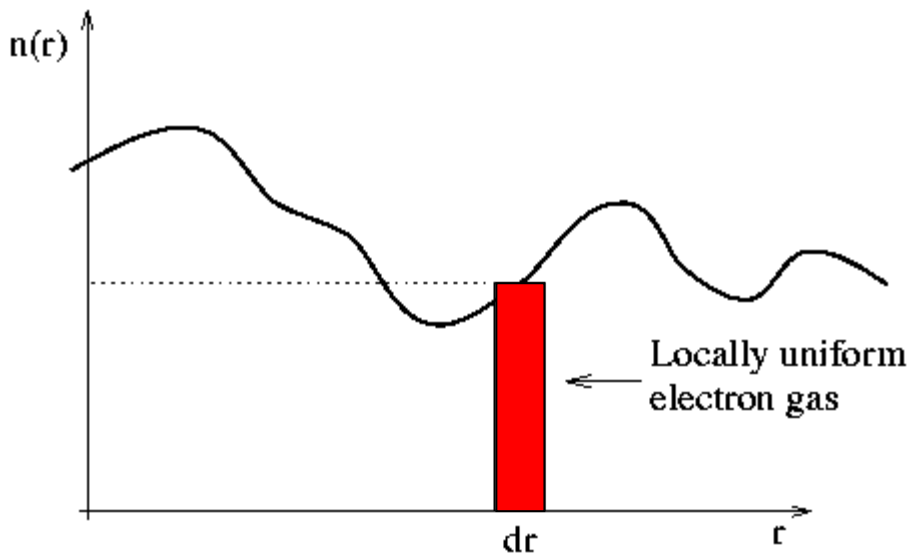
Increases

$$n(r) = N \int d^3r_2 \int d^3r_3 \dots \int d^3r_N \Psi^*(r, r_2, r_3, \dots, r_N) \Psi(r, r_2, r_3, \dots, r_N)$$

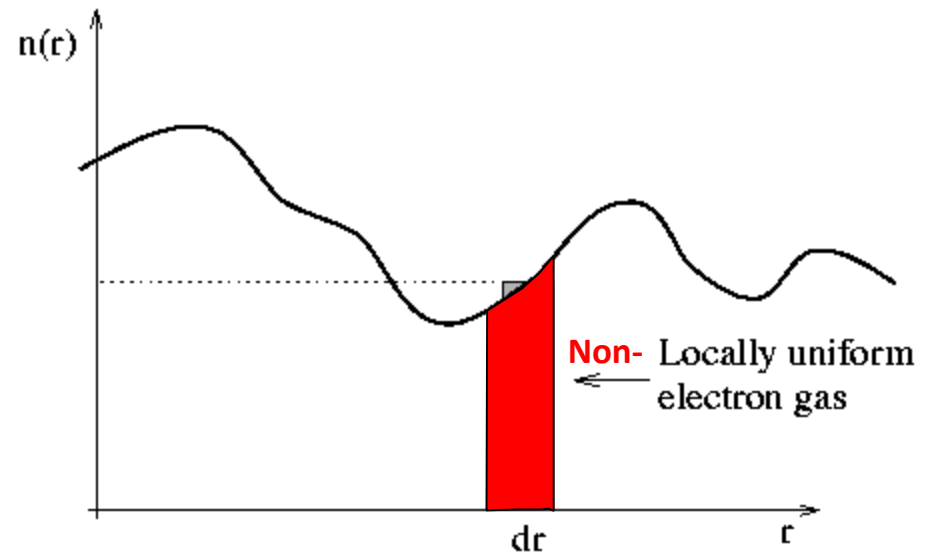
E_{xc} : LDA vs. GGA

LDA = Local Density Approximation

GGA = Generalized Gradient Approximation



$$E_{xc} = \int n(r) e_{xc}(r) dr$$



$$E_{xc} = \int n(r) e_{xc}(r, Dn) dr$$

Flowchart of a standard DFT calculation

Initialize wavefunctions and electron density

Compute energy and potential

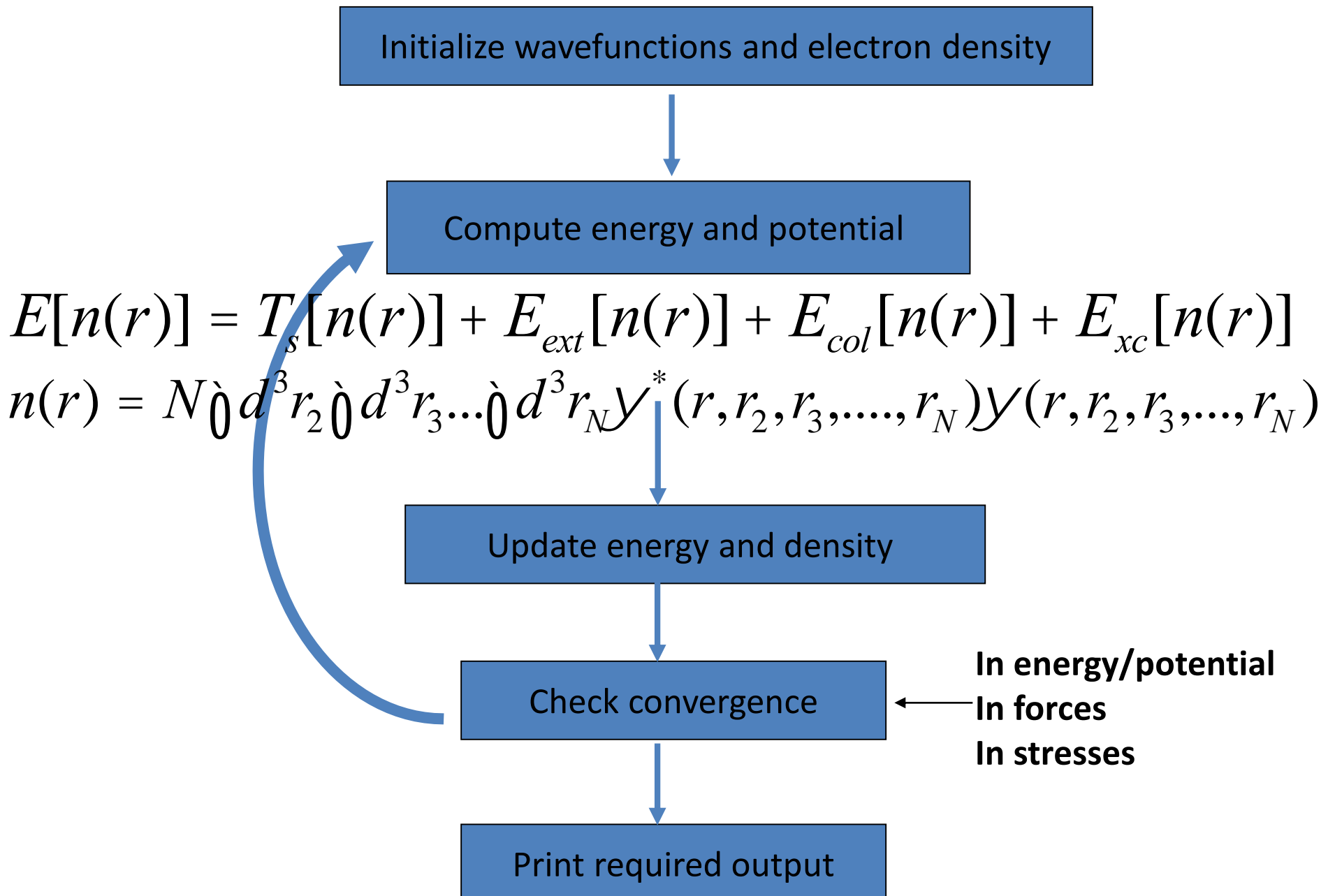
$$E[n(r)] = T_s[n(r)] + E_{ext}[n(r)] + E_{col}[n(r)] + E_{xc}[n(r)]$$
$$n(r) = N \int d^3r_2 \int d^3r_3 \dots \int d^3r_N \psi^*(r, r_2, r_3, \dots, r_N) \psi(r, r_2, r_3, \dots, r_N)$$

Update energy and density

Check convergence

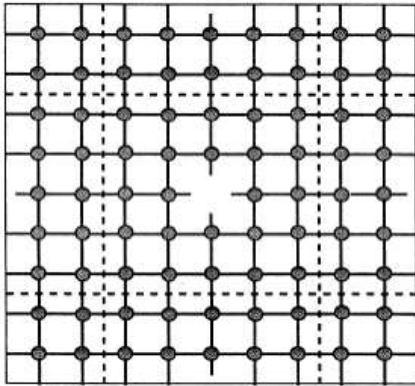
In energy/potential
In forces
In stresses

Print required output



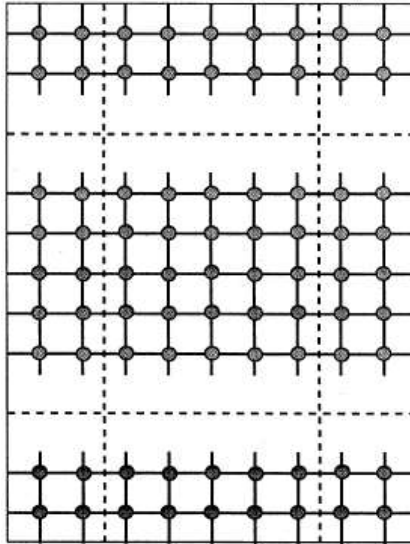
Crystal structure – non-periodic systems

Point-defect

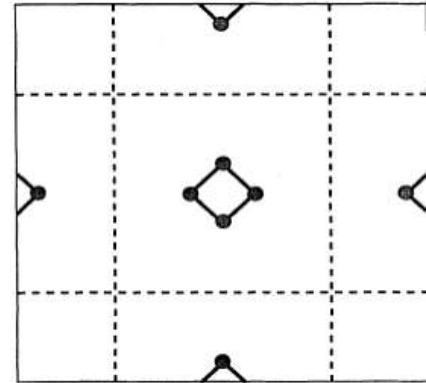


←→
“big enough”

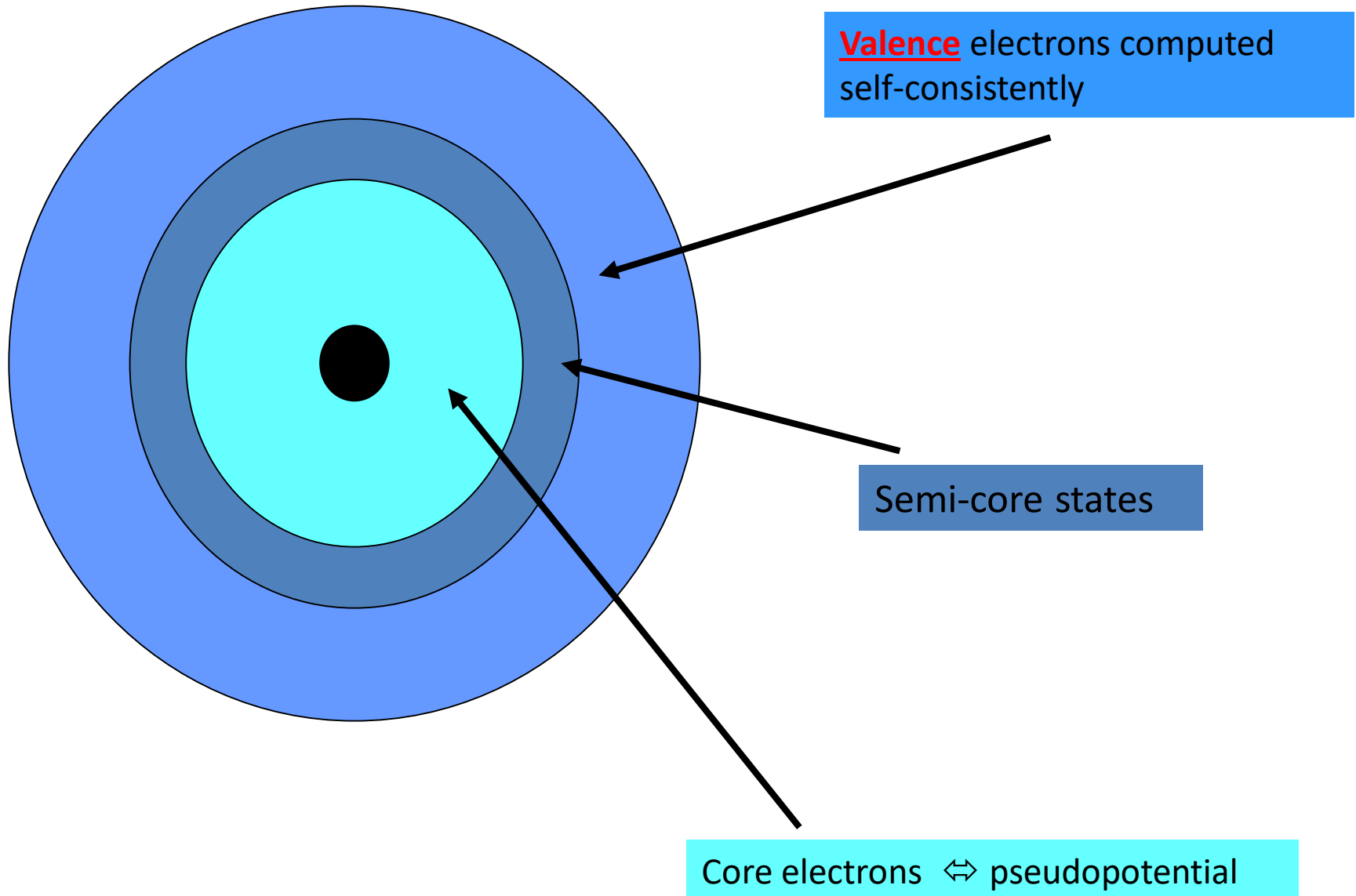
Surface



Molecule



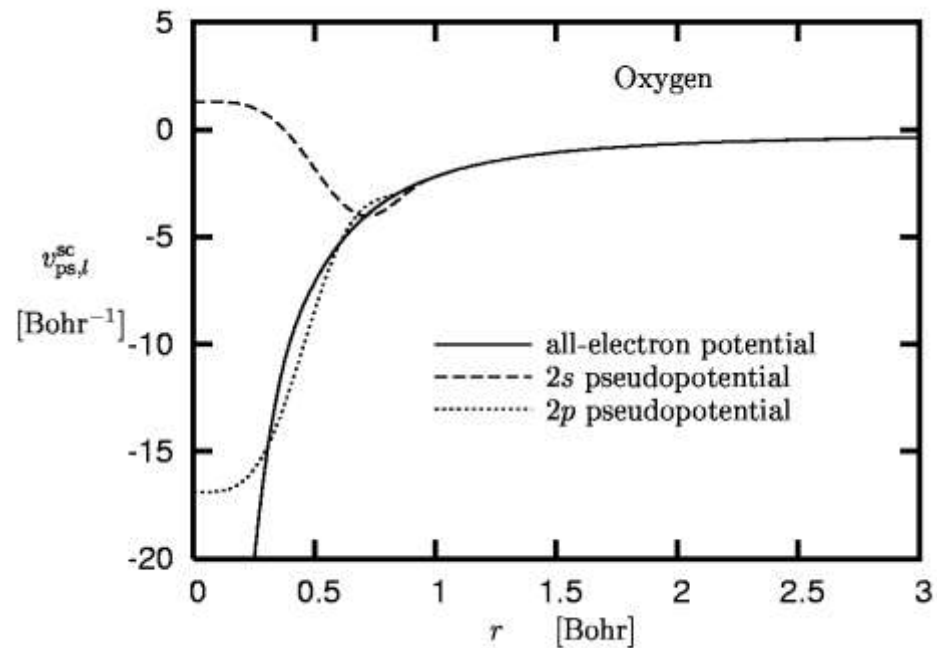
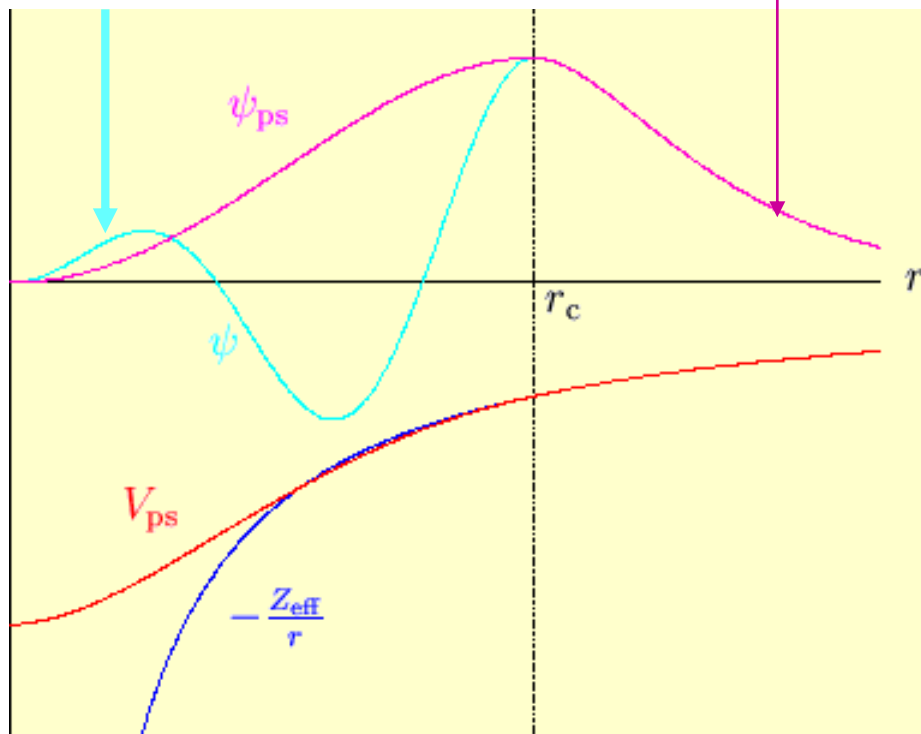
Input key parameters - pseudopotentials



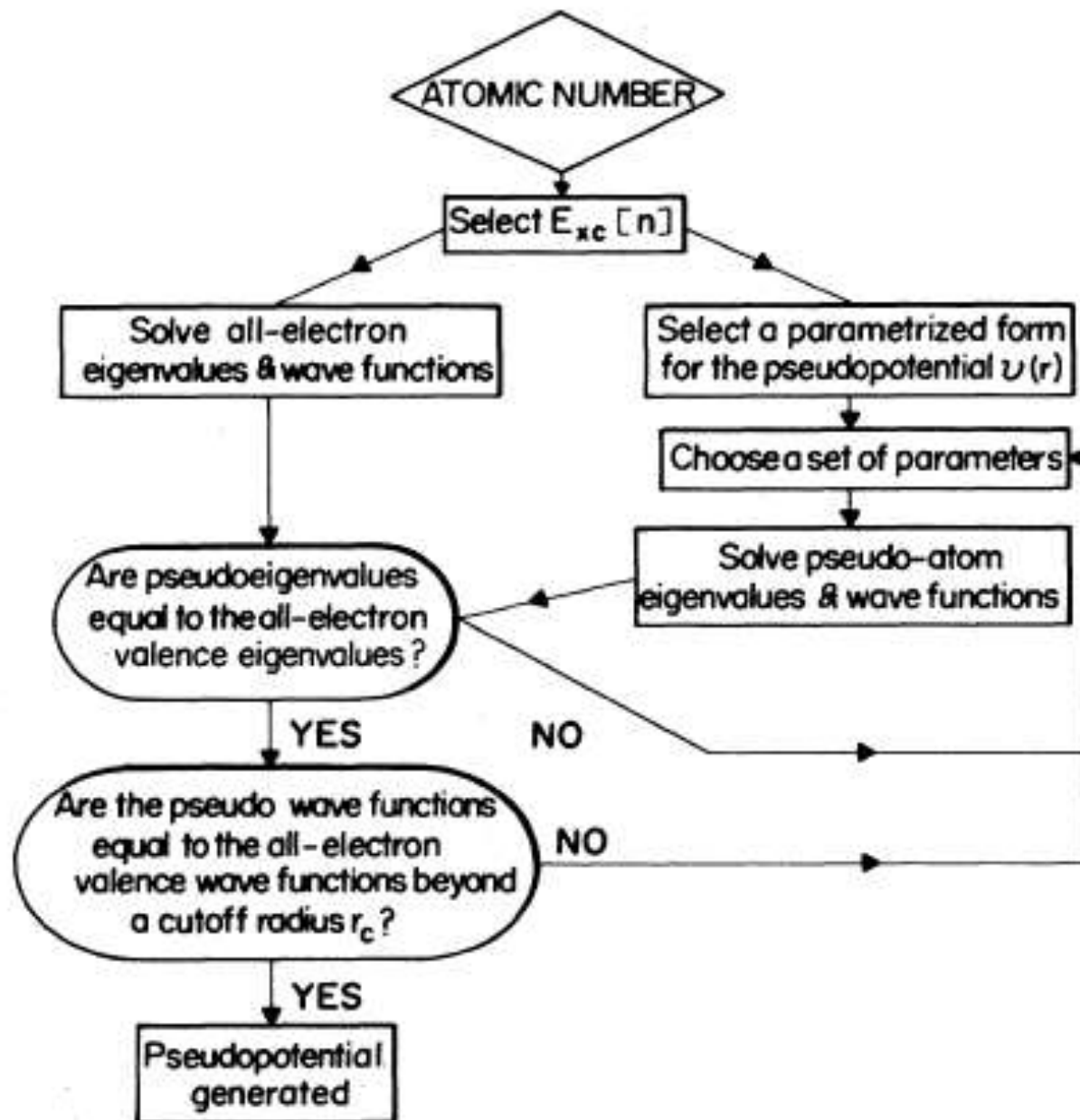
Input key parameters - pseudopotentials

Pseudo-wavefunction

All electron wavefunction



Input key parameters - pseudopotentials



Input key parameters - pseudopotentials

Opium - pseudopotential generation project

Latest release
 Version 2.0.6 - Sep. 29, 2006
 View the [2.0.5 -> 2.0.6 Changelog](#)

Disclaimer: Opium is distributed under the [GNU/General Public License](#). Like most open source software, it is not guaranteed to be bug free. Use at your own risk and please report any bugs to the Opium [mailing list](#).

Scientific background / Features

The all-atom pseudopotential method is now a well established tool in condensed matter physics, computational chemistry and material science. At the present time there are a good number of codes available, both commercial and in the public domain, that perform electronic structure calculations of molecules and solids based on the pseudopotential scheme. The most important input information that these programs require are the pseudopotentials used in the calculation.

Features included in the current release of Opium:

- Scalar-relativistic [1] and non-relativistic pseudopotential generation
- Ability to construct Optimized (FP-LMTO) [2] or Krieger [3] pseudopotentials
- Partial core correction of Louie, Froyen and Cohen [4]
- Can test and generate and test pseudopotentials that support semicore states
- Ghost state checking following the method suggested by Gonze, Stampf, and Scheffler [5]
- Automatic plotting of wavefunctions, potentials, and density using [cmtotrace](#)
- Implementation of the designed non-local potential approach of Hamann and Rappaport [6]

Computational packages that are currently supported by Opium output formats

Format	Package
*rcpp	CASTEP
*fls	ABINIT
*ncpp	PWSCF
*surf	Eb

LDA pseudopotential files from FHI code

H																	He										
Li	Be											B	C	N	O	F	Ne										
Na	Mg											Al	Si	P	S	Cl	Ar										
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr										
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe										
Cs	Ba											Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn	
Fr	Ra											Rf	Ha	Sg	Ns	Hs	Mt										
												La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu	
												Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lr	

■ only valence electrons ■ with semi-core ■ not available

Note : The size of the file may vary between 30 kB and 150kB.

The fhi98PP pseudopotential program

The package fhi98PP [1] in the present update of 27-06-02 allows one to

- Generate norm-conserving pseudopotentials of the Hamann [2] and Troulier-Martin [3] types, employing common parameterizations of the local-density approximation and generalized gradient approximations for exchange and correlation.
- Check the transferability of semi-pseudopotentials, in particular in search for (physical) ghost states of their fully separable (Hermann-Eskander) representation [4].
- Spin-polarized atoms can be calculated, and pseudopotentials for word John-Stem exchange calculations can be produced. These features are at a developmental stage and require care.

A database of generic input data files for most elements is part of this package. General background about pseudopotentials and practical considerations for generating them may be found in [1]. A [tutorial manual](#) and worked-out examples are available as well.

Unix shell scripts serve as command-line interfaces to facilitate the above tasks. Their pseudopotential output is compatible with the input format of the third program [7] and can be readily adapted to fit other codes. For visual inspection of results, these scripts provide graphics files to be viewed with the public domain plotting tool [VMDRAW](#).

Download

The source distribution is available from the [download page](#).

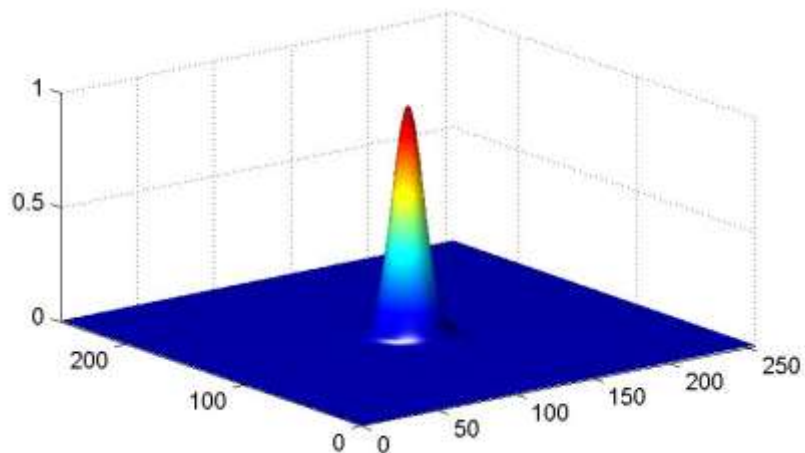
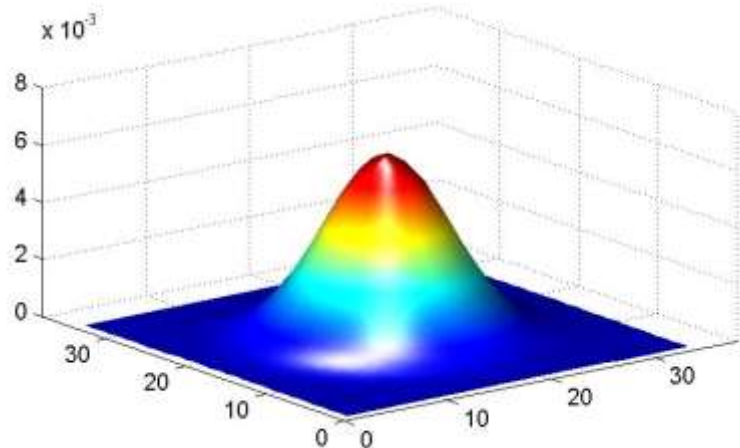
References

1. M. Fuchs, M. Scheffler, *Comput. Phys. Commun., Comput. Phys. Commun.* **119**, 87-96 (1998).
2. "All-atom pseudopotentials for electronic structure calculations of poly-atomic systems using density-functional theory". [fhi98pp.ccs](#)
3. [fhi98pp.ccs](#), European Patent: [fhi98pp.ccs](#)
4. D. R. Hamann, *Phys. Rev. B* **40**, 2980 (1989).
5. N. Troulier, J. L. Martin, *Phys. Rev. B* **43**, 1803 (1991).
6. K. Gonze, P. Stampf, M. Scheffler, *Phys. Rev. B* **44**, 2503 (1991).
7. M. Bockstedt, A. Foy, J. Neugebauer, M. Scheffler, *Comput. Phys. Commun.* **107**, 187 (1997).

[Home page of the Theory Department of FHI](#)

Last modified: Fri Aug 7 10:00:00 MET DST 2003

localized basis



The number of gaussian functions summed to describe the inner shell orbital

The number of gaussian functions summed in the second STO

3-21G

The number of gaussian functions that comprise the first STO of the double zeta

CARBON (6S,3P) -> [3S,2P]

S 3			
1	172.256000	0.617669000E-01	
2	25.9109000	0.358794000	
3	5.53335000	0.700713000	
L 2			
1	3.66498000	-0.395897000	0.23640000
2	0.770545000	1.21584000	0.860619000
L 1			
1	0.195857000	1.00000000	1.00000000



planewaves

wavelength

$$\lambda = 2\pi/G$$

frequency

$$f = \omega/2\pi$$

period

$$T = 1/f = 2\pi/\omega$$

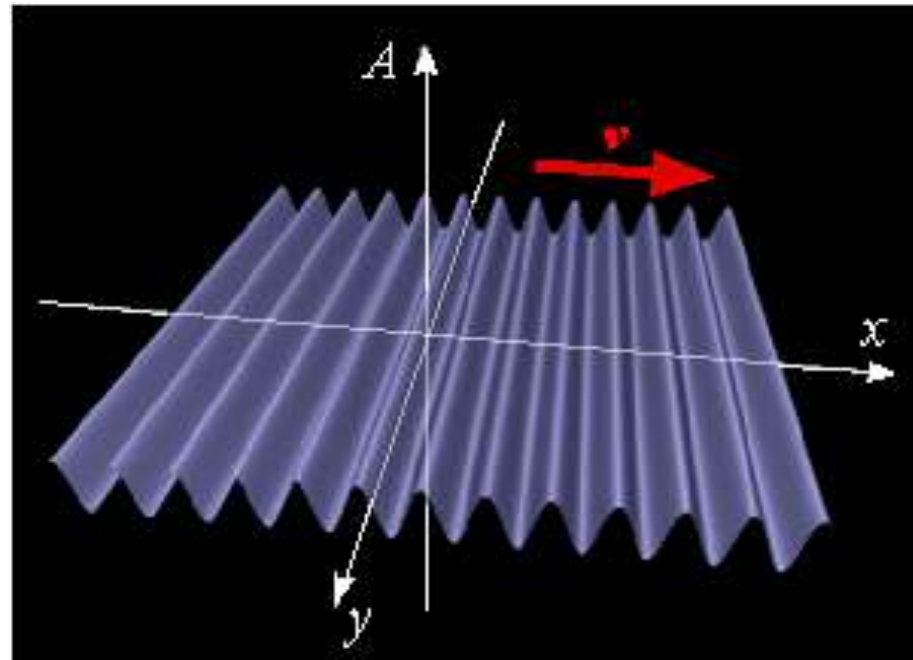
velocity

$$v = \lambda/T = \omega/k$$

Planewaves are characterized by their

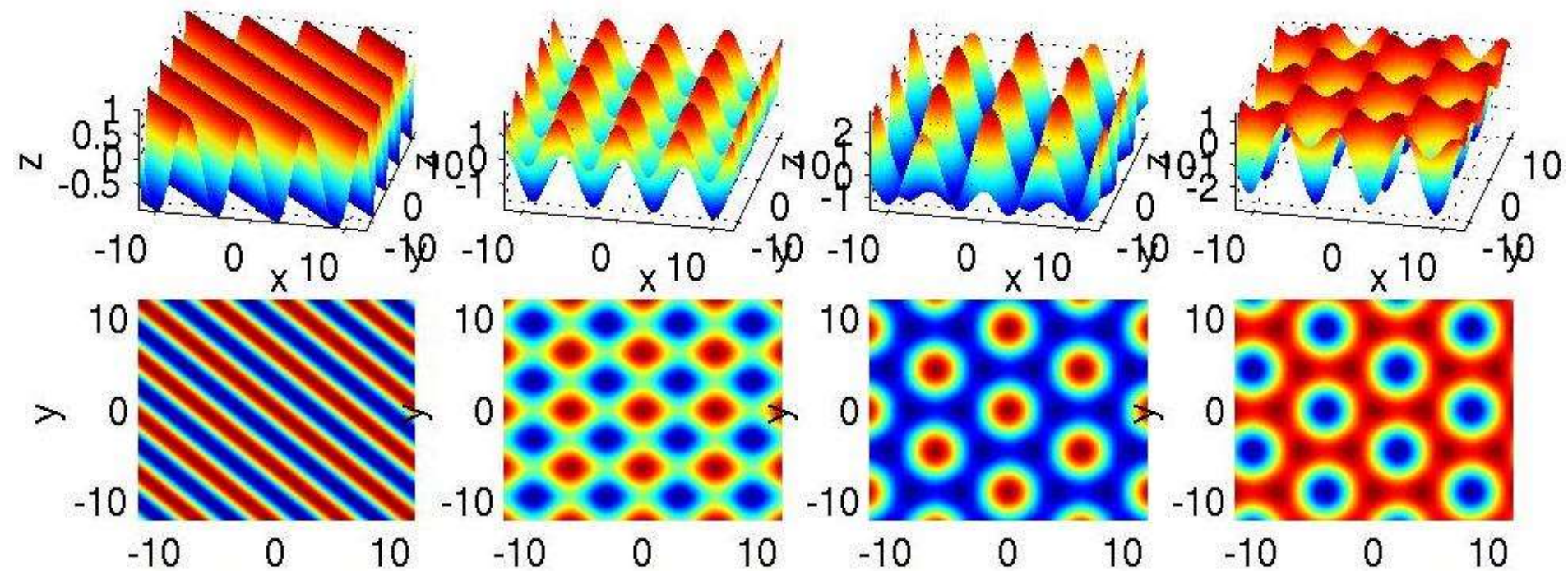
wavevector \mathbf{G} \rightarrow

angular speed ω



planewaves

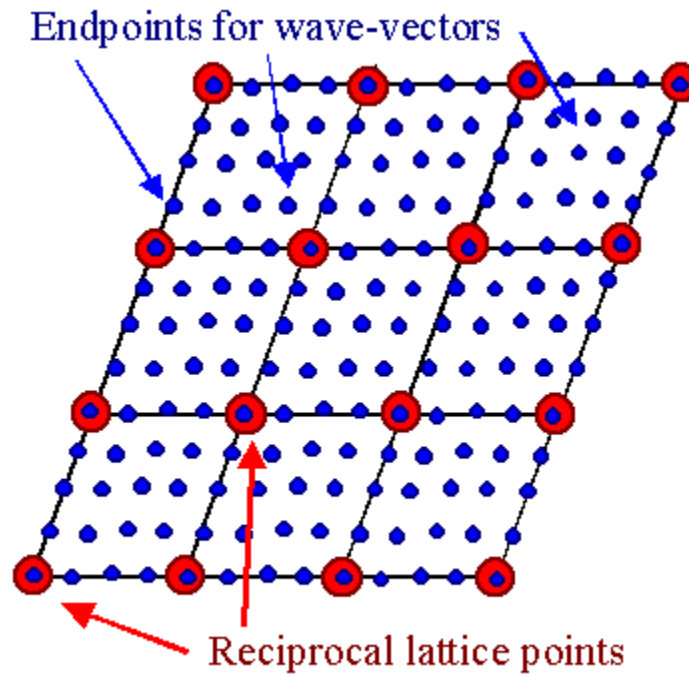
The electron density is obtained by superposition of planewaves

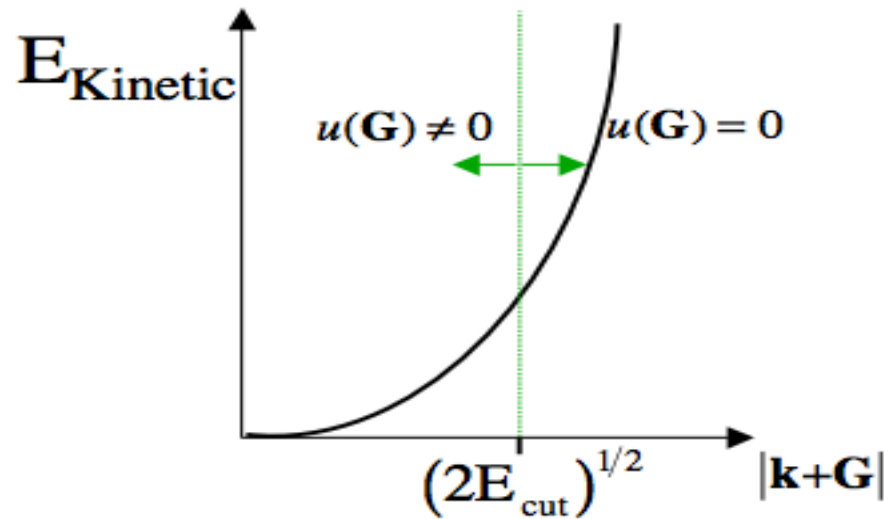
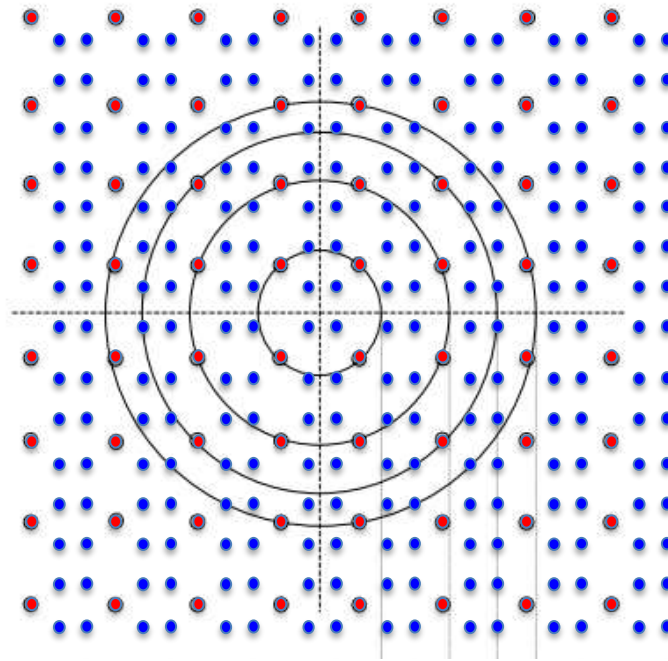


Input key parameters - K-points

$$\psi_i^k(r)$$

Limited set of k points ~
boundary conditions





after: <http://www.psi-k.org/Psik-training/Gonze-1.pdf>

PRACTICAL ASPECTS: Properties

Electronic properties: electronic band structure, orbital energies, chemical bonding, hybridization, insulator/metallic character, Fermi surface, X-ray diffraction diagrams

Structural properties: crystal structures, prediction of structures under extreme conditions, prediction of phase transitions, analysis of hypothetical structures

Mechanical properties: elasticity, compressibility

Dielectric properties: hybridizations, atomic dynamic charges, dielectric susceptibilities, polarization, non-linear optical coefficients, piezoelectric tensor

Spectroscopic properties: Raman and Infrared active modes, silent modes, symmetry analysis of these modes

Dynamical properties: phonons, lattice instabilities, prediction of structures, study of phase transitions, thermodynamic properties, electron-phonon coupling

How to choose between LDA and GGA ?

- relatively homogeneous systems LDA
- highly inhomogeneous systems GGA

- elements from “p” bloc LDA
- transitional metals GGA

- LDA **under**estimates volume and distances
- GGA **over**estimates volume and distances

- best: try both: you bracket the experimental value

How to choose pseudopotentials ?

- the pseudopotential must be for the same XC as the calculation
- preferably start with a Troullier-Martins-type
- if it does not work try more advanced schemes
- check semi-core states
- check structural parameters for the compound not element!

Values of the parameters

How to choose no. of planewaves and k-points ?

- check CONVERGENCE of the physical properties

```
setup2: Arith. and geom. avg. npw (full set) are      281.500      281.467
```

```
=====
iter  Etot(hartree)    deltaE(h)  residm    vres2    diffor    maxfor
ETOT  1  -12.038846358142  -1.204E+01 1.200E-02 2.114E+01 0.000E+00 0.000E+00
ETOT  2  -12.047681193210  -8.835E-03 5.326E-04 7.606E-02 3.000E-29 3.000E-29
ETOT  3  -12.047697107611  -1.591E-05 2.989E-04 2.507E-03 3.078E-29 1.094E-30
ETOT  4  -12.047697613689  -5.061E-07 2.156E-04 5.200E-06 1.113E-30 2.929E-31
ETOT  5  -12.047697613315   3.739E-10 1.131E-04 4.835E-06 1.504E-30 1.484E-30
ETOT  6  -12.047697614423  -1.109E-09 6.917E-05 8.465E-08 1.484E-30 0.000E+00
```

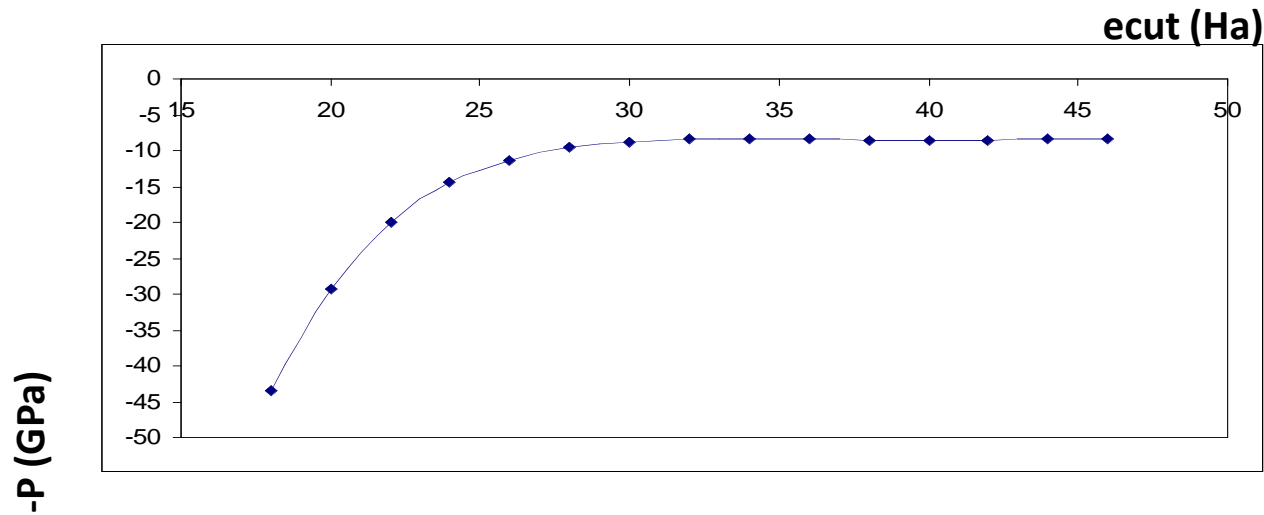
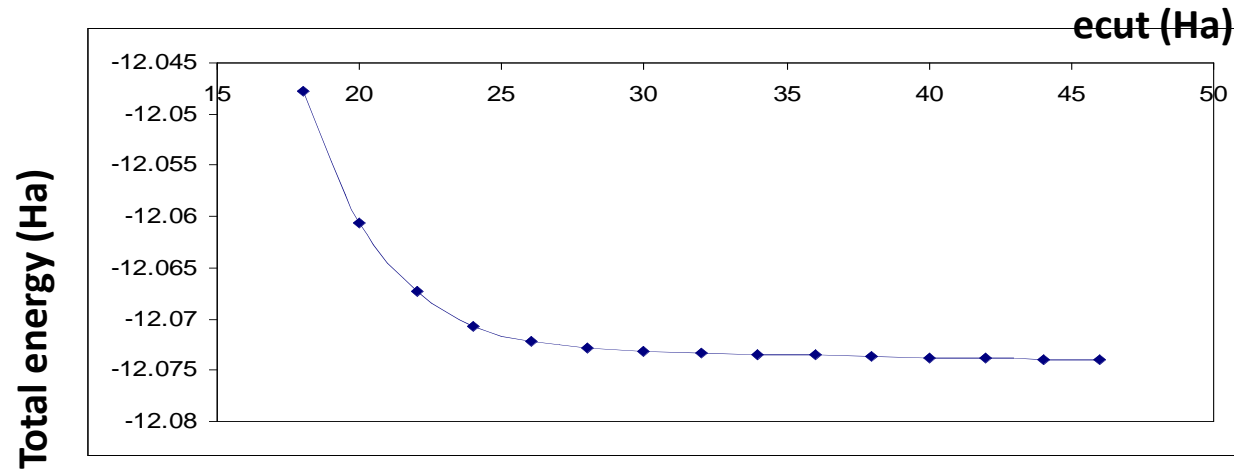
```
At SCF step    6, etot is converged :
for the second time, diff in etot= 1.109E-09 < toldfe= 1.000E-08
```

```
Cartesian components of stress tensor (hartree/bohr^3)
sigma(1 1)= 1.47510899E-03  sigma(3 2)= 0.00000000E+00
sigma(2 2)= 1.47510899E-03  sigma(3 1)= 0.00000000E+00
sigma(3 3)= 1.47510899E-03  sigma(2 1)= 0.00000000E+00
```

```
-----iterations are completed or convergence reached-----
```

Values of the parameters

- check CONVERGENCE of the physical properties



Usual output of calculations (in ABINIT)

Log (=STDOUT) file

detailed information about the run; energies, forces, errors, warnings, etc.

Output file:

simplified “clear” output:

full list of run parameters

total energy; electronic band eigenvalues; pressure; magnetization, etc.

Charge density = DEN

Electronic density of states = DOS

Analysis of the geometry = GEO

Wavefunctions = WFK, WFQ

Dynamical matrix = DDB

etc.

DFT codes

http://dft.sandia.gov/Quest/DFT_codes.html

<http://www.psi-k.org/>

• Periodic codes (principally)

- Local orbital basis codes
 - [QUEST: SeqQuest](#) - gaussian basis pseudopotential code
 - [SIESTA](#) - numerical atom-centered basis pseudopotential code
 - [CRYSTAL - CSE](#) - gaussian basis all-electron code
 - [AIMPRO](#)
 - [FPLO](#)
 - [OpenMX](#) - GPL - numerical atom-centered basis PP code (Ozal)
- All-electron (augmented methods) codes
 - [ELK](#) - GPL - FP-LAPW
(one branch from the [old EXCITING code](#))
 - [EXCITING](#) - FP-LAPW, focus on excited state properties (TDDF [license not apparent on website, probably open source] (another branch from the [old EXCITING code](#))
 - [FLEUR](#) - "freely available" - FLAPW code
 - [RSPt](#) - "Open Source" - FP-LMTO
 - [WIEN2k](#) - modest fee - full potential LAPW
- Plane wave and related (real space, wavelet, etc.) methods
 - [VASP](#) - although check out its [\(trial?\) spiffy new site](#)
 - [CASTEP and CETEP](#)
 - [CPMD](#)
 - [ABINIT](#) - GPL
 - [BigDFT](#) - wavelets
 - [Quantum-Espresso \(formerly PWscf\)](#) - GPL
 - [PEtot](#) - GPL
 - [DACAPO](#) - GPL
 - [Socorro](#) - GPL
 - [DFT++](#) - GPL
 - [Octopus](#) - GPL - real space TDDFT code
 - [Paratec](#)
 - [DoD Planewave](#)
 - [PARSEC](#) - GPL - real space, pseudopotential
 - [CP2K](#) - GPL (mixed basis DFT)
 - [GPAW](#) - GPL - real-space multigrid PAW code
 - [SPHINX](#)
 - [QBOX](#) - GPL - plane wave pseudopotential, large parallel

• Molecular codes (principally)

- [Gaussian.com](#) (unless, of course, you have been ["banned"](#))
- [NWChem](#)
- [DMol³](#)
- [Jaguar](#) - Schrodinger, Inc.
- [GAMESS](#) or [GAMESS-UK](#)
- [QCHEM](#)
- [NRLMOL](#)
- [MondoSCF](#) (Matt Challacombe's Home Page)
- [ADF - SCM](#)
- [deMon](#)
- [CADPAC - The Cambridge Analytic Derivatives Package](#)
- [PYQUANTE](#) - GPL - python-based development toolset for
- [TURBOMOLE](#) - DFT and HF for large molecular systems

About ABINIT

What is ABINIT?
 Latest news
 Partners / Sponsors
 Contact us
 Help ABINIT

ABINIT Package

Sources,help files...
 Binaries (summary)
 Browse latest sources

Extra Packages

Pseudopotentials
 PAW Atomic Data
 Atomic Densities
 Input Files

Documentation

Help files
 Input variables
 FAQ
 Benchmarks
 Gallery

Community

Netiquette
 Mailing lists
 Events
 Bibliographical database
 Job openings
 Problem reports
 Links
 Miscellaneous info

Developer's corner

Developer's docs
 Ongoing efforts
 How to contribute
 Miscellaneous

TLA Tutorial for ABINIT
 Getting started with TLA
 Writing change logs
 Advanced use of TLA

What is ABINIT ?

ABINIT is a package whose main program allows one to find the total energy, charge density and electronic structure of systems made of electrons and nuclei (molecules and periodic solids) within Density Functional Theory (DFT), using pseudopotentials and a planewave basis. ABINIT also includes options to optimize the geometry according to the DFT forces and stresses, or to perform molecular dynamics simulations using these forces, or to generate dynamical matrices, Born effective charges, and dielectric tensors. Excited states can be computed within the Time-Dependent Density Functional Theory (for molecules), or within Many-Body Perturbation Theory (the GW approximation). In addition to the main ABINIT code, different utility programs are provided.

First-principles computation of material properties : the ABINIT software project.

X. Gonze, J.-M. Beuken, R. Caracas, F. Detraux, M. Fuchs, G.-M. Rignanese, L. Sindic, M. Verstraete, G. Zerah, F. Jollet, M. Torrent, A. Roy, M. Mikami, Ph. Ghosez, J.-Y. Raty, D.C. Allan

Computational Materials Science, 25, 478-492 (2002)**A brief introduction to the ABINIT software package.**

X. Gonze, G.-M. Rignanese, M. Verstraete, J.-M. Beuken, Y. Pouillon, R. Caracas, F. Jollet, M. Torrent, G. Zerah, M. Mikami, P. Ghosez, M. Veithen, V. Olevano, L. Reining, R. Godby, G. Onida, D. Hamann and D. C. Allan

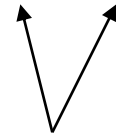
Z. Kristall., 220, 558-562 (2005)

Sequential calculations \Leftrightarrow one processor at a time

Parallel calculations \Leftrightarrow several processors in the same time



Rank	Site	Computer	Processors	Year	R _{max}	R _{peak}
1	DOE/NNSA/LLNL United States	BlueGene/L - eServer Blue Gene Solution IBM	212992	2007	478200	596378



These are Gflops / second (~ 0.5 petaflop)
= millions of operations / second

Earth Simulator

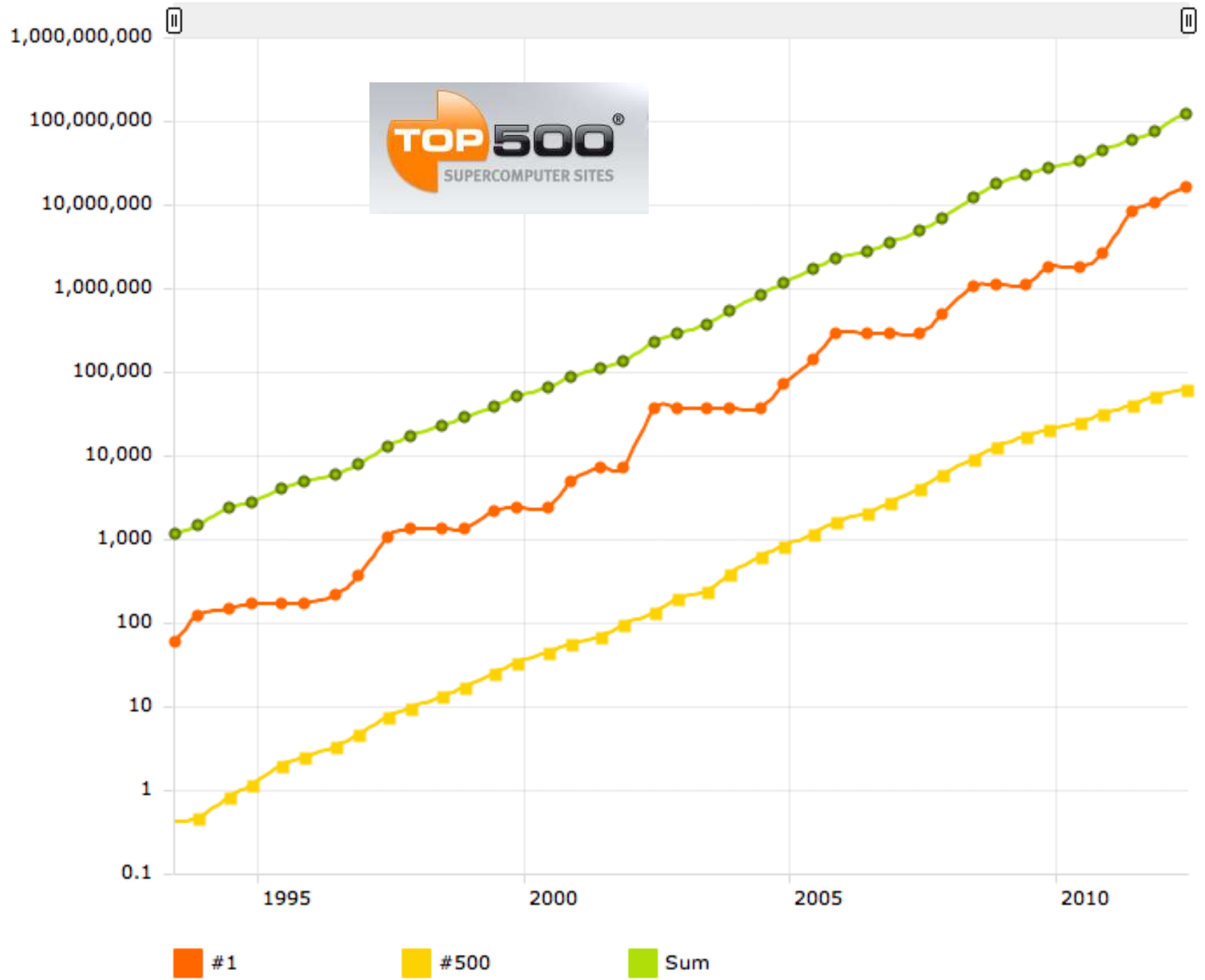


1 flop = 1 floating point operation / cycle

Itanium 2 @ 1.5 GHz \sim 6Gflops/sec = $6 \cdot 10^9$ operations/second



Performance (GFlop/s)



Welcome to the Partnerships Online Proposal System

Login

- Use TG Portal login
- Use POPS login
- Use NCSA Kerberos
- Use SDSC Kerberos

Manage Login

- TG Portal page
- Create POPS login
- Change POPS password
- NCSA Kerberos page
- SDSC Kerberos page

Support

- View POPS User Guide
- Contact POPS staff

(You cannot view your proposal after the deadline un

What's new:

Systems: Steele Cluster (Purdue), Kraken Upgrad

See www.teragrid.org/userinfo/hardware/resou

Request "Advanced Support Program" (ASP) for scalability and/or integration into Science Gateway

Guidelines for writing a successful gateway propo

New names: Startup/Educational Allocation has r Research Allocation request pool will not have the limited.

The Research Allocation request (proposal) docum requests greater than 10,000,000 SUs or Multi-ye

Welcome to POPS



[Transnational Access](#)
[HPC-Europa Archive](#)
[More info](#)

The Consortium is happy to announce that the Transnational Access Programme will continue up to the end of 2012.

Next deadline for submitting applications is 15 November 2008.

All the applications submitted from now on will be selected by the new selection panel and, if accepted, funded under the HPC-EUROPA2 Project.

The whole project website will be renewed.

The rules for the Access are unchanged, but researchers can benefit from a greater allocation of computational resources.

Are you a scientist of postgraduate level or above, working in the EU or an Associated State? Do you require large computing power to improve your research?



- Transnational Access
 - Call
 - Access Point

Transnational Access Programme will continue up to the end of 2012

Welcome to the TeraGrid User Portal

About

The TeraGrid User Portal is a Web interface for making TeraGrid account management easier, for getting information about TeraGrid resources, and for accessing many of the existing TeraGrid services in a single place.

While users may utilize many features of the User Portal without logging in, authenticating provides access to a full set of services available on the TeraGrid. All new users will receive a "New User Form" via U.S. postal mail containing a User Portal username and password along with their other TeraGrid system account usernames and passwords.

Feature Spotlight

TeraGrid User Portal Now Offers File Manager Service

The NSF TeraGrid project exposes the academic research community to more than twenty distributed computational, visualization, and storage resources. Consequently, a significant user interest has grown in having the ability to remotely manage files via a web interface. In response to this need, the TeraGrid User Portal development team is happy to announce the availability of a new beta file manager service. Available through the TeraGrid User Portal, this file manager includes the following features:

- A custom list of
- The ability to tr
- An easy to use
- The existence of
- The ability to vi
- Support for tra

To manage your files manager service.

Login

User Name

Password

Remember my login

[Forget your password?](#)

[Need help logging in?](#)

[Report Security Incident](#)

System Accounts					
Resource Name	Login Name	Institution	Username	Connect	
Abe	login-abe.ncsa.teragrid.org	NCSA	rcaracas	Login	
Big Red	login.bigred.iu.teragrid.org	IU	tg-rcarac	Login	
BigBen	tg-login.bigben.psc.teragrid.org	PSC	no account		
Cobalt	login-co.ncsa.teragrid.org	NCSA	rcaracas	Login	
Condor	tg-condor.purdue.teragrid.org	Purdue	rcaracas	Login	
Frost	tg-login.frost.ncar.teragrid.org	NCAR	no account		
Kraken	kraken.nics.teragrid.org	NICS	no account		
Lonestar	tg-login.lonestar.tacc.teragrid.org	TACC	tg458681	Login	
Maverick	tg-viz-login.tacc.teragrid.org	TACC	tg458681	Login	
NCSA TeraGrid Cluster	tg-login.ncsa.teragrid.org	NCSA	rcaracas	Login	
NSTG	tg-login.ornl.teragrid.org	ORNL	rcaracas	Login	
Pople	tg-login.pople.psc.teragrid.org	PSC	no account		
Queen Bee	queenbee.loni-lsu.teragrid.org	LONI	rcaracas	Login	
Ranger	tg-login.ranger.tacc.teragrid.org	TACC	no account		
SDSC TeraGrid Cluster	tg-login.sdsc.teragrid.org	SDSC	rcaracas	Login	
Steele	tg-steele.purdue.teragrid.org	Purdue	no account		
UC/ANL TeraGrid Cluster	tg-login.uc.teragrid.org	UC/ANL	rcaracas	Login	

RUN MD CODE

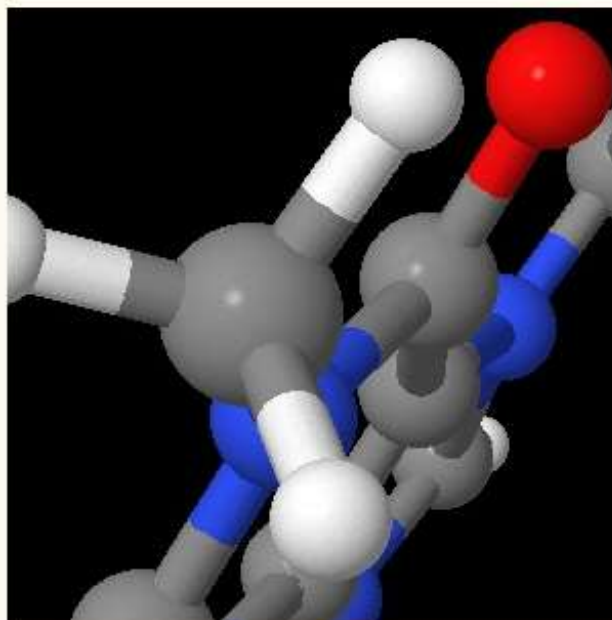
```
[rcaracas@x5650comp2 tutorial-cider]$ rm *IM*
[rcaracas@x5650comp2 tutorial-cider]$ ll
total 22582
-rw-r--r-- 1 rcaracas geol 186497 Jul 25 06:52 C.psp.paw
-rw-r--r-- 1 rcaracas geol 91 Jul 25 06:53 fe3c.P400.files
-rw-r--r-- 1 rcaracas geol 2894 Jul 25 08:10 fe3c.P400.inp
-rw-r--r-- 1 rcaracas geol 21133776 Jul 25 08:51 fe3c.P400.log
-rw-r--r-- 1 rcaracas geol 10682 Jul 25 08:51 fe3c.P400.OU_DDB
-rw-r--r-- 1 rcaracas geol 10808 Jul 25 07:43 fe3c.P400.output
-rw-r--r-- 1 rcaracas geol 12288 Jul 25 07:46 fe3c.P400.outputA
-rw-r--r-- 1 rcaracas geol 345431 Jul 25 08:08 fe3c.P400.outputB
-rw-r--r-- 1 rcaracas geol 456223 Jul 25 08:51 fe3c.P400.outputC
-rw----- 1 rcaracas geol 582 Jul 25 06:54 fe3c.P400.run
-rwxr-xr-x 1 rcaracas geol 318 Jul 25 07:42 fe3c.P400.scr*
-rwxr-xr-x 1 rcaracas geol 566099 Jul 25 06:52 Fe.psp.paw*
[rcaracas@x5650comp2 tutorial-cider]$ █
```


Jmol



Jmol: an open-source Java viewer for chemical structures in 3D

with features for chemicals, crystals, materials and biomolecules



Jmol is an interactive web browser applet.

This is a still image, but you can get an animated display of Jmol abilities by clicking [here](#).

(The applet may take some seconds to load. Please, wait and do not reload the page in the meantime.)

EXTRACT RELEVANT INFORMATION:

Atomic positions

Atomic velocities

Energy

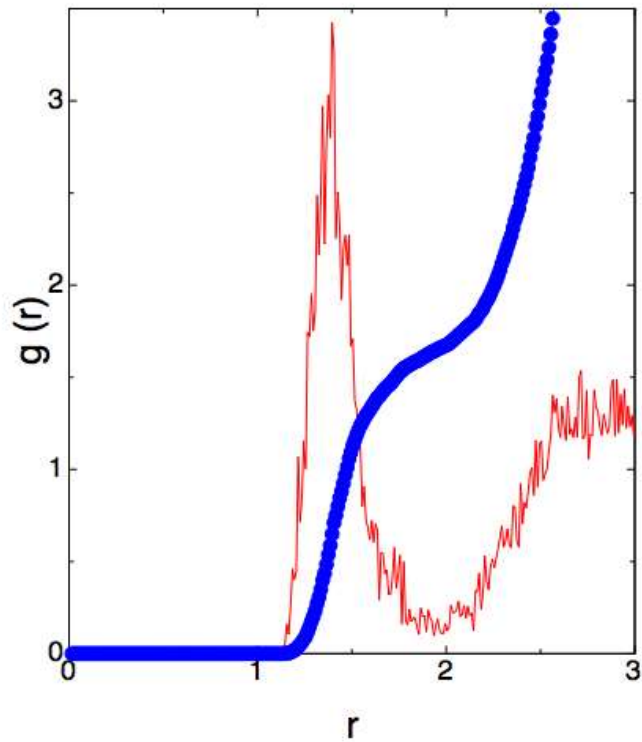
Stress tensor

VISUALIZE SIMULATION (ex: jmol, vmd)

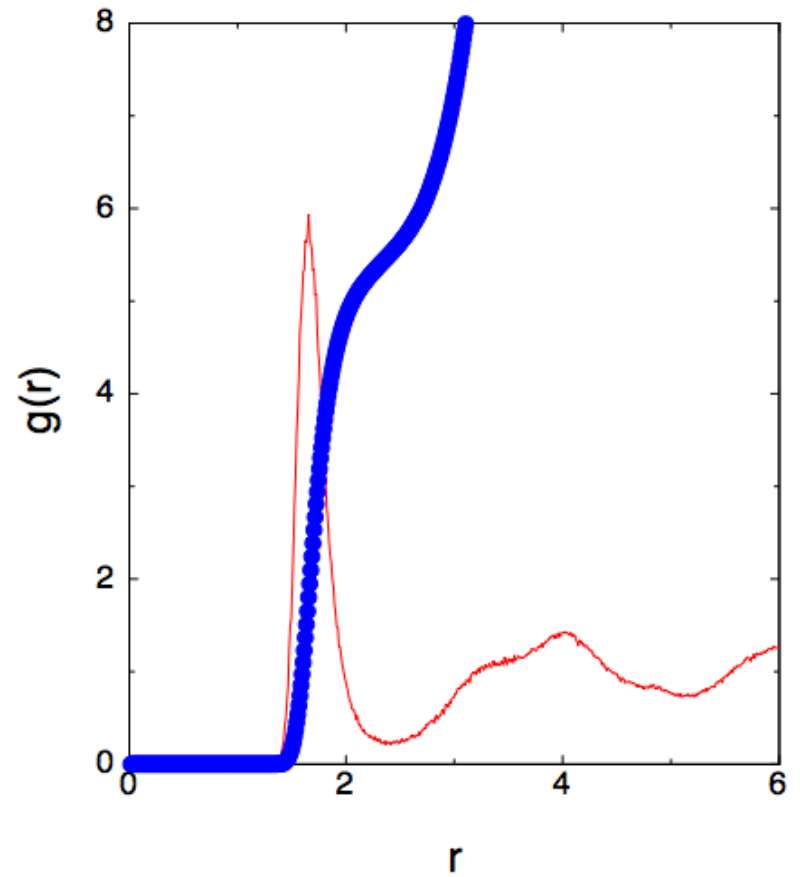
PERFORM STATISTICS

Ex: coordination in forsteritic melt at mid-mantle conditions

C-O



Si-O



Design of Catalysts and Electrocatalysts: From DFT Prediction to Experimental Verification

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Email: jgchen@columbia.edu

CFN Workshop, Nov. 5, 2014

Development of Novel Catalysts

- Supported catalysts:
 - More relevant to commercial catalysts and processes
 - Fast (high throughput) evaluation
 - “Heterogeneous” in electronic and catalytic properties
- Single crystal surfaces:
 - Atomic level understanding from experiments and theory
 - Materials gap: single crystal vs. polycrystalline materials
 - Pressure gap: ultrahigh vacuum (UHV: $\sim 10^{-12}$ psi)
- Need to bridge “materials gap” and “pressure gap”

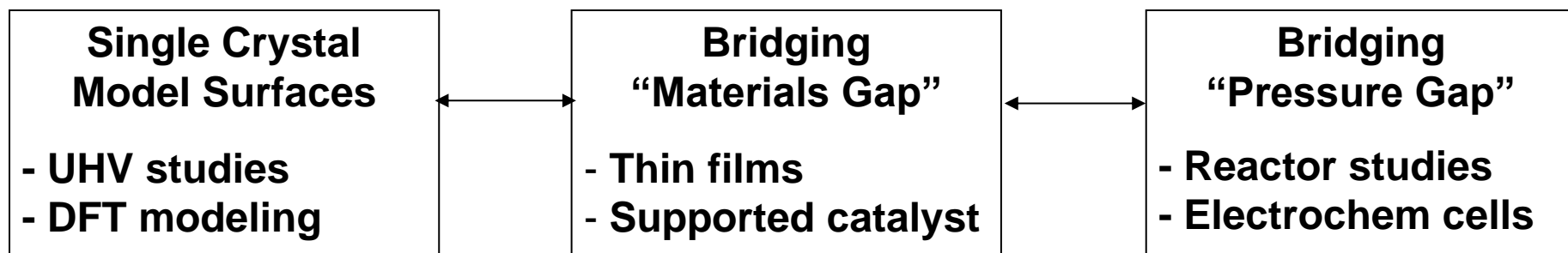
From DFT Prediction to Experimental Verification



Use DFT to assist catalysts design: (**activity, selectivity, stability, cost**):

- Binding energy calculations (activity, stability)
- Activation barriers and reaction network (selectivity)

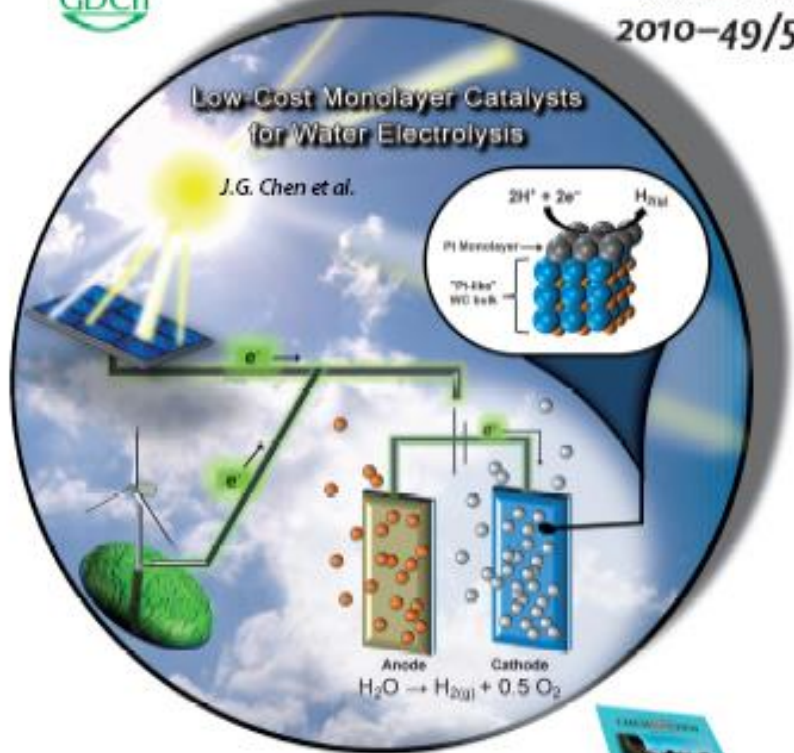
Outline of Presentation



Examples of DFT prediction and experimental verification:

- Correlating hydrogen binding energy (HBE) with water electrolysis activity
- Correlating hydrogen binding energy (HBE) with hydrogenation activity
- Correlating activation barrier with hydrogenation selectivity

Correlating HBE with Water Electrolysis Activity



Esposito, Hunt & Chen,
Angew. Chem. Int. Ed.
49 (2010) 9859

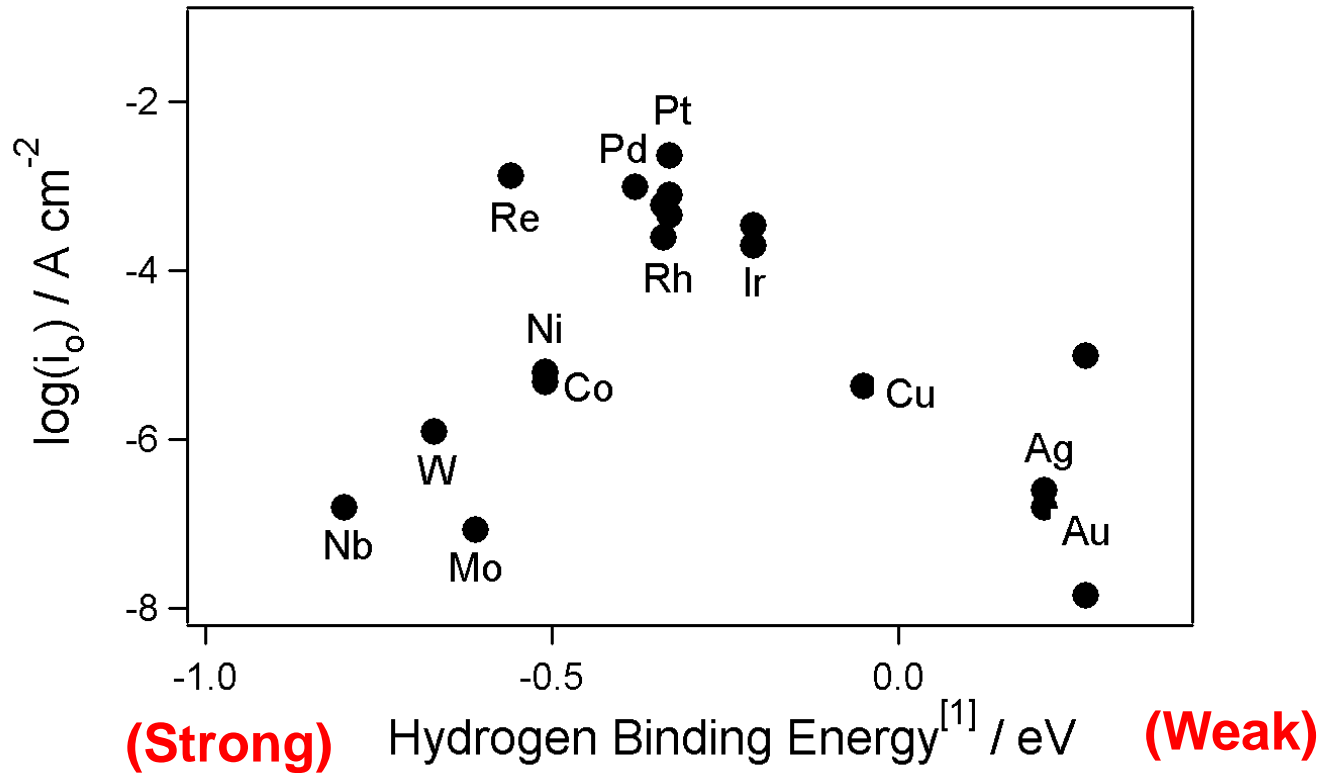
- Metamaterials
H. Giessen and N. Liu
- Metal-Salen Complexes
A. W. King et al.
- Highlights: CuH Catalysis • NIR Fluorescence Probes • Platinum Electrocatalysts

ANGEWANDTE CHEMIE 49 (51) 9787-9804 (2010) - ISSN 1433-7875 - VOL. 49, NO. 51

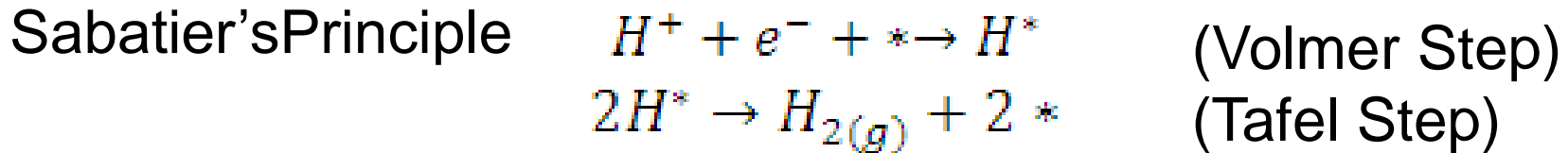


WILEY-VCH

HER Activity and Hydrogen Binding Energy (HBE)

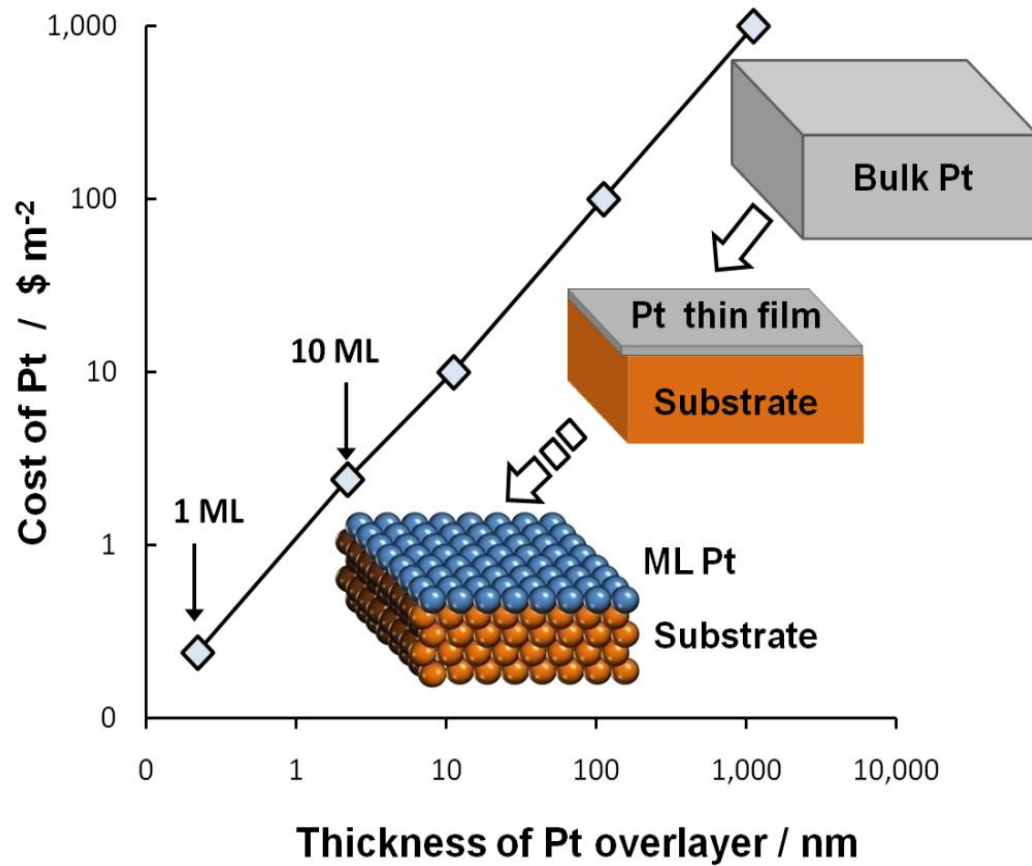


- Classic volcano curve observed for the HER is explained by Sabatier's Principle



[1] Data from: Norskov, Bligaard, Logadottir, Kitchin, Chen, Pandelov, Stimming, J.Electrochem. Soc., 152 (2005) J23-26.

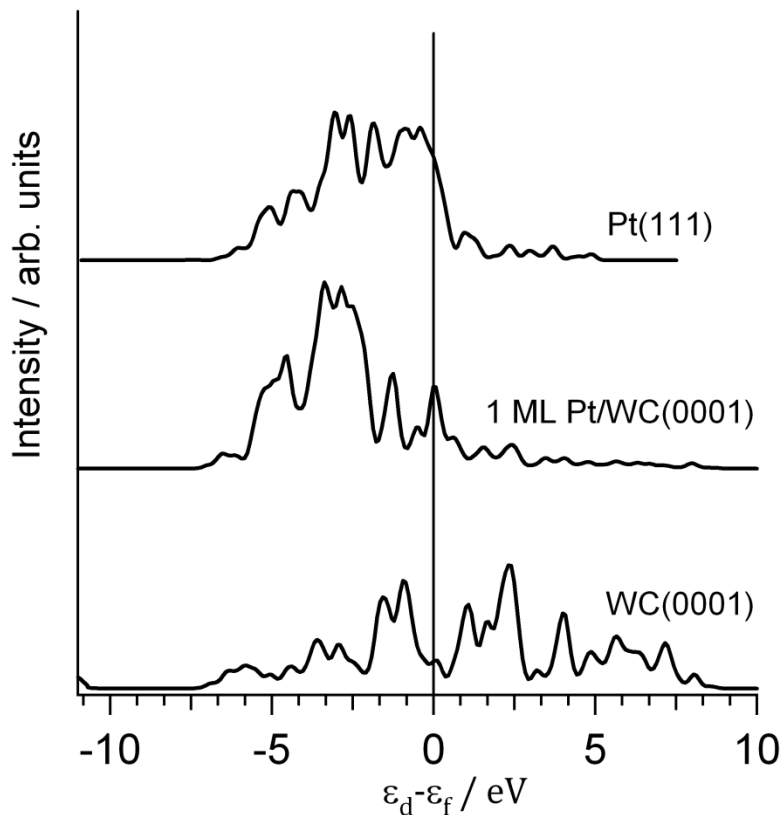
Reduce Pt Loading with Monolayer (ML) Pt



Goal: Supporting ML Pt on **Pt-like** substrates, such as WC

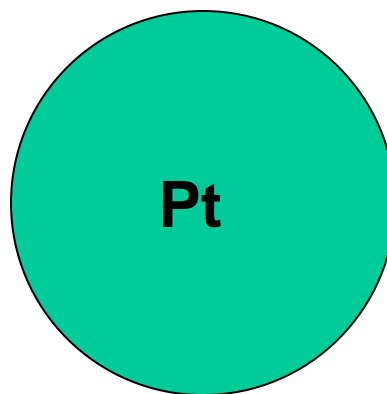
DFT Prediction: Similar HBE Values between Monolayer Pt-WC and Bulk Pt

d-band density of states

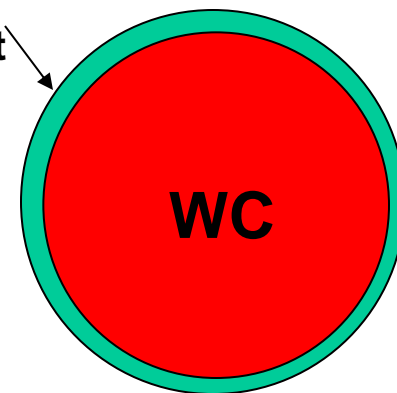


Surface	HBE (eV)
WC(001)	-0.99
Pt(111)	-0.46
1 ML Pt-WC(001)	-0.43

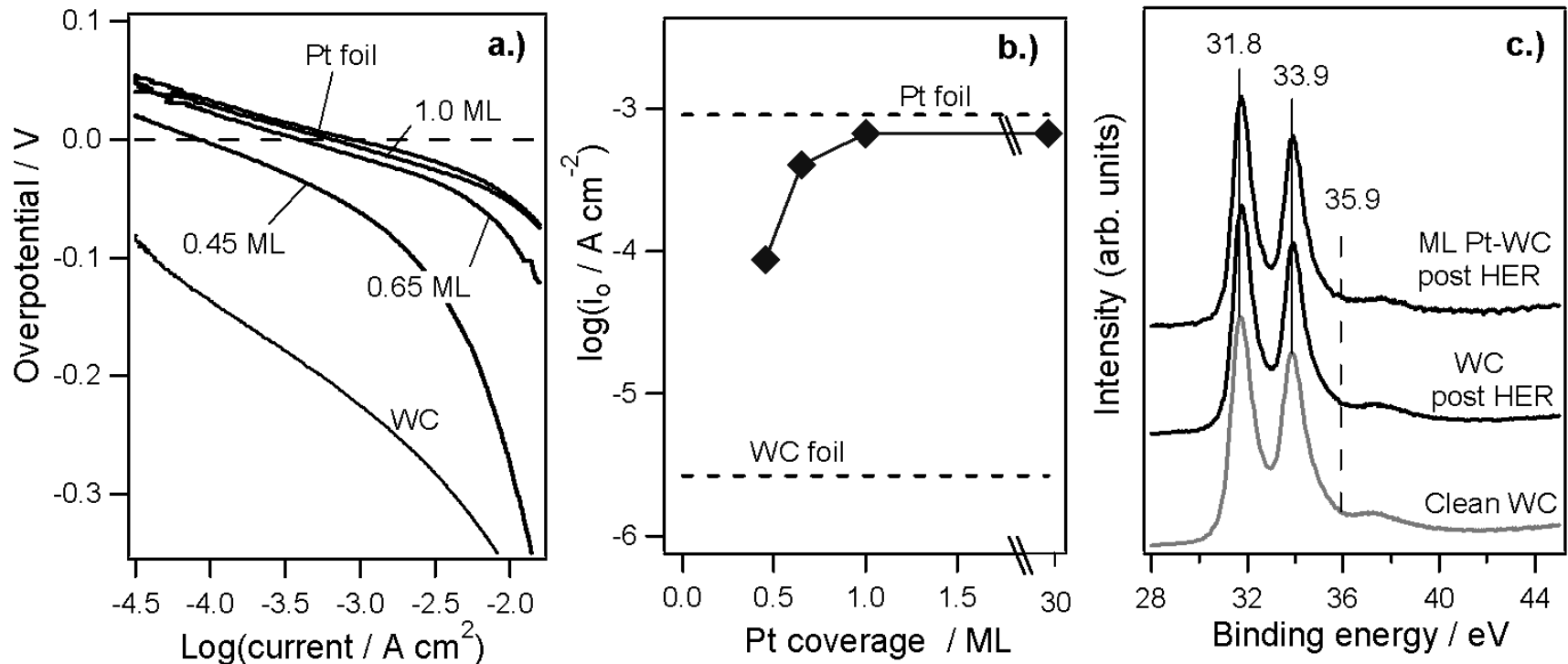
DFT-calculated per-atom hydrogen binding energy (HBE) for WC, Pt, and 1 ML Pt-WC surfaces with a hydrogen coverage of 1/9 ML.



1 atomic layer of Pt



Experimental Verification of Activity and Stability



HER Activity of 1 ML Pt/WC approaches to that of Pt foil

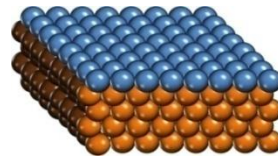
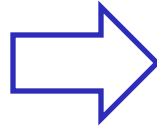
Esposito, Hunt, Birmire & Chen, *Angew. Chem. Int. Ed.* 49 (2010) 9859

DFT Prediction of Stability of Pt/WC and Pt/C

- Use DFT to compare adhesion of Pt atoms to WC and Pt surfaces:

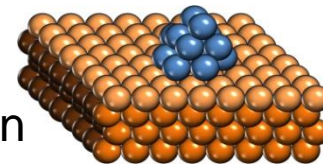
Binding Energy

Pt-(Substrate) < Pt-Pt



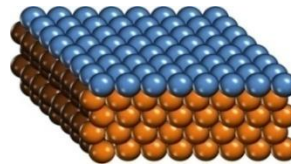
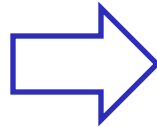
Outcome

Pt
migration



Particles favored

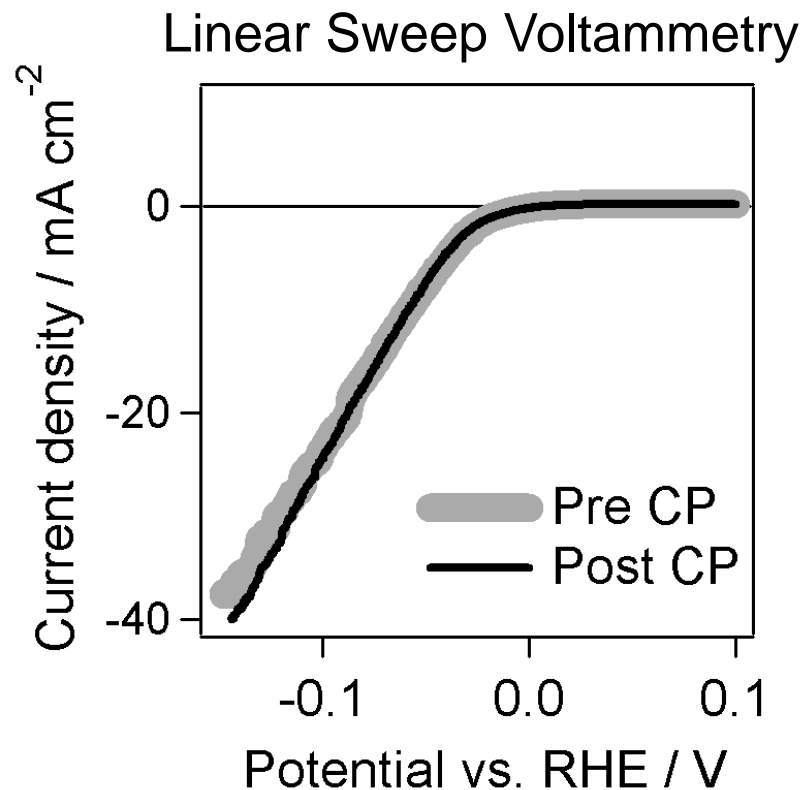
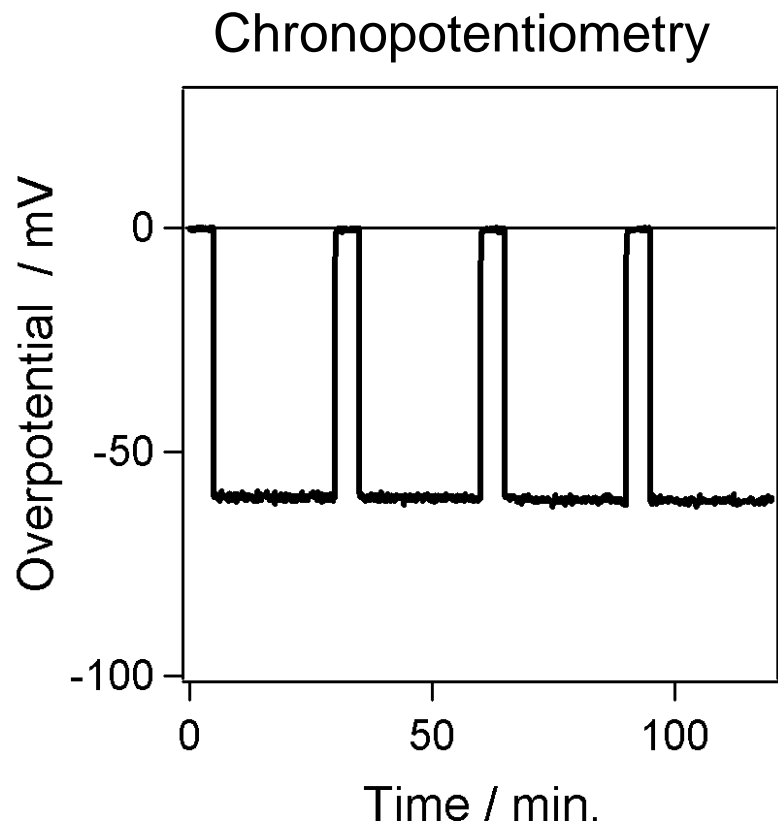
Pt-(Substrate) > Pt-Pt



ML configuration favored

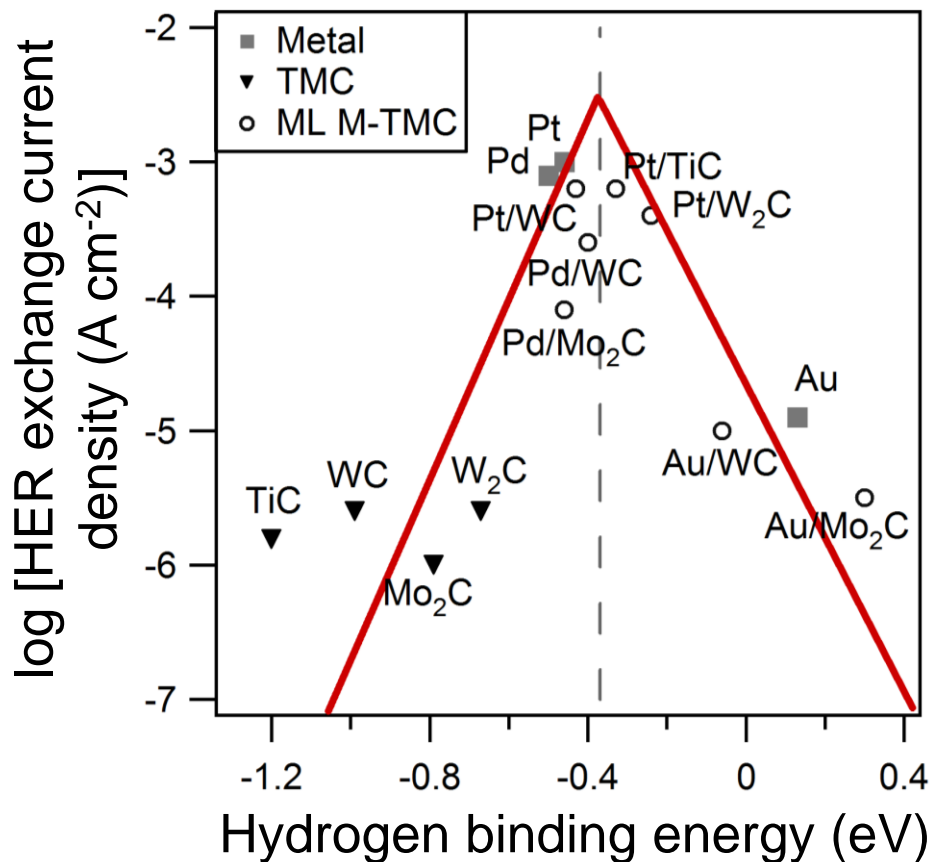
ML surface atoms	Substrate	Binding energy / eV	(M-X [^]) - (M-M) BE / eV
Pt	Pt(111)	-5.43	0.00
	C(0001)	-4.12	1.31
	WC(0001)	-6.59	-1.16
	W ₂ C(0001)	-6.51	-1.08

Experimental Verification of HER Stability



- No change in overpotential observed with time
- No change in LSV before and after CP
- XPS and SEM measurements confirmed stability

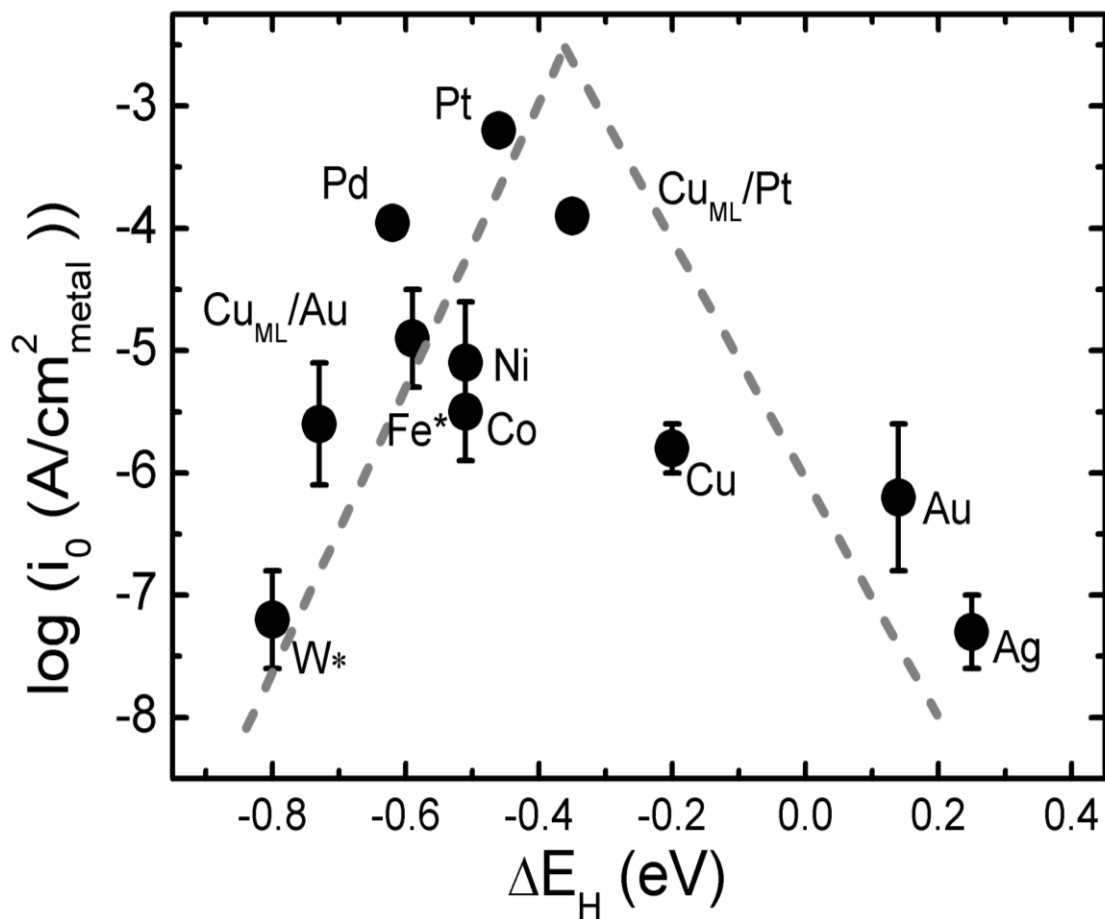
Other ML/TMC Electrocatalysts for HER in Acid



Volcano relationship provides design principles of electrocatalysts

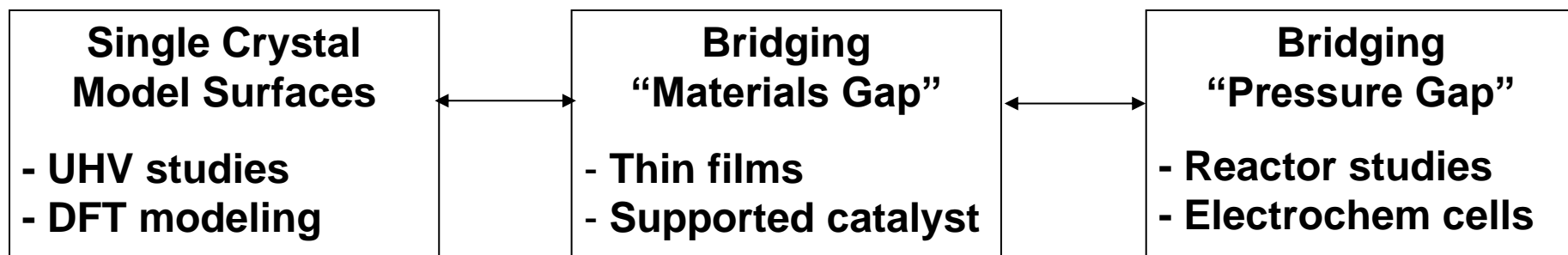
KimmeI , Yang & Chen, *J. Catalysis*, 312 (2014) 216

HER Catalysts in Alkaline Environment



Volcano relationship also appears to hold in alkaline electrolyte

Outline of Presentation

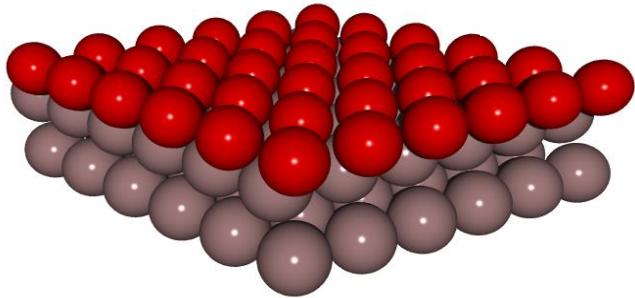
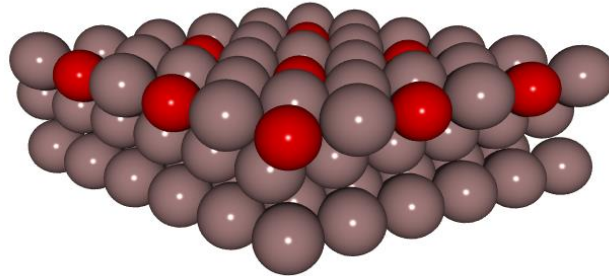


Examples of DFT prediction and experimental verification:

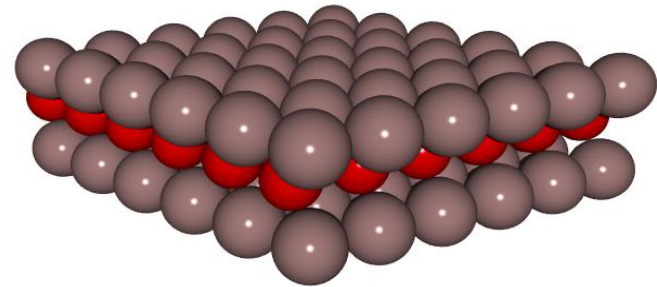
- Correlating hydrogen binding energy (HBE) with water electrolysis activity
- **Correlating hydrogen binding energy (HBE) with hydrogenation activity**
- Correlating activation barrier with hydrogenation selectivity

Monolayer Bimetallic Surfaces

Surface Alloy



Surface Monolayer

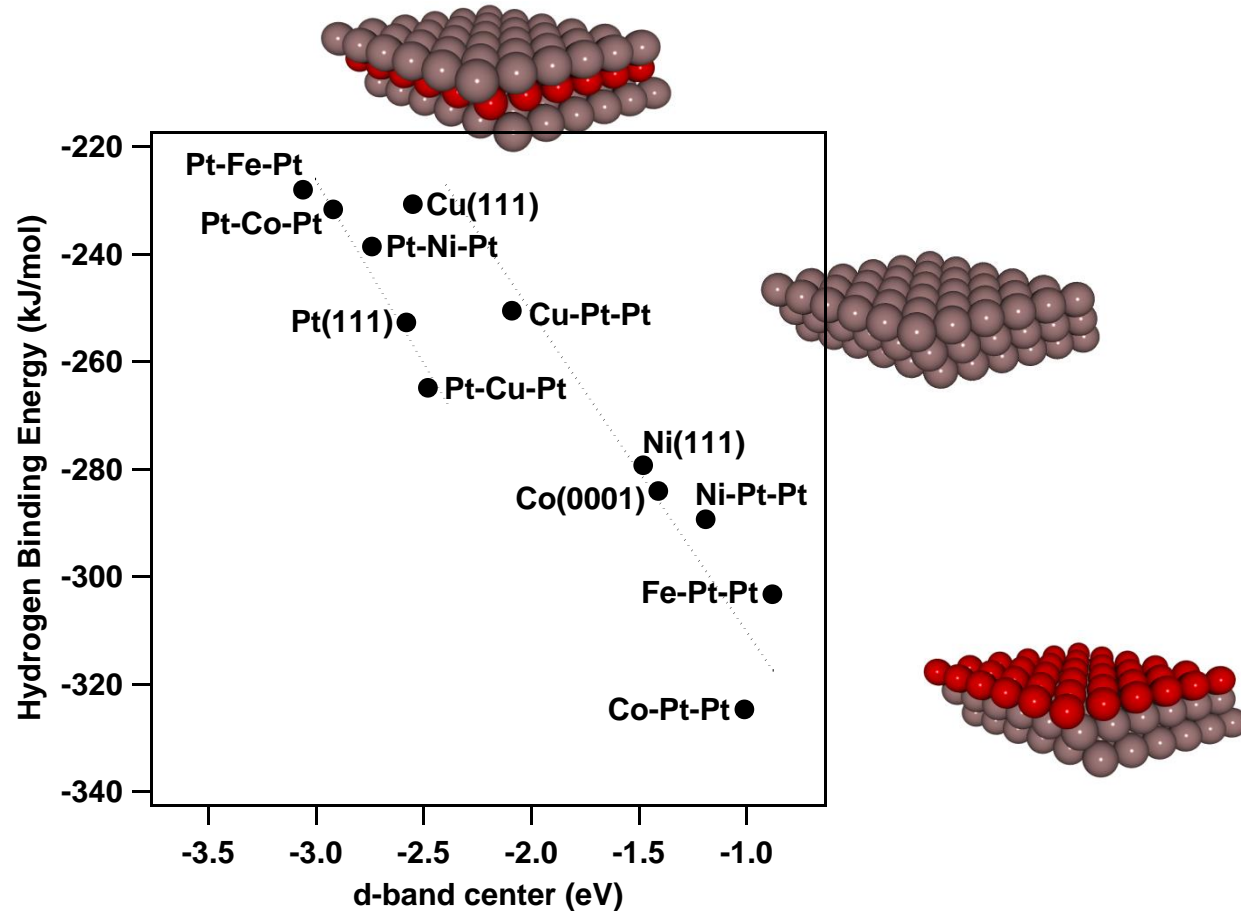


Subsurface Monolayer

Modified surface chemical properties due to:

- Ligand effect – electronic configuration
- Compressive and tensile strain - lattice mismatch

DFT Prediction of HBE Values



Hydrogen binding energy (HBE) can be controlled by surface structures

Kitchin, Norskov, Barteau & Chen, *Phys. Rev. Lett.* 93 (2004) 156801

Murillo, Goda & Chen, *J. Am. Chem. Soc.* 129 (2007) 7101

Controlling Hydrogenation **Activity:** Correlating with Binding Energy

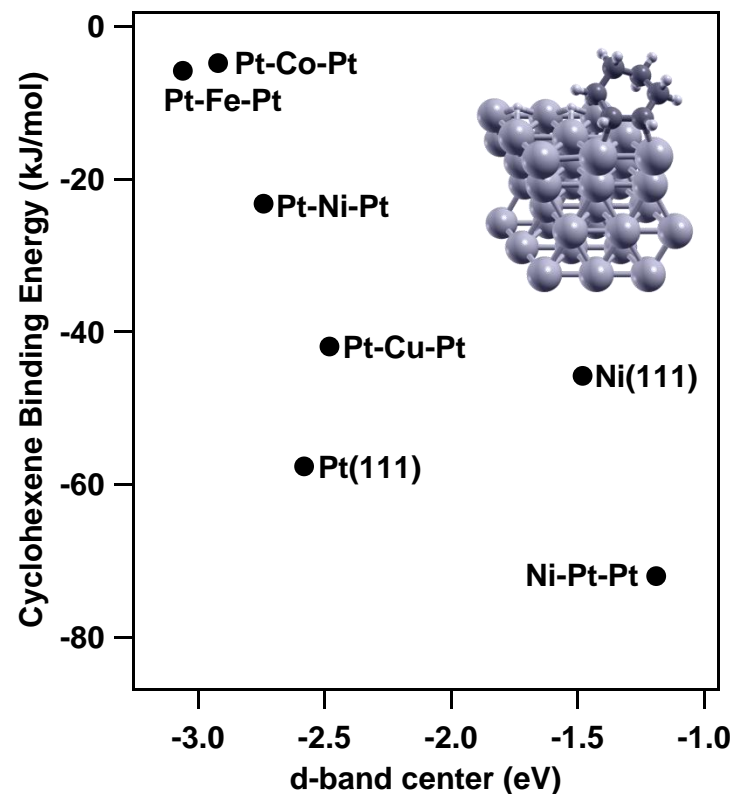
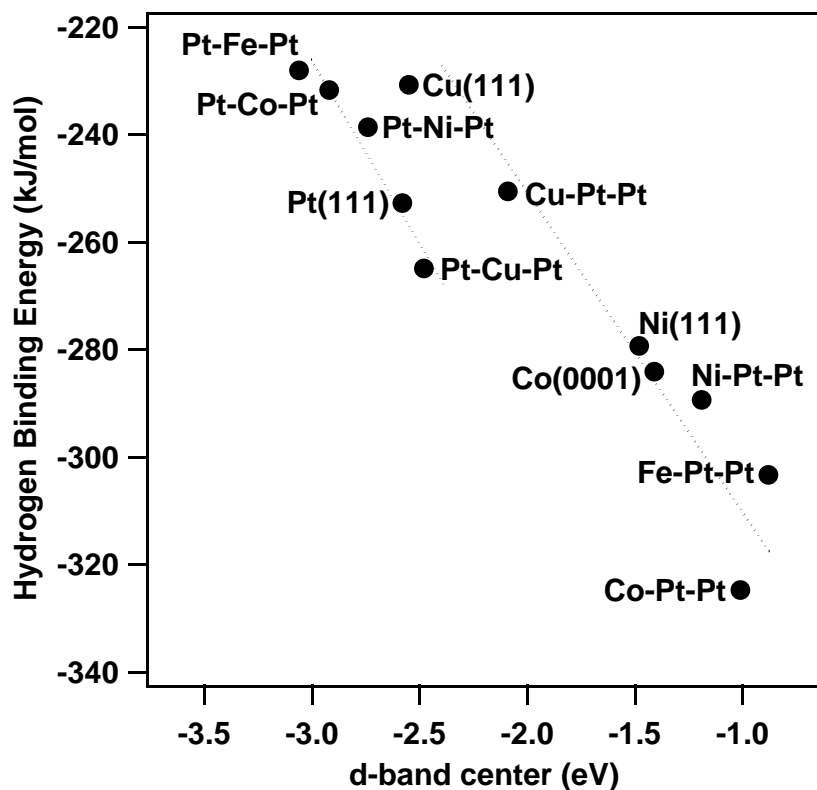
Low-Temperature Cyclohexene Hydrogenation:



Assumption for Higher Hydrogenation Activity:

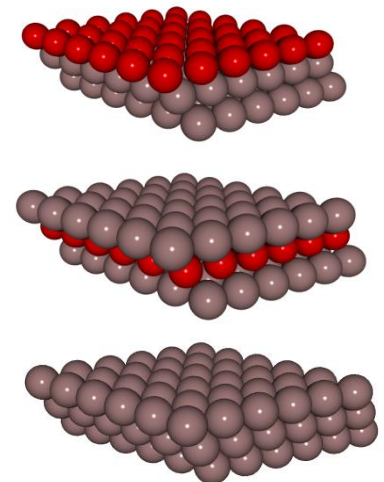
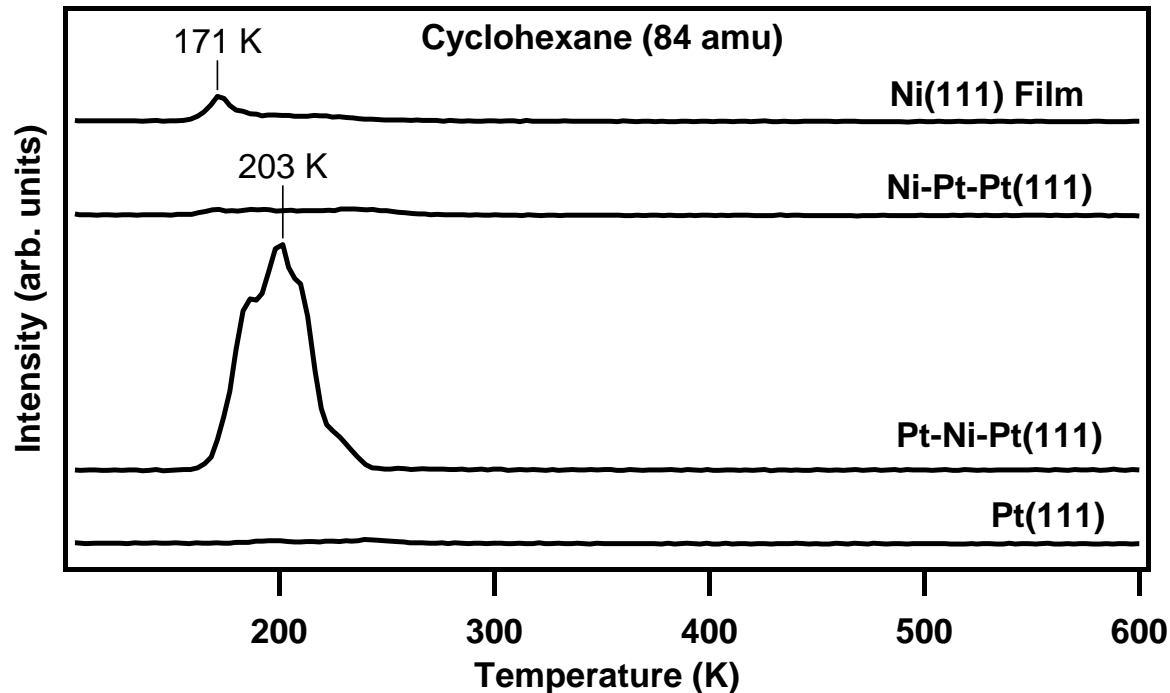
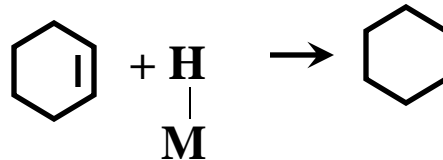
- Weakly bonded H atoms
- Weakly bonded cyclohexene

DFT Calculations of Binding Energies of Hydrogen and Cyclohexene



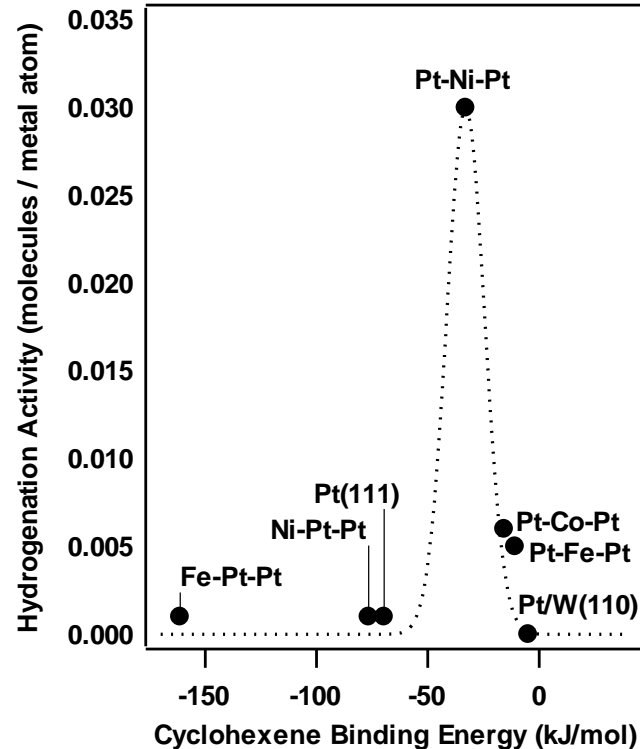
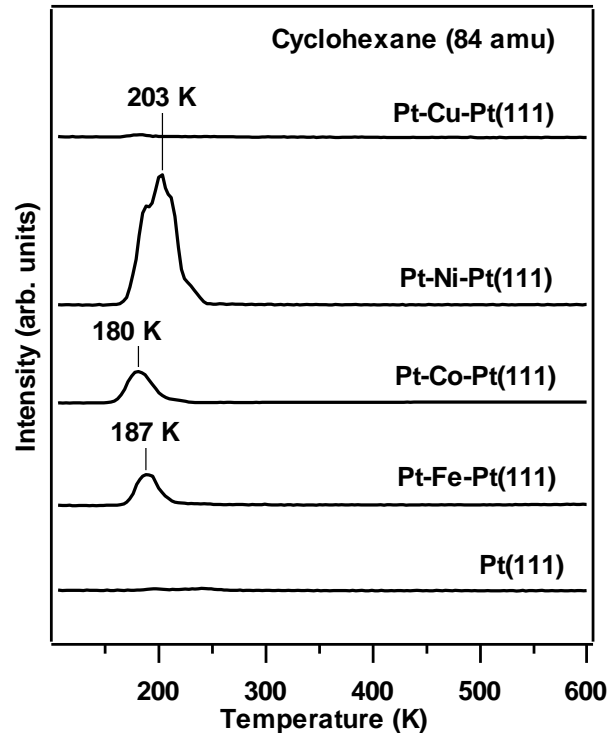
BE values follow the same trend: Ni-Pt(111) > Ni ~ Pt > Pt-Ni-Pt(111)

Low-Temperature Hydrogenation of Cyclohexene Due to Weakly Bonded H



- Weakly bonded M-H leads to low-T hydrogenation

Binding Energies Correlate with Cyclohexene Hydrogenation Activity



Sabatier's principle: not too strong, not too weak!

Volcano relationship allows prediction of hydrogenation activity

DFT Prediction of Stable Bimetallic Structures

General Stability includes:

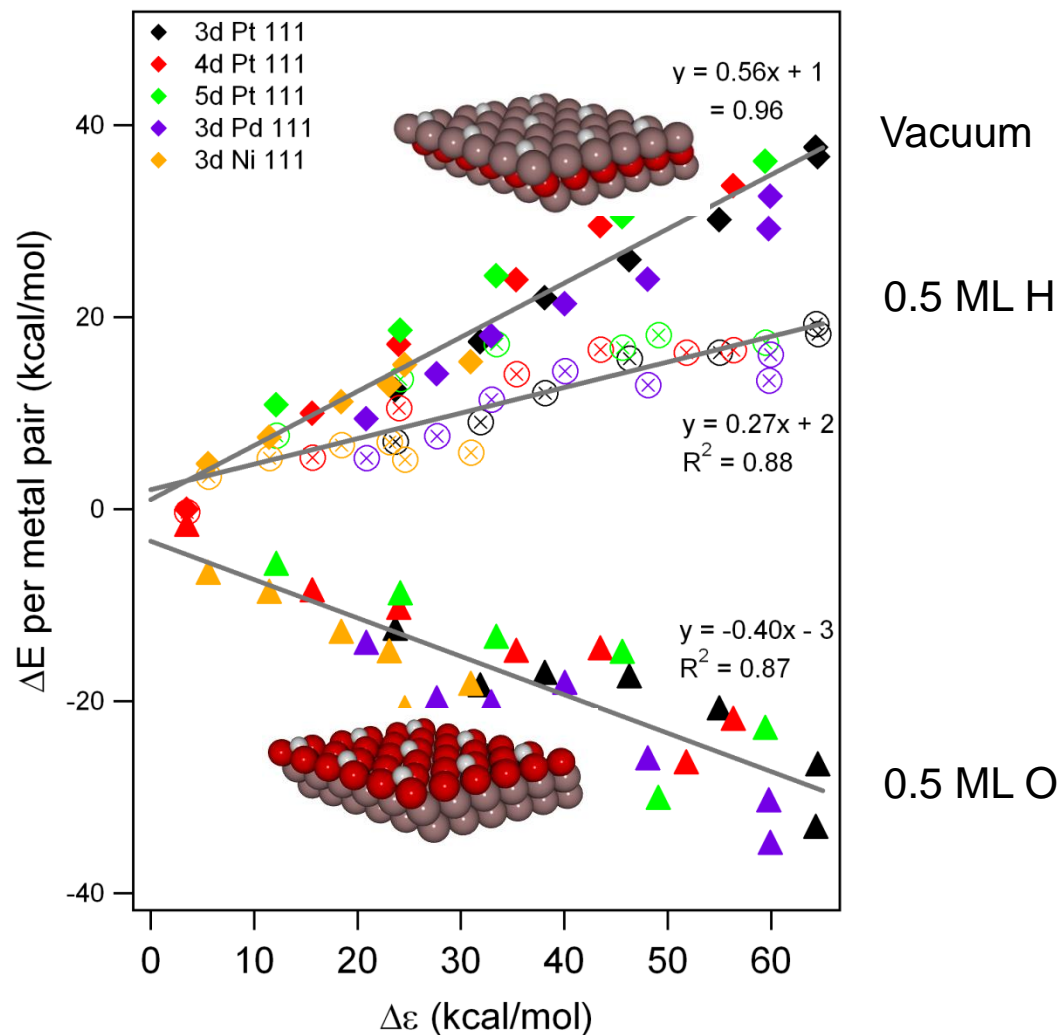
Admetals – 3d, 4d, 5d

Host metals – Ni, Pd, Pt

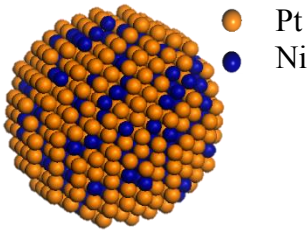
Ti	V	Cr	Mn	Fe	Co	Ni
Zr	Nb	Mo	Tc	Ru	Rh	Pd
Hf	Ta	W	Re	Os	Ir	Pt

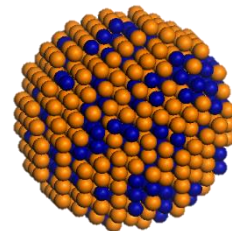
Menning & Chen,

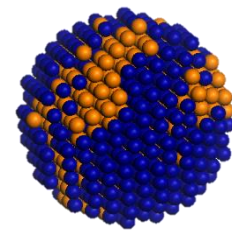
J. Chem. Phys., 130 (2009) 174709



Experimental Verification of Bimetallic Structures

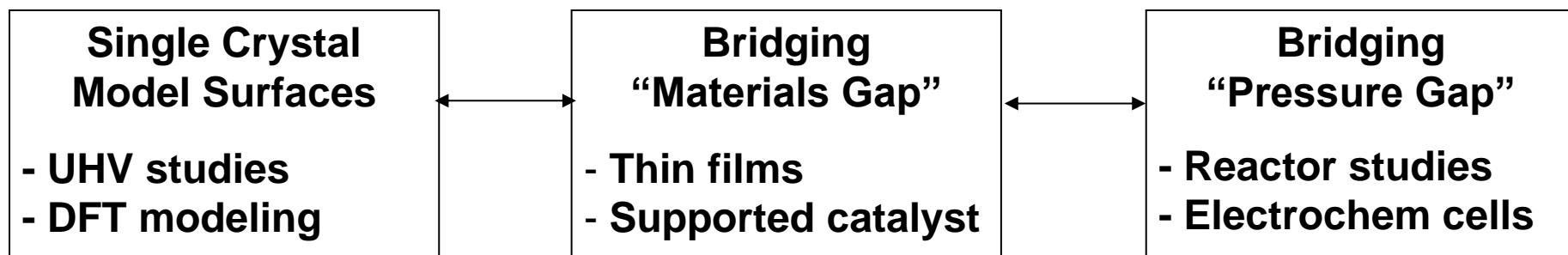
<i>Coordination Number</i>	<i>10% H₂ at 50 °C—NiPt/C</i>		
	Experimental	Simulation	
Pt-Pt	1.9 ± 0.8	2.696 ± 0.003	
Pt-Ni	3.5 ± 0.4	3.904 ± 0.007	

<i>Coordination Number</i>	<i>10% H₂ at 225 °C—NiPt/C</i>		
	Experimental	Simulation	
Pt-Pt	2.0 ± 0.7	2.675 ± 0.005	
Pt-Ni	3.8 ± 0.3	4.148 ± 0.008	

<i>Coordination Number</i>	<i>APR at 225 °C—NiPt/C</i>		
	Experimental	Simulation	
Pt-Pt	6.0 ± 1.4	4.429 ± 0.006	
Pt-Ni	1.9 ± 0.8	2.24 ± 0.03	

Tupy, Karim, Vlachos, Chen, *ACS Catalysis*, 2 (2012) 2290

Outline of Presentation

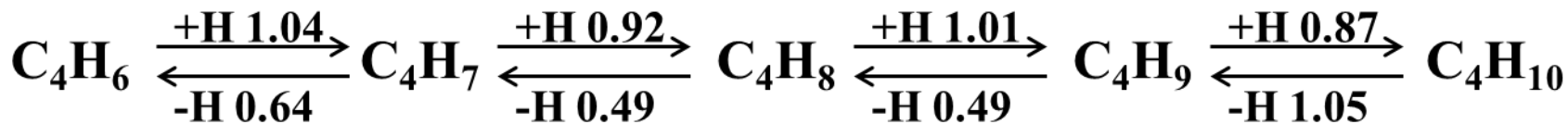
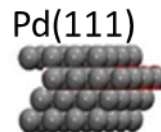


Examples of DFT prediction and experimental verification:

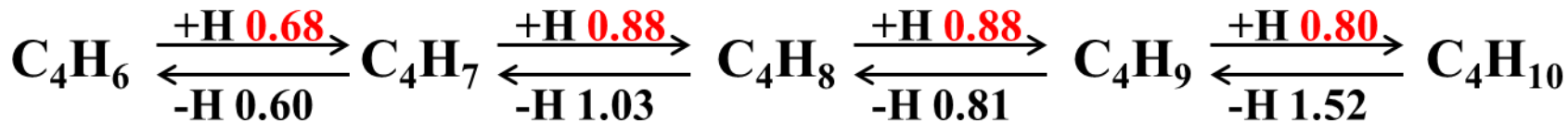
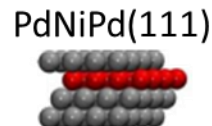
- Correlating hydrogen binding energy (HBE) with water electrolysis activity
- Correlating hydrogen binding energy (HBE) with hydrogenation activity
- **Correlating activation barrier with hydrogenation selectivity**

Selective Hydrogenation Requires More Complicated DFT Calculations

Activation barriers on Pd(111) surface (eV):



Activation barriers on PdNiPd(111) surface (eV):



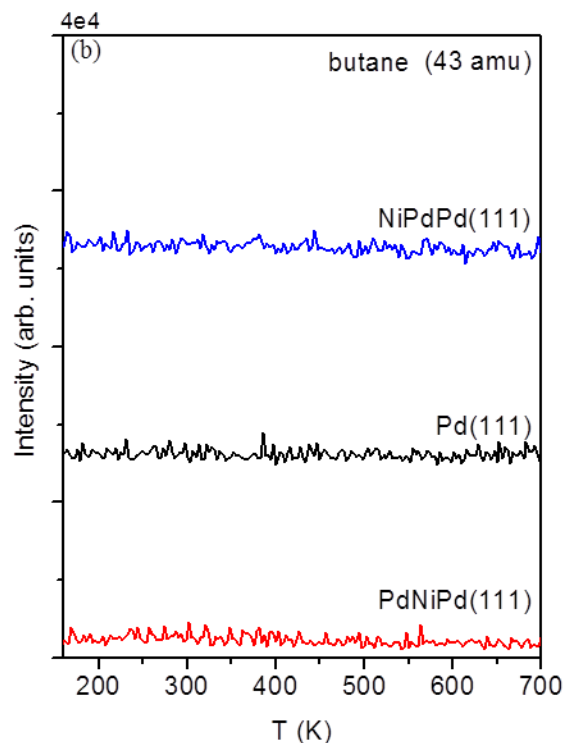
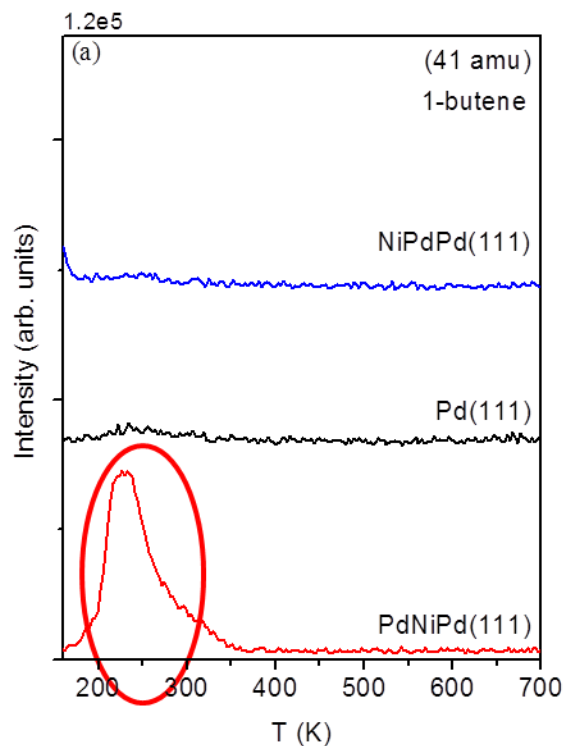
The activation barriers are generally lower on PdNiPd(111) than on Pd(111), leading to higher hydrogenation activity on PdNiPd(111)

Predicting Selectivity Requires DFT Calculations of Reaction Network

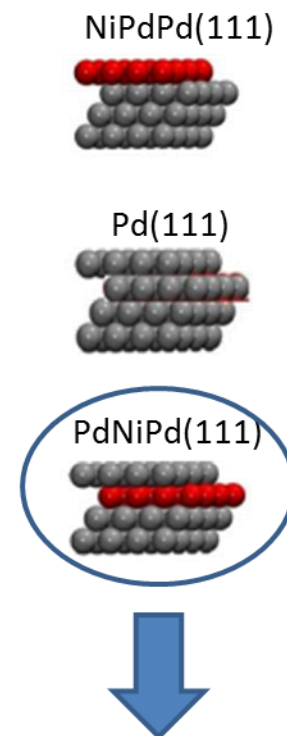
Surface		Pd(111)	PdNiPd(111)
	d-band center (eV)	-1.90	-2.25
Binding energy (kcal/mol)	C ₄ H ₆	-34.47	-19.19
	C ₄ H ₇	-37.27	-13.48
	C ₄ H ₈	-12.68	-2.90
	C ₄ H ₉	-36.54	-29.67
	C ₄ H ₁₀	-2.61	-2.54

Binding energy of butene is weaker on PdNiPd(111), leading to higher selectivity for butene production on PdNiPd(111)

Experimental Verification on Model Surfaces



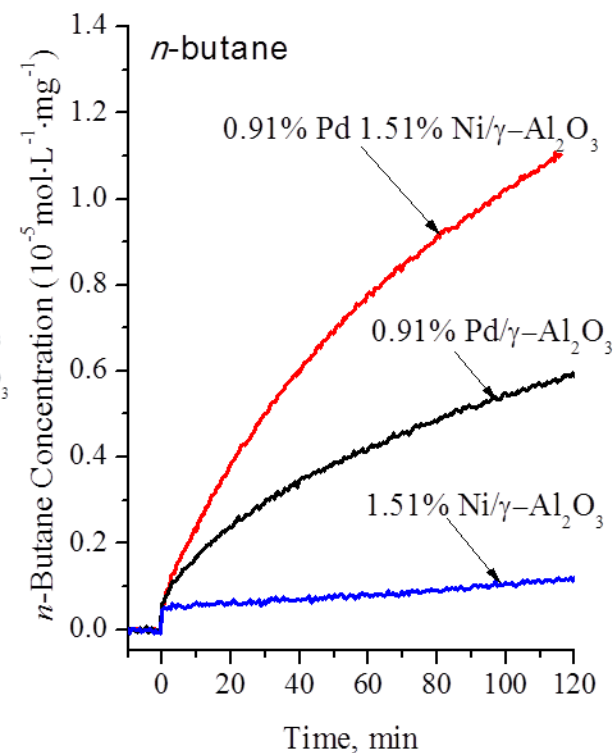
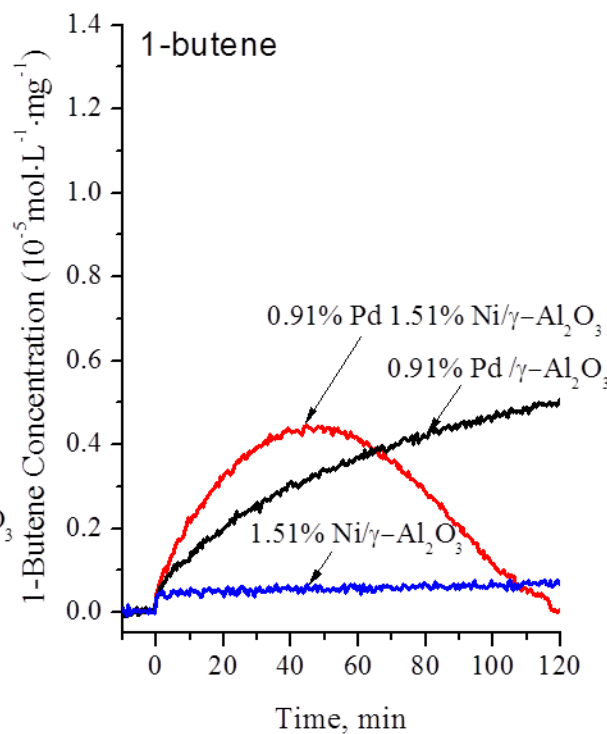
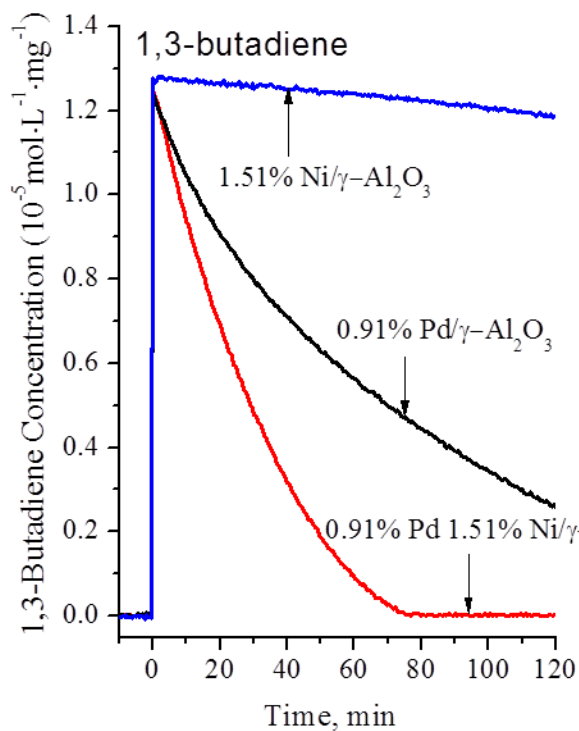
UHV-TPD: Ultra-High Vacuum Temperature Programmed Desorption



Stable bimetallic structure
under hydrogenation conditions

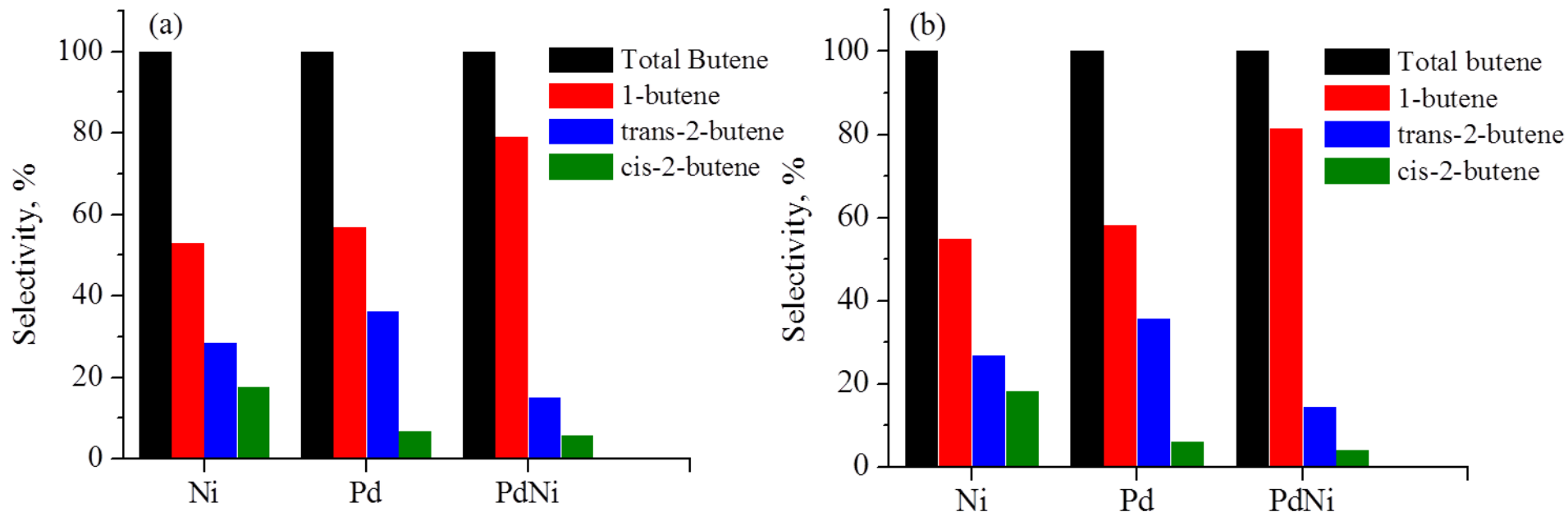
PdNiPd bimetallic structure is **very active for 1,3-butadiene hydrogenation**, and may also be selective for 1-butene production

Batch Reactor: Hydrogenation Activity



Activity trend: PdNi > Pd >> Ni

Flow Reactor: Hydrogenation Selectivity

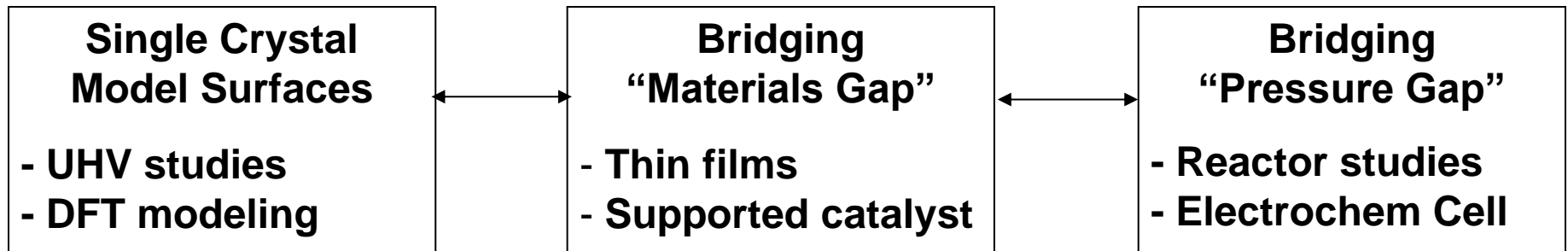


Selectivities in flow reactor at conversions of (a) 10% (b) 60%
 $\text{H}_2:\text{C}_4\text{H}_6 = 2.2:1$ Total Flow Rate: 9.6 ml/min

PdNi shows higher 1-butene selectivity than Pd,
and higher yield in producing 1-butene

Conclusions

- Bimetallic and carbide catalysts offer the advantages of **reduced cost** and **enhanced activity, selectivity and stability**
- Combined theory, surface science, and catalytic studies are critical in **design** of novel catalytic materials



Review: Yu, Porosoff & Chen, "Review of Pt-based Bimetallic Catalysis: From Model Surfaces to Supported Catalysts", *Chemical Reviews*, 112 (2012) 5780